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Effects of temperature, pressure and chemical compositions on the electrical conductivity of carbonated melts and its relationship with viscosity

David Sifrê, Leïla Hashim, Fabrice Gaillard

Abstract

Carbonated melts constitute a key medium in the global deep carbon cycle: their impact on the geochemical signature of deep rocks is well studied because of their role as metasomatic agents in the deep mantle. However, their physical properties and in particular their electrical conductivity at high temperature and high pressure remain poorly constrained. In this study, we investigated the effect of chemical composition on the electrical conductivity of carbonated melts. We characterized this effect for various temperatures (1000–1700 °C) and pressures (1 to 4 GPa). Measurements show a very high electrical conductivity (> 100 S·m⁻¹) with weak temperature, pressure and chemical composition dependence. Carbonated melts are five orders of magnitude more conductive than mantle olivine, and up to two orders of magnitude more conductive than basalts at similar T and P. The electrical conductivity of molten carbonates follows an Arrhenius law and the different parameters were determined. A common activation volume was defined with $\Delta V = 0.275$ J/bar. As a result, we are able to calculate the electrical conductivity for larger temperature and pressure ranges for the melt compositions considered here. By combining the Nernst–Einstein and Eyring equations, a remarkably simple correlation was established between electrical conductivity and viscosity. The viscosity of carbonated melts, which is a key parameter defining the rate of metasomatic fluids flowing in the earth's mantle, can therefore be calculated as a function of pressure and temperature.

We used these new data to interpret the high electrical conductivity recently observed in the mantle under the Brazilian craton. The anomalously elevated conductivity most likely images the process of lithospheric rejuvenation involving 0.03 to 0.2% of carbonated melt.

Keywords

- Carbonated melt;
- Electrical conductivity;
- Viscosity;
- CO₂;
- Mantle geophysics;
- Piston cylinder

1. Introduction

Melting in the Earth's mantle due to CO₂ has long been proved by geochemical observations (Yaxley et al., 1991, Rudnick et al., 1993, Walter et al., 2008, Collerson et al., 2010 and Simonetti and Neal, 2010), and experimental petrology (Wyllie and Huang, 1975, Falloon and Green, 1990, Presnall and Gudfinnsson, 2005, Dasgupta and Hirschmann, 2006 and Ghosh et al., 2014) on peridotitic and eclogitic rocks (Hammouda, 2003 and Dasgupta et al., 2004). All these studies reported that carbon dioxide can strongly reduce mantle solidus temperature even at the very low concentration levels expected in the mantle. Carbonate melts are therefore expected in most of the P–T domains of the upper
mantle, being merely confined by redox boundaries marking the diamond stability field (Rohrbach and Schmidt, 2011 and Stagno et al., 2013).

Their physical properties are much less known (Jones et al., 2013), in particular their electrical conductivity. Existing data were obtained at low pressure (Gaillard et al., 2008 and Kojima et al., 2008) on alkali-rich carbonate melts or at high pressure (Sifré et al., 2014) and on alkali-free carbonate melts with poorly defined temperature and pressure dependence (Yoshino et al., 2010 and Yoshino et al., 2012). Gaillard et al. (2008) showed that carbonated melts are five orders of magnitude more conductive than mantle olivine, and up to three orders of magnitude more conductive than basalt at atmospheric pressure. This observation was corroborated by Sifré et al. (2014) at 3 GPa. However, in the absence of systematic data on the effect of chemical composition, pressure and temperature on the conductivity of carbonated melts, it is difficult to quantitatively interpret mantle electrical properties.

In this study, we report measurements of electrical conductivities of dry and hydrous carbonated melts at HT (> 1000 °C) and HP (1 to 4 GPa) in a piston cylinder. We describe the effect of chemical composition, temperature and pressure on the electrical conductivity. We propose a calculation to predict the effect of T and P by using the Arrhenius parameters determined in this study. These new data provide a link between electrical conductivity and viscosity of carbonatic melts, with a remarkably simple equation. Finally, laboratory-based values and geophysical measurements are confronted allowing quantitative estimations of melt fraction below the Brazilian craton.

2. Material and methods

2.1. Starting material

Electrical conductivity measurements were performed on six different mixtures (Table 1 and Table 2): 5 dry carbonated melts (CO₂ ranging from 38 to 49 wt.%) and a hydrous carbonated melt (CO₂ = 28 wt.%; H₂O = 8 wt.%), the latter corresponding to fluid inclusion compositions found in diamonds (Klein-Bendavid et al., 2009). To obtain these mixtures, starting materials were natural calcite (CaCO₃), natural dolomite (MgCa(CO₃)₂), natural magnesite (MgCO₃), potassium carbonate (K₂CO₃), sodium carbonate (Na₂CO₃), silicon dioxide (amorphous SiO₂), halite (NaCl) and brucite (Mg(OH)₂). The starting materials were stored in an electrical oven at 120 °C before use. Potassium and sodium carbonates were however previously dried at 250 °C to ensure that they were fully dehydrated. Samples C and MC were cored from natural rocks (Carrara Marble and natural dolomite, Table 1) whereas the other mixtures were weighed in suitable amounts for the desired compositions (MK, CK, MN and MCKNw samples) and then cold pressed into pellets. For samples MK, CK and MN, a one to one molar ratio was respected, that is MgCO₃/K₂CO₃ = 1 for MK, CaCO₃/K₂CO₃ = 1 for CK and MgCO₃/Na₂CO₃ = 1 for MN. Concerning composition MCKNw, a molar ratio of CO₂/(CO₂ + H₂O) = 0.65 was considered for the volatile content.

2.2. High-pressure conductivity measurements

Experiments were performed in ½-inch piston cylinders (graphite–Pyrex–talc assemblages) at pressures ranging from 1 to 4 GPa. These experiments were connected to a 1260 Solartron Impedance/Gain Phase Analyzer for the electrical conductivity measurements. Temperature
was measured with a B-type thermocouple localized on top of the sample (Figs. 1 and S1). The oxygen fugacity (fO$_2$) was not controlled during the measurements but the presence of the graphite furnace and molten carbonate samples should imply a fO$_2$ close to FMQ-2 (Stagno et al., 2013).

A protocol used for measurements has been specifically developed for electrical conductivity measurements on highly conductive molten materials (Fig. 1). A pseudo 4-wire configuration was designed, which removed the electrical contribution of the electrical cell itself. A blank test is presented in Fig. 2, the measured resistance consisting of the blank assemblage (i.e. the metal plug (Fig. 1), the Pt foil (25 μm in thickness and 15 mm length) and a Ni sample). Fig. 2 shows that the electrical conductivity of the electrical cell itself is two orders of magnitude lower than the carbonatized melt, confirming the use of a 4-wire configuration. Such a configuration previously adapted at 1 atm (Gaillard, 2004 and Gaillard et al., 2008) and 3 GPa (Sifré et al., 2014) was judged necessary for our measurements at high pressure.

An inner Pt electrode (1 mm diameter) was placed in the centre of the cold-pressed pellets or the centre of the natural cores (5 mm outer diameter). A Pt foil was positioned around the sample and used as the second outer electrode. The sample (approx. 2 mm length) was surrounded at the top and at the bottom by two MgO plugs (Fig. 1) that were previously annealed at 1000 °C/1 atm for 2 h. The length of each MgO plug was adjusted in order to position the sample at the centre of the furnace. The entire electrical cell was isolated from the graphite furnace by an alumina jacket (Fig. 1). The sample impedance was therefore measured between the two electrodes in a co-axial geometry (Gaillard, 2004 and Hashim et al., 2013). Connection between the inner electrode and the impedance spectrometer was achieved by means of the thermocouple wires (Hashim et al., 2013). The outer electrode was connected to a nickel cylinder (metal plug in Fig. 1), the latter located 5 mm above the sample, which was mounted in series with two additional wires (B-type thermocouples), as shown in Fig. 1.

Impedance spectra were collected during heating and cooling cycles at different temperature plateaux in the frequency range of 1 Hz to 1 MHz (data collection < 2 h, Fig. S2). In the case of the hydrated sample (MCKNw), the heating cycle was slightly different: the temperature was rapidly (less than 2 min) increased to the maximum temperature (Table 3) in order to trap the volatiles (released by the brucite compound) into the molten sample. Electrical conductivity measurements were therefore conducted during the cooling cycle and the second heating cycle (except for the run at 4 GPa where no measurement was made during the second heating cycle; Table 3).

For temperatures lower than the melting temperature, T$_m$, the collected spectra showed impedance arcs, whereas for temperatures > T$_m$, the produced spectra exhibited vertical lines. These low- and high-temperature spectra correspond to capacitance-dominated and inductance-dominated signals, where the intercept of each spectrum with the X-axis yielded the resistance of the studied sample.

Validation of each electrical value was achieved by reproducing the measurements during both heating and cooling cycles. Any measurements failing in satisfying this requirement were discarded.

The electrical conductivity of the different samples was calculated from the measured resistance, R (in Ω) (Gaillard, 2004 and Hashim et al., 2013), using:
\[ \sigma = \frac{(\ln(r_{\text{out}}/r_{\text{in}}))}{(2\pi h R)} \]

with \( \sigma \), the calculated electrical conductivity in \( \text{S} \cdot \text{m}^{-1} \); \( r_{\text{out}} \) and \( r_{\text{in}} \), the outer and inner radii of the sample (in m) and \( h \), the height of the sample (in m). Density measurements (Dobson et al., 1996 and Liu et al., 2011) show that for the temperature ranges considered in this study, melt should expand on average by 3%. Since the sample is placed in a co-axial geometry, thermal expansion is therefore considered negligible and the radii ratio should be kept constant during the experiments.

Uncertainties on \( \sigma \) were calculated considering the geometrical factors of the sample (i.e. \( r_{\text{out}}, r_{\text{in}} \) and \( h \); see Section 2.4.) and the propagated errors of each measured resistance. Uncertainties on \( \sigma \) are of about 10% on average for all measurements.

2.3. Validation of protocol

In order to validate the previously described protocol, the electrical conductivity measurements of samples C and MC obtained during the heating cycle were compared to established phase diagrams of natural calcite and dolomite. Fig. 3 shows an increase in electrical conductivity with increasing temperature and a slope shift for each sample. Sample MC presents a slope rupture between 1300 and 1315 °C at 3 GPa. The slope rupture for sample Ca is at higher temperatures, i.e. between 1416 and 1423 °C at 1 GPa, and between 1542 and 1558 °C at 3 GPa. These slope shifts indicate a change within the samples, from solid state for low temperatures to molten state at higher temperatures. The electrical conductivity is therefore directly impacted by this state change since the conduction mechanism switches from an electronic conduction mechanism in solids at low temperatures (Mirwald, 1979) to an ionic conduction mechanism in liquids (Gaillard et al., 2008 and Kojima et al., 2008). Previously, phase diagrams indicate that CaCO\(_3\) melt at such temperatures at 1 and 3 GPa (1417 °C and 1558 °C, respectively; Suito et al., 2001). The melting temperature of MgCa(CO\(_3\))\(_2\) at 2.7 GPa, approx. 1300 °C, determined in this study is also in agreement with previously published phase diagrams (Wyllie and Lee, 1998). Melting temperatures determined from the in situ electrical conductivity measurements on natural calcite and dolomite are in very good agreement with previously published phase diagrams.

Electrical conductivity measurements were also conducted on solid calcite at a pressure of 2.5 GPa (Bagdassarov and Slutskii, 2003) and are in good agreement with the measurements on sample C from this study for temperatures < 1558 °C at 3 GPa (Fig. S3). The electrical conductivity measurements performed here are therefore consistent with previous studies on CaCO\(_3\) and MgCa(CO\(_3\))\(_2\), thus validating our newly defined protocol.

2.4. Analytical techniques and imaging

After each experiment, the samples were mounted in epoxy and cut along the longitudinal axial section. A scanning electron microscope (SEM) was used on the polished mounts, that were previously carbon-coated, at back-scattered electron mode to image the sample geometry. A decrease of about 20% for parameters \( r_{\text{out}}, r_{\text{in}} \) and \( h \) compared to the initial geometry was observed on the SEM images, which is most likely due to porosity loss during melting (Fig. S1). The SEM images also showed that the samples remained sandwiched between the MgO plugs and electrodes. A 3D reconstruction of one of the samples by
microtomography showed that the sample remained cylindrical after the experiment and indicated the same diameter than on SEM images.

Chemical compositions and homogeneity of the melt phase were obtained using an electron microprobe (EMP). EMP analyses were conducted at 15 kV, 10 nA and 10 s counting on peak elements. In order to smooth heterogeneities due to quench crystallizations, the beam size (100 μm × 100 μm) was adapted to obtain average chemical compositions. Table 1 and Table 2 indicate that there was no contamination by the MgO plugs surrounding the sample (except for a minor increase for sample C, that has been exposed to very high T, whereas most of our measurements were conducted at T < 1500 °C) and no considerable volatile loss from the sample after the experiments. The carbonate (CO₂) concentration in the melt was calculated from the total weight deficit of the EMP analyses (Dasgupta et al., 2007) and indicates negligible decarbonation. Sample C was also analysed by X-ray diffraction before and after the experimental run (Fig. S4) confirming that no decarbonation happened.

For sample MCKNw, an elemental analyzer, type Flash 2000 (Thermo Scientific), was used to measure the H₂O content of the sample before and after the experiments. During these analyses, H₂O was released from the samples as reduced elemental H during the heating step (up to > 1500 °C), which was detected by a highly sensitive thermal conductivity detector. This method thus provides the water content of the samples with a precision of ± 0.5 wt.%. During the electrical conductivity measurements, dehydration was therefore considered negligible, as noticed by Laumonier et al. (in press) on dacite up to 12 wt.% of H₂O.

3. Results

3.1. Chemical and temperature effects

The electrical conductivity of the molten state samples at constant pressure (3 GPa) is given in Fig. 4 in a log $\sigma$ vs. $10^4/T$ plot (the complete data set from solid state to molten state is shown in Fig. S3). The collected data for each experiment shows good reproducibility during both the heating and cooling cycles. Every sample displays the same behaviour; basically the electrical conductivity is poorly dependent on temperature since it increases linearly by only a factor of 2 with increasing temperature within the T-range investigated here. As an example, the electrical conductivity of sample CK has a value of 60 S·m⁻¹ at 1090 °C and increases to a value of 120 S·m⁻¹ at 1433 °C (Fig. 4).

Electrical conductivity values for the samples studied at 3 GPa range between 54 S/m for sample CK at 1000 °C and 260 S/m for sample C at 1677 °C. These electrical values show that carbonated melts are five orders of magnitude more conductive than mantle olivine (Wang et al., 2006 and Yoshino et al., 2006), and up to two orders of magnitude more conductive than basalt (Tyburczy and Waff, 1983 and Ni et al., 2011) at mantle conditions. Fig. 4 shows that there is essentially a poor chemical composition effect on the electrical conductivity for the compositions studied here.

Since the electrical conductivity behaves linearly with the reciprocal temperature, each sample can be fitted using the following Arrhenius expression:

$$\sigma=\sigma_0 \cdot \exp(-E_a/RT)$$
where $\sigma_0$ is the pre-exponential factor (in S/m), $E_a$ is the activation energy (in J/mol), $R$ is the gas constant and $T$ is the temperature (in K). For a better fit of the above parameters, the high-temperature dataset of sample C (1570 °C < $T$ < 1677 °C, small circles in Fig. 4) was neglected because of a low temperature precision due to the B-type thermocouple. Fitted parameters are listed in Table 3. Fitted $E_a$ values are essentially similar for all compositions, ranging from 37 to 47 kJ/mol.

**3.2. Pressure effect**

Electrical conductivity measurements were also performed at different pressures in order to assess the effect of pressure on the conductivity. Three different compositions (C, CK and MCNKw) were studied for pressures ranging from 1 to 4 GPa and are shown in Fig. 5 (the complete data set is shown in Fig. S5). Essentially, the electrical conductivity decreases with increasing pressure. This pressure effect seems however lowered with increasing temperature as seen with sample C in Fig. 5.

The activation energy decreases with increasing pressure as shown in Fig. 6a. A common activation volume ($\Delta V$) was calculated by linear regression of the data in an $E_a$ vs. pressure plot (Fig. 6a), yielding a value of $0.275 \pm 0.05$ J/bar, with the following equation:

$$E_a = \Delta H + P \cdot \Delta V$$

with $\Delta H$, the activation enthalpy (J·mol$^{-1}$) and $P$, the pressure (in bar).

Fig. 6b shows that the limited experimental data from this study does not indicate any clear pressure effect on the pre-exponential factor, $\sigma_0$.

Notwithstanding, the linear fit based on Eqs. (2) and (3) reproduces remarkably well the electrical conductivity values collected in this study (Fig. S5). By using these equations, it is therefore possible to determine the electrical conductivity of molten carbonates at higher or lower pressures than in this study within reasonable error.

**4. Discussion**

**4.1. Chemistry and water effects**

The results of this study globally show a poor chemical effect on the electrical conductivity of molten carbonates. In detail, a correlation between electrical conductivity and the size/charge ratio of the different ions introduced into the melt was observed. Indeed, introduction of small cations, with small charges increases the bulk electrical conductivity of the melt, as shown by the addition of a sodium component in the melt in Fig. 4. This is consistent with the observation of Kojima et al. (2008) and Gaillard et al. (2008) based on 1 atm measurements, where they demonstrated that molten carbonates are increasingly conductive as alkali substitution follows the order Li > Na > K.

Replacement of alkalis by alkali-earth elements triggers a decrease in electrical conductivity, as observed in the measurements on Mg-bearing samples, and noticed in earlier studies (Gaillard et al., 2008 and Kojima et al., 2008) on Mg-free Ca-bearing samples.
Fig. 7 shows a comparison of the electrical conductivity of samples MC and MCKNw with a previous experimental study on molten carbonates (Sifré et al., 2014). For dry carbonated melts, the addition of 10 wt.% Na₂CO₃ or 10 wt.% NaCl to a dolomitic composition showed an increase of the electrical conductivity on average by 10%. Adding Na₂CO₃ or NaCl into the melt also decreases the activation energy of the melt (Table 3). Two different hydrated carbonated compositions with essentially the same CO₂/(H₂O + CO₂) molar ratio are also compared in Fig. 7. A weak increase of electrical conductivity is observed when CO₂ is replaced by H₂O with respect to a molar ratio of CO₂/(H₂O + CO₂) = 0.65. In silicate melts, water increases the mobility of Na⁺, being the main charge carrier, which results in an increase of its electrical conductivity (Gaillard, 2004, Pommier et al., 2008 and Ni et al., 2011). In ionic liquids, this water effect is negligible since there is no polymerised structure in the melt inhibiting the diffusion of all cations.

The different melt compositions from this study are compared to previous numerical and experimental studies (Yoshino et al., 2010, Yoshino et al., 2012 and Vuilleumier et al., 2014) on molten carbonated melts in Fig. 8. There is a good agreement between molecular dynamics (MD) simulation on calcitic melts (Vuilleumier et al., 2014) and the electrical conductivity measurement from this study on sample C at 3 GPa. In contrast, the activation energies from both studies at 3 GPa are different, which is also apparent at 1 GPa on the activation energy and the electrical conductivity values. However, it is worth noting that the conductivity values are similar at 1 GPa between the study by Vuilleumier et al. (2014) and this study. Yoshino et al. (2010) measured the electrical conductivity of a molten calcitic melt containing 10 wt.% SiO₂ at 1427 °C. This study yielded a conductivity value slightly lower than the extrapolated electrical conductivity determined here (89 S/m vs. 120 S/m, respectively), probably due to the presence of 10% silica in Yoshino et al (2010)'s experiments. Indeed, Sifré et al. (2014) showed that addition of silicate into a carbonated melt moderately decreases its electrical conductivity for such small amounts of silica added. Yoshino et al. (2012) measured the electrical conductivity of molten dolomite, which is about a factor 2 lower than the values measured on sample MC in this study. Although the melting temperatures in both studies are the same for an equivalent pressure, we are not able to explain the discrepancy in the electrical conductivity values. As a consequence, the studies of Yoshino et al., 2010 and Yoshino et al., 2012 do not show any chemical effect on the electrical conductivity of carbonate melts. The molten calcitic melt containing 10 wt.% SiO₂ and the molten dolomite incidentally show the same electrical conductivity at 1427 °C (89 S·m⁻¹). We also note that there is no indication in the electrical measurements of a transition from a molten state to a solid state when decreasing the temperature (Yoshino et al. (2012)). Nonetheless, the electrical conductivity values from the study of Yoshino et al. (2012) are of the same order of magnitude than the values collected in this study.

4.2. Pressure effect

Similarly to the chemical effect, the pressure moderately affects the electrical conductivity, as shown in Fig. 5 and Fig. 8 and by the activation volume, ΔV, determined in this study. This negative pressure effect on electrical conductivity is consistent with MD simulations performed by Vuilleumier et al. (2014). The activation volume deduced from the latter study is in excellent agreement with the ΔV determined here. However, unlike Vuilleumier et al. (2014), the current experimental dataset is insufficient to determine a pressure effect on ΔV.
Additionally, a previous study on MD simulations performed by Genge et al. (1995) showed that diffusion coefficient of ionic species in molten CaCO$_3$ moderately decreases with increasing pressure from 0.1 to 10 GPa. This is consistent with the decrease of $\sigma$ with increasing P, as observed in this study. In silicate melts, it has also been shown that pressure tends to decrease electrical conductivity as reported in Tyburczy and Waff (1983), Gaillard (2004), Pommier et al. (2008).

4.3. Electrical conductivity and viscosity relationship

Electrical conductivity and diffusion are two physical parameters that are related by the Nernst–Einstein equation, as follows:

$$D=(\sigma \cdot k_B \cdot T)/(q^2 \cdot N)$$

with $D$, the diffusion coefficient (in m$^2 \cdot $s$^{-1}$), $k_B$, the Boltzmann constant, $T$, the temperature (in K), $q$, the charge and $N$, its concentration. Furthermore, viscosity can be related to diffusion by the Eyring equation (Reid et al., 2001):

$$D=(k_B \cdot T)/(\eta \cdot \lambda)$$

where $\eta$ is the viscosity (in Pa·s), and $\lambda$ is the translation distance of the diffusing ion (Å).

With Eqs. (4) and (5), a relationship between electrical conductivity and viscosity can be deduced:

$$H(T)=(q^2 \cdot N)/\sigma(T) \cdot \lambda).$$

The combination of Eqs. (4) and (5) assumes that the same diffusion process controls electrical conductivity and viscosity. For natural silicate melts, it has been shown that Na$^+$ is the main charge carrier contributing to their electrical conductivity. The particular diffusive behaviour of alkalis in silicate melts (Jambon, 1982, Henderson et al., 1985, Watson, 1994 and Gaillard and Iacono Marziano, 2005) is uncorrelated to the relaxation time of the Si–O bounds (Pfeiffer, 1998 and Mungall, 2002), which, in turn controls the viscosity. It is therefore clear that Eq. (6) is not valid for natural silicate melts. In alkali-free silicate melts, Eq. (6) has been relatively successfully tested by Gruener et al. (2001) indicating that viscosity, Al–O bounds relaxation times and electrical conductivity can be related. We tested Eq. (6) for carbonate melts: we have therefore used published data on the viscosity of carbonated melts at different pressures and temperatures (Dobson et al., 1996, Kono et al., 2013 and Vuilleumier et al., 2014) and plotted them as a function of the electrical resistivity ($\rho$, i.e. inverse of electrical conductivity) in Fig. 9. The electrical resistivity for each composition was recalculated with the Arrhenius parameters, as determined in Table 3, at the pressure and temperature conditions at which viscosity values were obtained by Dobson et al. (1996), Kono et al. (2013), and Vuilleumier et al. (2014). Fig. 9 shows that the electrical conductivity values calculated with the parameters in Table 3 are in good agreement with the electrical conductivity deduced by MD simulation (Vuilleumier et al., 2014; open circles vs. half empty circles in Fig. 9). In the first approximation, there is a remarkably simple correlation for carbonated melts between viscosity and electrical conductivity, such as:

$$\eta=1/\sigma(\pm 10\%).$$
The diffusive process rate-limiting electrical conductivity and viscosity must be similar in molten carbonates. It is certainly related to the ionic nature of such liquids (like molten salts see Jones et al, 2013 and ref therein) implying the combined and coordinated motion of all charged ions (at the same rates). The use of the Nernst–Einstein (Eq. (4)) indicates an average diffusion rate of $10^{-9}$ to $10^{-8}$ m·s$^{-1}$ for all the melt compositions investigated in this study. Similar diffusion rates were also obtained on alkali-rich melts (Gaillard et al., 2008). A diffusion rate of ca. $10^{-9}$ m·s$^{-2}$ is also deduced by MD simulations for all ionic species in molten calcite (Genge et al., 1995).

A brief application of the electrical conductivity/viscosity correlation defined in Eq. (7) can be made on the peridotitic or eclogitic mantle. Most diamond inclusions, which are probably residual carbonated melts (e.g. sample MCKNw), indicate growth conditions at temperatures of 900–1300 °C and at a depth range of 150 to 200 km (Meyer and McCallum, 1986, Harris, 1993 and Stachel and Harris, 2008). By using Eq. (3), such hydrated carbonated melts should imply electrical conductivities ranging between 20 and 115 S·m$^{-1}$ and as a result, viscosities between 0.008 and 0.05 Pa·s.

4.4. Geophysical application

Recent electromagnetic surveys (Bologna et al., 2005 and Pinto et al., 2010) revealed highly conductive intrusions in the lithosphere beneath the Brazilian craton: −2 to −1.25 S·m$^{-1}$ at 120–140 km depth. Carbonatite melts have been suggested to rise in the resistive cratonic lithosphere generating rejuvenation and re-oxidation of the reduced mantle (Pinto et al., 2010).

By using the carbonatite melts hypothesis and well-known electrical mixing laws, such as the tube model (Grant and West, 1965 and Schmeling, 1986), we can estimate the amount of melts needed to infer such high conductivity values beneath the Brazilian craton. For the calculations, we consider a reference geotherm at 4 GPa from Pollack and Chapman (1977), the electrical conductivity of the solid framework assessed by Jones et al. (2012) as well as the electrical conductivity of sample MCKNw, which represent the closest melt composition to the mantle from this study (Klein-Bendavid et al., 2009). Small amounts of carbonated melts (0.03 to 0.2%) are therefore needed to explain the electrical anomalies beneath the Brazilian craton detected by geophysical surveys (Fig. 10a).

As suggested in Pinto et al. (2010), electromagnetic surveys are expected to image large scale geological processes such as lithospheric rejuvenation, metasomatic oxidation and their relationships with diamond destruction/formation (Fig. 10b; Klein-BenDavid et al., 2007, Creighton et al., 2009, Foley, 2008 and Shirey et al., 2013).

5. Conclusion

A new dataset of electrical conductivity measurements performed on molten carbonates at high temperature and high pressure reveals poor temperature, pressure and chemical effects. Since the electrical conductivity of carbonated melts follows an Arrhenius law, the Arrhenius parameters were determined and the electrical conductivity of these melts can therefore be extrapolated to higher and lower temperatures and pressures. This study also established a correlation between electrical conductivity and viscosity of carbonated melts, making it possible to predict the viscosity in the underlying mantle of these ionic liquids.
The electrical conductivity/viscosity relationship established here is however simple and requires further study with a wider range of temperatures and pressure as well as a more complex range of chemical composition of molten carbonates and carbonated basalts. These future studies will subsequently help us in establishing a more robust correlation between these two physical properties.

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Table 1.

Chemical composition normalized to 100% of the volatile-free starting dacite after melting at 1450 °C and 1 atm.

<table>
<thead>
<tr>
<th></th>
<th>SiO₂</th>
<th>TiO₂</th>
<th>Al₂O₃</th>
<th>FeO</th>
<th>MnO</th>
<th>MgO</th>
<th>CaO</th>
<th>Na₂O</th>
<th>K₂O</th>
</tr>
</thead>
<tbody>
<tr>
<td>Value</td>
<td>67.93</td>
<td>0.91</td>
<td>15.30</td>
<td>4.24</td>
<td>0.07</td>
<td>1.45</td>
<td>3.31</td>
<td>2.09</td>
<td>4.69</td>
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<td></td>
<td>1.33</td>
<td>0.21</td>
<td>0.65</td>
<td>0.60</td>
<td>0.08</td>
<td>0.20</td>
<td>0.33</td>
<td>0.10</td>
<td>0.21</td>
</tr>
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</table>

Fig. 1: Profile of water concentration in the 3 hydrated syntheses obtained by FTIR over 1.3 to 2.0 mm.
Fig. 2.: 3/4 assembly used for electrical conductivity measurements in piston cylinder.

Fig. 3.: Impedance spectra represented in the Nyquist plan ($Z'$, $Z''$) for the dacite. A) At low temperature and relatively large resistance, the frequency range investigated defines a semicircle corresponding to the electrical response of the sample ($R < Z'$) and a linear part ($R > Z'$) due to the effect of the interface between the sample and the electrodes. At higher temperatures (B & C), corresponding to lower resistance values, no impedance arcs were observed; the electrical response is characterized by the decrease and the increase of $Z''$ (B), and by an increase of $Z''$ crossing $Z'' = 0$. The arrow indicates the resistance $R$ of the sample in each type of spectra.
Table 2.
Experimental details.

<table>
<thead>
<tr>
<th># exp.</th>
<th>Starting material</th>
<th>P (GPa)</th>
<th>T°C range</th>
<th>Duration (h)</th>
<th>Water content (wt.%)</th>
<th>Ea (kJ/mol)</th>
<th>Log σ₀ (S·m⁻¹)</th>
</tr>
</thead>
<tbody>
<tr>
<td>UTU41B-0g</td>
<td>UTU41B-dry</td>
<td>0.30</td>
<td>400–1214</td>
<td>7</td>
<td>–</td>
<td>88.3</td>
<td>2.6</td>
</tr>
<tr>
<td>UTU41B-0e</td>
<td>UTU41B-dry</td>
<td>0.49</td>
<td>400–1211</td>
<td>24</td>
<td>–</td>
<td>88.2</td>
<td>2.5</td>
</tr>
<tr>
<td>UTU41B-0i</td>
<td>UTU41B-dry</td>
<td>2.78</td>
<td>400–1350</td>
<td>2</td>
<td>–</td>
<td>96.4</td>
<td>2.7</td>
</tr>
<tr>
<td>UTU41B-3d</td>
<td>UTU41B-3</td>
<td>0.15</td>
<td>500–1107</td>
<td>2.5</td>
<td>3.25</td>
<td>75.8</td>
<td>2.8</td>
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<tr>
<td>UTU41B-3c</td>
<td>UTU41B-3</td>
<td>0.52</td>
<td>400–1218</td>
<td>3</td>
<td>3.28</td>
<td>79.1</td>
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<td>2.78</td>
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<td>2.04</td>
<td>400–1211</td>
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<td>7.11</td>
<td>84.0</td>
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<tr>
<td>UTU41B-7d</td>
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<td>3.00</td>
<td>400–1204</td>
<td>2</td>
<td>7.03</td>
<td>92.6</td>
<td>3.8</td>
</tr>
<tr>
<td>UTU41B-12</td>
<td>UTU41B-12</td>
<td>3.00</td>
<td>400–1219</td>
<td>2</td>
<td>11.76</td>
<td>91.0</td>
<td>4.2</td>
</tr>
</tbody>
</table>
Fig. 4. : Water concentration in the run products according to their pressure, compared with the starting glasses. No significant water loss was noticed.

Fig. 5. : Arrhenius diagram of the electrical conductivity of UTU41B-0e and UTU41B-7c versus 1/Temperature (K) along heating and cooling paths. The black arrow indicates the temperature at which the conductivity is reproduced during the next temperature paths. The solid line is the fit of experimental data at temperature higher than 950 °C. See the text for details.
Fig. 6. Arrhenius diagram of the electrical conductivity of dry and hydrous dacitic melts versus 1000/Temperature in K (open symbols). The legend indicates the water contents of the dacite (numbers in color, refer to the online version) and the pressure of the experiments (black number). The fits of the models for dry and hydrous dacitic compositions at various pressures using Eqs. (1), (5) and (7) and the fitting values in Table 3 are represented by curves with the same color code.

Fig. 7. Activation energy (A) and pre-exponential factor (B) versus the pressure. Same legend in A and B.
Fig. 8. Na, H⁺ diffusivities determined from electrical conductivity measurements for the four dacitic melts with different water contents (number on the right of curves), and H₂O diffusivity reported by Zhang and Behrens (2000). The Na diffusivities are compared with values from the literature (Lowry et al., 1982 and Henderson et al., 1985).

Table 3.

Fitted parameters of the electrical model of dry and hydrous dacitic melts.

<table>
<thead>
<tr>
<th>Parameter</th>
<th>Value</th>
</tr>
</thead>
<tbody>
<tr>
<td>a</td>
<td>0.064</td>
</tr>
<tr>
<td>b</td>
<td>5.96</td>
</tr>
<tr>
<td>σ₀</td>
<td></td>
</tr>
<tr>
<td>c</td>
<td>1.06E−05</td>
</tr>
<tr>
<td>d</td>
<td>2.49E−05</td>
</tr>
<tr>
<td>P dep.</td>
<td></td>
</tr>
<tr>
<td>e</td>
<td>−6146</td>
</tr>
<tr>
<td>H</td>
<td></td>
</tr>
<tr>
<td>f</td>
<td>88,440</td>
</tr>
<tr>
<td>Ea</td>
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<tr>
<td>g</td>
<td>0.176</td>
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<tr>
<td>ΔV</td>
<td></td>
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<tr>
<td>h</td>
<td>0.388</td>
</tr>
</tbody>
</table>
Fig. 9: Log of conductivity versus water content of the dacite at 850 °C (A) and 1000 °C (B) compared with data from the literature. Rhyolite is from Gaillard (2004), Sigmelts is a model simulating the dacitic composition (Pommier & Le Trong, 2011) and metapelite is from Hashim et al. (2013).
Fig. 10: Electrical conductivities of Altiplano–Puna Magma Body (Bolivia, A), Usu (Japan, B), Taupo (New Zealand, C), Merapi (Indonesia, D) and St Helens (Washington state, E) areas from magnetotelluric survey (gray rectangles) and corresponding conductivities from the model. Conductivities are represented until water saturation (H$_2$O sat.) estimated from the conductive body depth. Solid lines are calculated conductivity from the model of this study, the thick and thin ones being respectively the melt and a mixture of a melt + crystal conductivities. See the text for details.

Fig. 11: Electrical conductivity of the cold mantle wedge percolated by a small fraction of dacite. MT data are not specific to any particular melt fraction and come from Booker et al. (2004) (Argentina), Soyer and Unsworth (2006) (Northern Cascades), Jodick et al. (2006) (Mexico), Patro and Egbert (2008) (Cascades), Brasse et al. (2009), Worzewski et al. (2011) (Costa Rica), Matsuno et al. (2010) (Mariana) and Rosell et al. (2011) (Alboran).