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SUPPLEMENTARY MATERIAL 1

TOWARDS THE RECONCILIATION OF VISCOSITY CHANGE AND CO₂-INDUCED
POLYMERIZATION IN SILICATE MELTS

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1. CO₂ content determination

We used the method described by Morizet et al. (2013) to determine the CO₂ content via
Raman spectroscopy. This Raman calibration for CO₂ content is a linear calibration function
such as wt.% CO₂ = 15.17 x CO₃/HF relating the CO₂ content to the ratio between the area of
the CO₃²⁻ peak and the high frequency envelop of the silicate glass vibrational signature

27 (CO₃/HF). The calibration was established from a database on CO₂-bearing silicate glasses
28 (65 data points) with CO₂ up to ~16 wt.% CO₂. We extended this Raman calibration to
29 additional CO₂ content data (89 data points) with higher CO₂ content (up to 23 wt.% CO₂)
30 determined using bulk analyzer (Figure S1). The entire dataset for this calibration is provided
31 in the Supplementary material 2 as a spreadsheet. The calibration function has been modified
32 according to this new database and the linear relationship is wt.% CO₂ = 13.5 x CO₃/HF;
33 where HF stands for the high-frequency of the symmetric stretch of the aluminosilicate
34 network. Using this method, the typical error on the CO₂ content is 10% in relative to the
35 value. We made an analysis of the typical error with the updated database. In Figure S1, we
36 added the error dispersion lines as reported by the fit. The standard deviation on the CO₃/HF
37 ratio is 0.125 and the r² parameter is 0.969. As a result, for a CO₃/HF = 1.25, the error is 10%
38 in relative to the value; it will be lower than 10% at higher CO₃/HF and higher than 10% at
39 lower CO₃/HF. Considering the dispersion of the data points in Figure S1, we definitely think
40 this cannot be the truth: we observe that the scattering of the data point at low CO₃/HF ratio is
41 small and on the contrary is more important at high CO₃/HF ratio. In order to be on the safe
42 side, we consider a 10% error in relative to the value for the whole range of CO₃/HF ratio
43 determination.

44 We show in Figure S2 a comparison of the CO₂ content determination using Raman
45 spectroscopy and EPMA shortfall. This latter approach has been commonly used to estimate
46 CO₂ content dissolved in silicate glasses (Dasgupta et al., 2007; Moussallam et al., 2015);
47 however, the accuracy in the determination of the CO₂ content is low (error above 20% in
48 relative to the value as reported by Dasgupta et al., 2007). The corresponding data are
49 provided in Table 1. In Figure S2, the wt.% CO₂ by EPMA is calculated using the difference
50 to 100% from the major element analyses and corrected by the H₂O content determined by
51 FTIR spectroscopy. We observe in Figure S2 that the data points except for RB8E-13 are

52 aligned along the 1:1 line suggesting that the Raman calibration method is accurate enough to
53 determine the CO₂ solubility in the present investigated glasses.

54 The typical deconvolution of the Raman spectra for the CO₂-bearing glasses is shown in
55 Figure S3. We report the deconvolution for several samples with CO₂ ranging from 6.2
56 (RB8E-6) to 18.0 wt.% (RB8E-11). On each spectrum, we added the derived CO₃/HF ratio
57 from the area of the peaks as well as the calculated CO₂ content using the calibration factor of
58 13.5 mentioned earlier. We should emphasize that such approach is only one possible
59 solution; however, there is a clear definition of the ν_1 CO₃²⁻ peak at 1075 cm⁻¹, hence any
60 variation in the deconvolution would not induce strong variation in the CO₃/HF ratio and
61 therefore in the derived CO₂ content. The entire set of simulation parameters is provided in
62 Supplementary material 2.

63

64 **2. Change in ¹⁷O species as a function of CO₂ content**

65 We present in Figure S4 the change in the O species abundances as a function of XCO₂ as
66 determined from Raman spectroscopy (see Table 1). The O species concentrations were
67 determined from the ¹⁷O NMR spectra simulations shown in Figure 1B. We clearly observe as
68 the XCO₂ increases that there is a gradual increase in the abundance of bridging oxygens
69 species (BO), whereas in the same time the concentration of non-bridging oxygens (NBO)
70 decreases. These opposite trends are consistent with the identified effect of CO₂ towards an
71 increase in the degree of polymerization of the silicate melt structure.

72 The concentration of the oxygen triclusters (O^{III}) does not seem to change as a function of
73 CO₂ content, however, due to the low abundance of the O^{III} identifying any fluctuation would
74 be unwise. The peak identified at ~+150 ppm in the ¹⁷O NMR spectra has been attributed to

75 the signature of ^{17}O nuclei in CO_3^{2-} dissolved within the glass structure. This peak appears to
76 increase with increasing CO_2 content with an increasing factor very close to 1. Similar
77 chemical shift is also mentioned in early work (Klemperer, 1978) which reports a ^{17}O
78 chemical shift ranging from +100 to +180 ppm for carbonate groups in organic compounds.

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80 **3. Change in NBO/T as a function of CO_2 content**

81 In the main text (see Figure 2), we report the deviation in the NBO/T ($\Delta\text{NBO/T}$) as a function
82 of CO_2 content calculated as the molar fraction (XCO_2). In Figure S5, we show the actual
83 change in the NBO/T measured by ^{17}O and ^{29}Si NMR spectroscopy as a function of the
84 XCO_2 . As expected, the observed change in the NBO/T from ^{17}O and ^{29}Si NMR corroborates
85 the drawn conclusions in the main text: 1) the measured NBO/T from ^{17}O and ^{29}Si NMR is
86 higher than the one obtained from stoichiometric calculation indicating an excess NBOs
87 concentration ($\text{NBO/T} > 2$ at $\text{XCO}_2 = 0$); 2) CO_2 induces a polymerization of the silicate melt
88 structure; 3) the induced polymerization is much more important than the predicted
89 polymerization from the theoretical dissolution mechanism (see Eq. 1). For this latter point,
90 we derived in Figure S5A and B a slope of -3.1 and -3.2 for ^{17}O and ^{29}Si NMR results,
91 respectively; in comparison to the slope of -2 following the dissolution mechanism of Eq. 1.

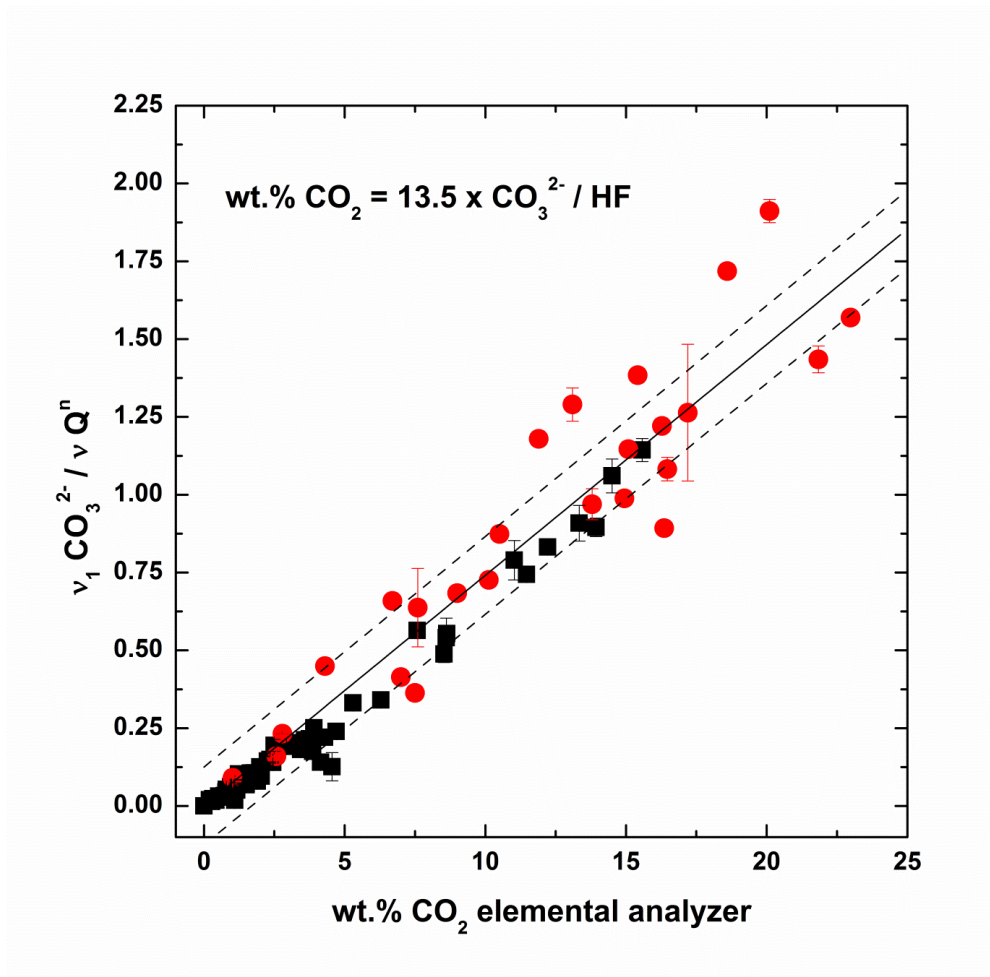
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93 **References Supplementary material 1:**

94 Dasgupta, R., Hirschmann, M.M., Smith, N.D., 2007. Partial melting experiments of
95 peridotite + CO_2 at 3 GPa and genesis of alkalic ocean island basalts. *J. Petrol.* 48, 2093–
96 2124.

97 Klemperer, W.G., 1978. ^{17}O -NMR spectroscopy as a structural probe. *Angew. Chem. Int.*
98 *Engl.* 17, 246–254.

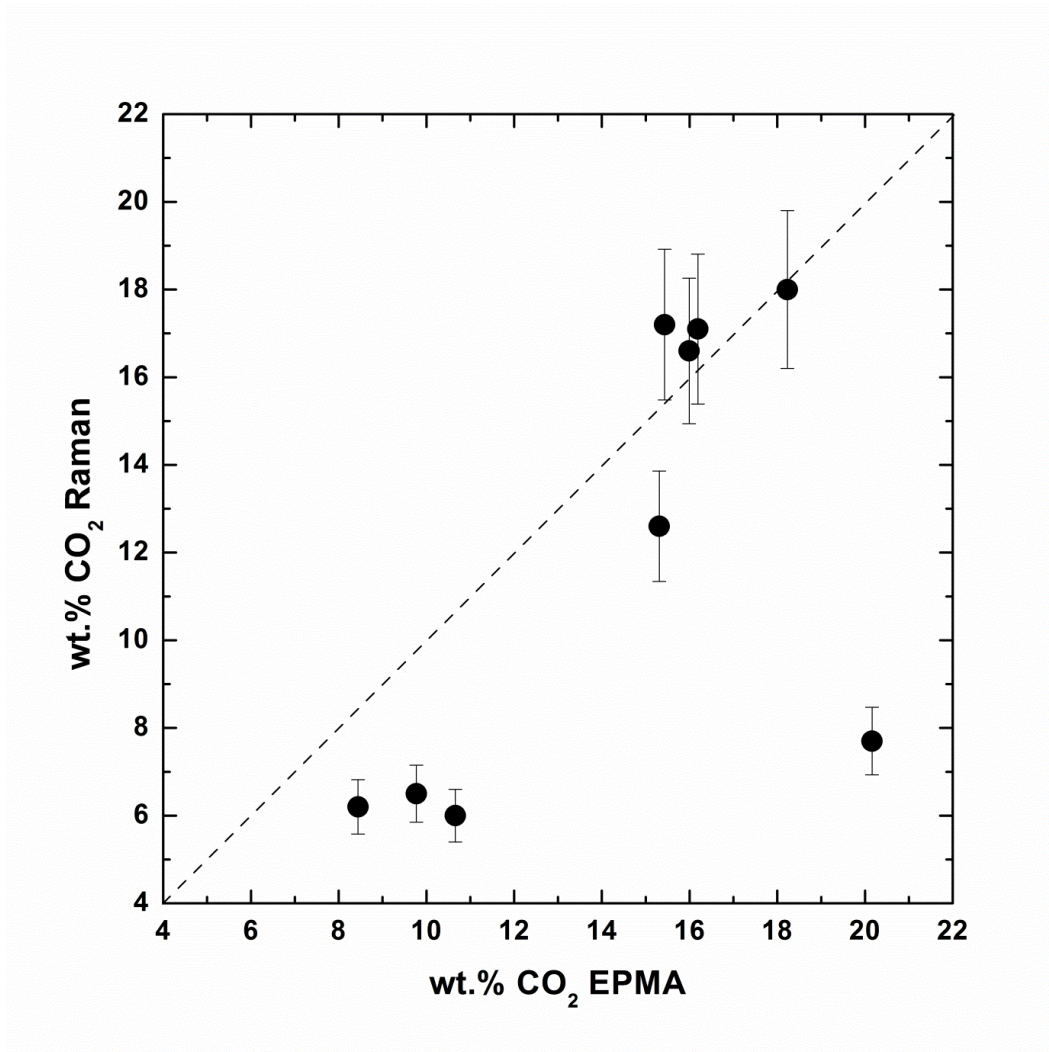
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102 Figure S1: Calibration for CO₂ content in silicate glasses using Raman spectroscopy. This
103 calibration represents an updated version of the one proposed by Morizet et al. (2013) based
104 on 65 data points and for CO₂ content up to 16 wt.%. The new calibration is based on 89 data
105 points and for CO₂ content up to 23 wt.% in silicate glasses (see Supplementary material 2 for
106 the entire dataset). The dashed lines represent the 95% confidence interval. The r² of the linear
107 regression is 0.969.

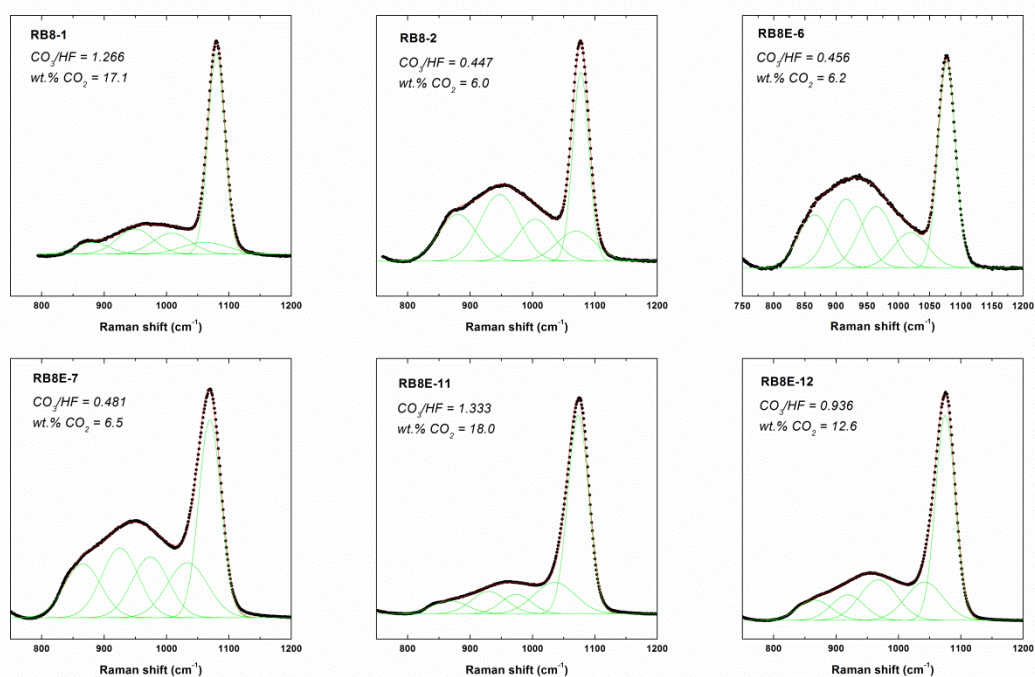
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110 Figure S2: wt.% CO₂ determined by Raman versus wt.% CO₂ estimated from the EPMA
 111 shortfall. The EPMA shortfall represents the difference to 100% in the major element
 112 concentration (see Table 1). The EPMA shortfall is corrected by the H₂O content determined
 113 by FTIR to extract the CO₂ content. No error is reported for the CO₂ content determined by
 114 EPMA as this method leads to high uncertainty (see the outlier at 20 wt.% CO₂ from EPMA
 115 and 8 wt.% CO₂ from Raman).

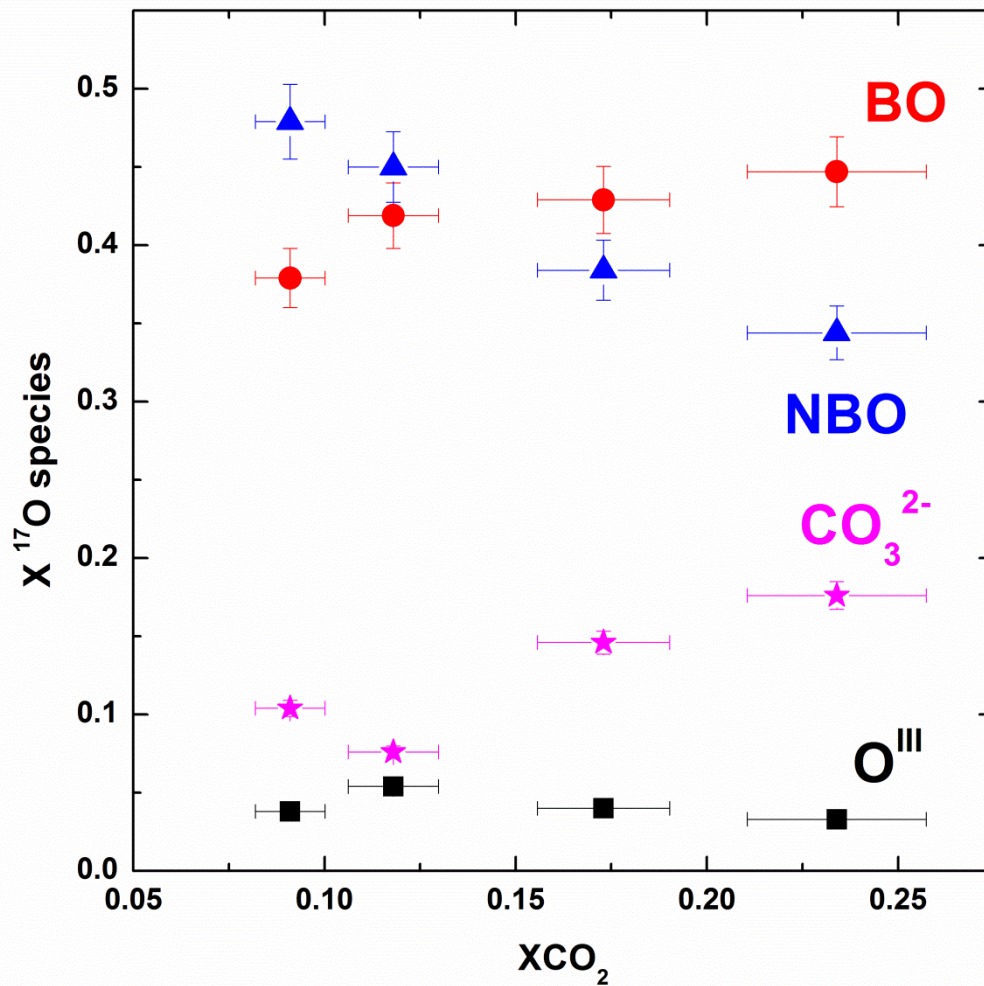
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118 Figure S3: Glass sample Raman spectra simulated with five Gaussian lines corresponding to
 119 the symmetric stretch of the CO_3^{2-} group ($\nu_1 \text{CO}_3^{2-}$) and silicate network ($\nu_1 \text{Q}^n$). The
 120 simulation is reported for RB8 and RB8E samples. The ratio CO_3/HF is reported next to each
 121 simulation and is used to calculate the CO_2 content from the linear correlation (see text for
 122 detailed discussion).

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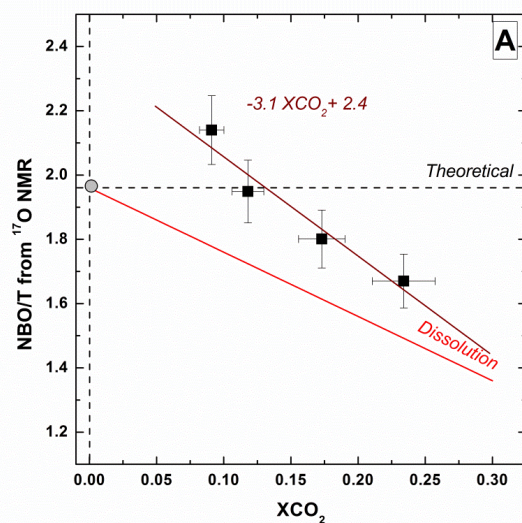


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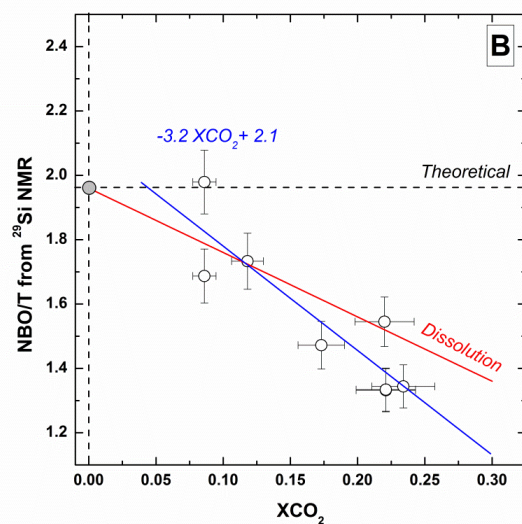
125 Figure S4: Distribution of ¹⁷O species as a function of XCO₂. The XCO₂ was determined from
 126 the Raman simulation results given in Table 1 and using the major element concentrations
 127 obtained from EPMA analyses. The molar fraction of the ¹⁷O species was obtained from the
 128 simulation of the ¹⁷O NMR spectra shown in Figure 1B of the manuscript main text.

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133 Figure S5: Change in the degree of polymerization (NBO/T) determined from ^{17}O (A) and
 134 ^{29}Si (B) NMR as a function of XCO_2 (see Table 2). Line corresponding to the theoretical
 135 degree of polymerization (1.96, see Table 1) is shown. Line corresponding to the change in
 136 the degree of polymerization following the Eq. 1 dissolution mechanism is reported. On each
 137 plot, we added the linear regression for ^{29}Si and ^{17}O NMR data points.

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