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1	Revision 1
2	¹⁷ O NMR evidence of Free Ionic clusters M ⁿ⁺ CO ₃ ²⁻ in silicate glasses: Precursors for
3	carbonate-silicate liquids immiscibility
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23	Abstract
23	110501400

- 25 melilitites. It is currently assumed that CO₂ molecules dissolving in low-silica melts as
- 26 carbonate groups (CO_3^{2-}) induce a strong polymerization of the silicate network; however, the
- 27 exact molecular configuration of this dissolution mechanism is still debated.
- 28 Using ¹⁷O MAS NMR spectroscopy, we have investigated the carbonate molecular
- environment in a series of synthesized low-silica (31-41 wt.% SiO₂), CO₂-bearing (from 2.9
- to 13.2 wt.% CO₂) silicate glasses analogous to melilitites and kimberlites. With the selective

31 ${}^{13}C{}-{}^{27}Al{}-$ and ${}^{29}Si{}-{}^{17}O$ J HMQC NMR method, we show that CO₂ dissolved in the

32 studied low-silica glasses is totally disconnected from the silicate network, forming Free Ionic

33 Clusters (FIC) M^{n+} ..CO₃²⁻ with M^{n+} a charge compensating cation.

- The M^{n+} .. CO_3^{2-} FIC are considered as precursors to immiscibility in between carbonate and silicate liquids. Observed in all studied compositions, we suggest that this immiscibility can be produced from moderately to strongly depolymerized silicate melt compositions.
- 37

INTRODUCTION



Early work on CO₂ dissolution in silicate melt (Mysen et al. 1976; Eggler 1978; Mysen and
Virgo 1980; Spera and Bergman 1980) suggested that CO₂ plays the role of polymerizing

47 agent of the silicate network structure according to the following reaction:

48
$$CO_2^{fluid} + 2Q^{n \, melt} \leftrightarrow CO_3^{2-melt} + 2Q^{n+1 \, melt}$$
 Eq. 1

In this reaction, a CO_2 molecule scavenges available oxygen from a Q^n (a SiO₄ tetrahedron 49 unit with n bridging oxygen) unit to form a CO_3^{2-} group charge compensated by a network-50 modifying cation (e.g. Ca^{2+}). Raman spectroscopic studies (Mysen et al. 2011; Mysen 2012; 51 Moussallam et al. 2016) suggested that Eq. 1 proceeds to the right upon CO_2 dissolution 52 inducing a change in the proportion of the different oxygen species (i.e. Non-Bridging 53 Oxygen NBO and Bridging Oxygen BO). However, the exact molecular configuration of the 54 CO_3^{2-} groups is not identified. Several dissolution mechanisms have been invoked for CO_2 55 molecules in silicate melts. It has been suggested that CO₂ can form network carbonates (T-56 $CO_3^{2-}T$ with T = Si or Al) and NBO- CO_3^{2-} where a CO_2 molecule connects to an available 57 NBO of a tetrahedron (Brooker et al. 2001; Guillot and Sator 2011; Morizet et al. 2014). An 58 59 additional CO₂-dissolution mechanism has been proposed on the basis of molecular dynamic 60 simulations (Xue and Kanzaki 2013; Vuilleumier et al. 2015) and also inferred from Raman 61 spectroscopic studies (Mysen et al. 2011; Mysen 2012; Moussallam et al. 2016). In this mechanism the CO_3^{2-} molecular groups are be present in the silicate network as Free Ionic 62 Clusters M^{n+} .. CO_3^{2-} where M^{n+} is a charge balancing cation (i.e. alkali or alkaline-earth). Each 63 64 type of CO₂-dissolution mechanism is a strong function of the glass chemical composition (Brooker et al. 2001). For a given chemical composition, the melt physical and chemical 65 properties will be strongly affected by the type of CO₂-dissolution mechanism. 66 In the present work, we show direct proof of the M^{n+} .. CO_3^{2-} FIC existence in low-silica 67

depolymerized melts. Based on ${}^{17}O$ MAS NMR spectroscopy, we demonstrate that CO_3^{2-}

69 molecular groups in low-silica melt are disconnected from the silicate network. Such a

70 dissolution mechanism represents the precursor of the immiscibility process between a silicate

- 71 melt and a carbonate melt.
- 72

EXPERIMENTAL METHOD

Several Fe-free glass compositions were investigated via ¹⁷O NMR spectroscopy. The studied 73 compositions are depolymerized with SiO₂ content ranging from 30 to 40 wt.% (Table 1) and 74 a Non-Bridging Oxygen concentration per Tetrahedron (NBO/T) ratio of 1.2 to 3.2 calculated 75 from the chemical composition obtained from Electron Probe Micro-Analyses (EPMA, see 76 Table 1 and Supplementary material). Studied compositions are analogous to natural 77 78 kimberlites (Kjarsgaard et al. 2009) and melilitites (Keller et al. 2006). 79 The starting material was weighed into Pt capsules, which were then sealed by arc-welding. 80 Experiments were performed in piston-cylinder apparatus using ³/₄ inch talc-Pyrex high-81 pressure assemblies between 0.5 and 1.5 GPa and at 1525°C. Details of the experimental 82 syntheses are provided in the Supplementary material. 83 ANALYTICAL METHOD Major elements and CO₂ concentrations: EPMA and Raman spectroscopy 84 We performed micro-Raman analyses on a Jobin-Yvon LabRam 300 spectrometer equipped 85 with an Innova 300-5W Argon ion laser from Coherent© operating at a wavelength of 514 nm 86 to determine the CO₂ content dissolved in silicate glasses using the protocol proposed by 87 Morizet et al. (2013). A detailed description of the Raman analyses and the CO₂ 88 determination from them is provided in Supplementary material. 89

90 NMR spectroscopy

91	All Solid State ¹⁷ O NMR were conducted on a Bruker 850 MHz Avance III Wide Bore
92	spectrometer operating at a frequency of 115.3 MHz and referenced against liquid H_2O at O
93	ppm. The ¹⁷ O spectra were acquired using Full Hahn-echo and { ²⁷ Al, ¹³ C, ²⁹ Si} J-HMQC
94	(Heteronuclear Multiple-Quantum Correlation) pulse sequences (Amoureux et al. 2007;
95	Keeler 2010). The details of the analytical conditions are given in the Supplementary material.
96	The J-HMQC experiment uses the so-called scalar coupling ("J") which arises from the
97	electrons involved in the bond between two nuclei and allows the creation of Heteronuclear
98	Multiple-Quantum Coherences. In an $\{X\}$ - ¹⁷ O J-HMQC experiment where oxygen is the
99	observed nuclei and X the indirect one, the resulting ¹⁷ O spectrum displays only the oxygen
100	environments that are chemically bound to X.

101

RESULTS AND DISCUSSION

102	The ¹⁷ O MAS and $\{X\}$ - ¹⁷ O J-HMQC NMR spectra are shown in Figure 1 for several samples:
103	XE-2 (Figure 1A), HK-1 (Figure 1B), HK-M (Figure 1C) and RB8E-12 (Figure 1D). Several
104	oxygen environments are witnessed in the 17 O MAS NMR spectra: ~+50 ppm assigned to 17 O
105	nuclei in BO configuration associated with Si or Al tetrahedra; \sim +100 ppm assigned to ¹⁷ O
106	nuclei in NBO configuration (Stebbins et al. 1999; Lee and Stebbins 2003; Kelsey et al. 2008)
107	and ~+160 ppm assigned to 17 O nuclei in CO ₃ ²⁻ environments (Morizet et al. 2017). The peak
108	at \sim +100 ppm is less apparent in Mg-bearing compositions (HK and XE) whereas it is
109	prominent in Mg-free Ca-rich RB8E composition consistent with the peak assignment to Ca^{2+}
110	NBO (Thompson and Stebbins 2011). The presence of Mg^{2+} surrounded by oxygen atoms
111	units is proposed at ~+50-70 ppm (Allwardt and Stebbins 2004) but not clearly visible here
112	due to strong overlap with the BO-(Si,Al) signal at ~+50 ppm.
113	The selective heteronuclear $\{X\}$ - ¹⁷ O J-HMQC NMR pulse sequence probes local

environments of oxygen atoms bound to X atoms. Hence, it should be pointed out that 114

chemical bonds involving nuclei other than the considered one (X) will not contribute to the 115 ¹⁷O NMR signal in the $\{X\}$ -¹⁷O J-HMOC spectrum. Serendipitously, the $\{^{13}C\}$ -¹⁷O J-HMOC 116 spectra show a single line located at ~+160 ppm confirming that this peak is associated with 117 CO_3^{2-} molecular groups. More importantly, the absence of signal at ~+50 or +100 ppm 118 indicates that the CO_3^{2-} is connected neither to an available NBO nor to a BO. This is also 119 confirmed by the $\{^{29}Si\}$ -¹⁷O J-HMQC spectra, which do not exhibit a signal at +160 ppm but 120 only at +50 and +100 ppm; this is consistent with ¹⁷O atoms in BO and NBO configuration, 121 respectively. The absence of common ¹⁷O signals in the ${}^{29}Si$ J-¹⁷O J-HMQC and ${}^{13}C$ J-¹⁷O J-122 HMQC spectra shows that there is no oxygen atom simultaneously bound to silicon and 123 carbon atoms (within the detection level of $\sim 5\%$) thereby precluding the existence of T-CO₃²⁻-124 T and NBO- $CO_3^{2^2}$ carbonate units. This CO_2 -dissolution mechanism has been proposed on the 125 basis of FTIR results (Brooker et al. 2001), inferred from Molecular Dynamic simulations 126 (Vuilleumier et al. 2015) and indirectly suggested from the change in the network structure 127 upon CO₂-dissolution (Mysen et al. 2011; Mysen 2012; Moussallam et al. 2016). Our results 128 are consistent with the MD results from Vuilleumier et al. (2015) suggesting that FIC 129 represents 75% of the dissolved CO_3^{2-} molecular groups and 25% corresponding to network 130 carbonate units (T-CO₃²⁻-T with T = Si or Al; Brooker et al. 2001). In the FIC M^{n+} ..CO₃²⁻, 131 M^{n+} is a charge balancing cation which can be Mg^{2+} , Ca^{2+} and Na^+ to a lower extent (Na₂O is 132 ~5 wt.%) for the currently investigated compositions. Although, Morizet et al. (2017) showed 133 that CO_3^{2-} groups dissolve preferentially in the vicinity of Ca^{2+} cations, we do not exclude the 134 possibility for Mg^{2+} .. CO_3^{2-} being formed considering that Mg^{2+} could play a role comparable 135 to Ca^{2+} (Cormier and Cuello 2013). 136

137 At this point, we wish to raise the level of confidence in the above discussion. Indeed, one

138 could argue that the ¹⁷O signal from Si-O-C linkages could be absent from the ${^{29}Si}$ -¹⁷O J-

HMQC spectrum if the associated $J(^{29}Si^{-17}O)$ coupling value strongly differs from those

140	occurring for ²⁹ Si- ¹⁷ O(BO) or ²⁹ Si- ¹⁷ O(NBO) linkages. A simple way to rule out this
141	possibility is to show that the full ¹⁷ O spectrum can be totally reconstructed as the sum of the
142	3 selective {X}- 17 O J-HMQC spectra. The reconstructions of the 17 O MAS NMR spectra from
143	the deconvolution of the $\{X\}$ - ¹⁷ O J-HMQC spectra are shown in Figure 2 for three ¹⁷ O MAS
144	NMR spectra: Mg-free RB8E-12 (Figure 2A) and Mg-bearing XE-2 and HK-1 (Figure 2B and
145	C). The deconvolution procedure was conducted as follows: each $\{X\}^{-17}$ O J-HMQC NMR
146	spectrum was fitted with Gaussian lines; one line for the ${}^{13}C$ - ${}^{17}O$ J-HMQC, one line for the
147	${}^{27}Al$ - ¹⁷ O J-HMQC and two lines for the ${}^{29}Si$ - ¹⁷ O J-HMQC corresponding to the BO-Si
148	and NBO-Si environments, respectively. The parameters obtained are reported in Table 1.
149	Since the { ²⁷ Al}- ¹⁷ O J-HMQC spectrum was not acquired for all the samples due to the
150	requirement of very long acquisition times, we used the δ_{iso} and FWHM from the XE-2
151	deconvolution for HK spectra; and the average δ_{iso} and FWHM obtained from RB8E-12 and
152	RB8E-13 for RB8E-7 spectrum. In Figure 2A, we show that the performed deconvolution
153	adequately reproduces the ¹⁷ O MAS NMR spectrum for the Mg-free RB8E-12 sample. The
154	high carbon content of this sample excludes the possibility of a lack of ¹⁷ O signal of
155	significant intensity in the ${}^{29}Si$ }- ${}^{17}O$ J-HMQC spectrum. For Mg-bearing compositions, the
156	¹⁷ O MAS NMR spectra are not perfectly reproduced by the deconvolution. Regardless of the
157	glass composition (XE-2 and HK), the residuals of the simulation exhibit a signal located at
158	~+70 ppm not accounted for. Considering that this residual NMR signal is only observed in
159	Mg-bearing glass composition, the signal could be attributed to oxygen atoms in MgO _x
160	configuration. This assignment would be consistent with the results of Hung et al. (2016)
161	which identified ¹⁷ O resonance for MgO_x oxygens in Ca/Mg orthosilicate glasses at +60 ppm.
162	Considering that the NMR signal of the CO_3^{2-} environments is reproduced adequately by a
163	single line at ~+160 ppm, the majority of the CO_3^{2-} groups are FIC in the investigated glass
164	compositions.

165	$Implication \ for \ Carbonate-Silicate \ liquid \ immiscibility$
166	Numerous implications arise from the present results; however we only focus on two of them:
167	1) the immiscibility between carbonate and silicate liquids and 2) the impact of FIC on the
168	carbonated melt transport properties. It is reasonable to consider that the $M^{n+}CO_3^{2-}$ FIC
169	constitutes a carbonate subnetwork intimately mixed within a silicate subnetwork. For
170	instance, Kubicki and Stolper (1995) showed that the molecular energy involved for ion pairs
171	complexes (similar to FIC M^{n+} CO ₃ ²⁻) is low as compared to any other type of carbonates
172	dissolved in silicate melt. Therefore, the energy barrier to dissociate the carbonate subnetwork
173	from the silicate subnetwork is lower in the case of FIC.
174	Network-modifying cations have varying coordination numbers in a silicate melt (e.g. 5 to 8
175	for Ca^{2+} ; Cormier and Cuello 2013). Upon CO_2 dissolution, the network-modifying cation
176	will share less of its positive charges to the silicate network in order to compensate the
177	negative charges of the CO_3^{2-} groups; resulting in a network-modifying cation that is less
178	connected to the silicate network. In particular, Vuilleumier et al. (2014) showed that the Ca-
179	O distance in molten CaCO ₃ is on average 2.35 Å, suggesting that Ca^{2+} is more closely
180	connected to the carbonate subnetwork than to the silicate subnetwork. Consequently, we
181	believe that the presence of FIC M^{n+} CO ₃ ²⁻ could be viewed as a possible precursor to
182	immiscibility. For instance, removal of the carbonate network from the silicate network is a
183	proposed mechanism to form calciocarbonatite from a melilitite parental melt (Guzmics et al.
184	2015).
185	We suggest that this immiscibility could also proceed from kimberlite owing to the fact that

186 FIC groups are observed in the investigated HK composition analogous to kimberlite melt.

- 187 However, the molecular configuration $M^{n+}..CO_3^{2-}$ FIC poses the problem of the way CO_2
- degasses from depolymerized silicate melts. The CO₂ migration into the fluid phase imposes a

189	molecular structure reconfiguration to dissociate the $M^{n+}CO_3^{2-}$ groups and causes the
190	network-modifying cation to remain in the silicate melt, thus depolymerizing the silicate
191	network structure. This difficulty could explain the absence of degassing in kimberlitic melt
192	upon decompression as witnessed by Moussallam et al. (2015).
193	Finally, it turns out that the identified CO ₂ -dissolution mechanism can also affect the physical
194	properties of carbonated melts (e.g. magma transport and electrical conductivity).
195	Interconnection between FIC molecular groups could explain the high electrical conductivity
196	observed in CO ₂ -bearing silicate melt (Sifré et al. 2014). There is an intimate mixture between
197	a carbonate subnetwork having an extremely low viscosity and a silicate subnetwork with a
198	higher viscosity (Morizet et al. in press). Through immiscibility, the carbonate liquid will
199	migrate towards the surface due to its low viscosity and density (Vuilleumier et al. 2014),
200	whereas the silicate liquid, which becomes enriched in SiO ₂ , is stalled at depth due to the
201	increased viscosity (Giordano et al. 2008). Therefore, the direct proof of the existence of FIC
202	gives legitimacy to the development and application of the immiscibility theory in a wide
203	range of processes taking place in the Earth's interior.

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210

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318 Figure caption:

319	Figure 1. ¹⁷ O MAS and ${^{13}C} - {^{27}Al} - {^{29}Si} - {^{17}O} J-HMQC NMR spectra of CO2-$
320	bearing silicate glasses: XE-2 (A), HK-1 (B), HK-M (C) and RB8E-12 (D). Three oxygen
321	environments are identified with MAS spectra: FIC M ⁿ⁺ CO ₃ ²⁻ , Ca ²⁺ NBO-Si and BO-(Si,Al)
322	The NMR spectra for the other synthesized glasses can be found in Supplementary material.

Figure 2. ¹⁷O MAS spectrum deconvolution obtained from Gaussian lines derived from $\{X\}$ -¹⁷O J-HMQC spectra. The deconvolution of the ¹⁷O NMR spectra for the other synthesized glasses can be found in Supplementary material.

Revision 1

Table 1: Experimental conditions, chemical compositions of the investigated synthetic Fe-free silicate glasses.

Sample	HK-1	HK-2	HK-M	RB8E-7	RB8E-12	RB8E-13	XE-2	Natural Kimberlite [§]	Natural Melilitite [§]
Pressure (GPa)	1.5	0.5	1.5	0.5	1.0	1.0	1.5		
Glass chemical composition in wt.%									
SiO ₂	40.2	40.1	31.8	33.3	30.3	30.1	40.6	30.4	36.5
Al_2O_3	4.3	4.4	3.5	8.7	8.1	8.6	14.8	2.3	7.7
MgO	21.9	21.3	21.9	0.0	0.1	0.2	12.6	29.7	11.8
CaO	22.0	22.3	22.0	41.8	39.8	36.0	17.8	9.6	14.2
Na ₂ O	4.5	4.5	3.5	5.7	4.8	5.0	4.8	0.1	4.3
$\mathrm{CO_2}^*$	7.1	2.9	13.2	6.5	12.6	7.7	4.2	6.40	0.52
H_2O^{\dagger}	0.8	1.0	1.3	0.7	1.5	1.3	1.0	7.10	1.06
NBO/T from stoichiometry [‡]	2.6	2.5	3.2	2.1	2.2	1.9	1.2	3.60	1.66
				NM	R parameters				
				Free	$e Ca^{2+}CO_3^{2-}$				
δ_{iso}	154.2	157.3	155.7	164.9	164.6	166.0	157.8		
FWHM	42.0	41.3	44.2	34.7	30.7	33.6	45.3		
					BO-Si				
δ_{iso}	54.4	57.4	55.2	61.0	51.6	53.4	48.7		
FWHM	24.4	29.3	24.3	23.3	24.3	22.3	25.5		
BO-Al									
δ_{iso}	42.5	42.5	42.5	47.8	45.5	47.5	42.5		
FWHM	25.8	25.8	25.8	25.6	25.7	25.5	25.8		
NBO-Si									
δ_{iso}	95.9	96.7	96.0	106.5	106.8	106.2	99.3		
FWHM	31.4	33.1	23.7	24.6	25.2	24.0	47.1		

^{*} The glass CO₂ content has been determined using Raman spectroscopy and the method described in Morizet et al. (2013). With the derived CO₃/HF ratio obtained from the deconvolution of the Raman spectra (see Supplementary material), it is possible to determine the CO₂ content using the linear relationship wt.% CO₂ = 13.5 x CO₃/HF. The typical error on the CO₂ content using Raman method is 10% in relative to the value.

[†] The H₂O content was determined by FTIR spectroscopy on a doubly polished glass plate. Contribution for OH⁻ and H₂O^{mol} vibrational peaks at 4500 and 5200 cm⁻¹ were summed to obtain the H₂O^{tot}. The error does not exceed ± 0.2 wt.% H₂O and corresponds to the standard deviation obtained on the replicated measurements.

[‡] The NBO/T describing the average polymerization was calculated on a stoichiometric basis using the chemical composition measured by EPMA.

[§] An average typical natural composition of kimberlite and melilitite were obtained from Kjarsgaard et al. (2009) and Keller et al. (2006), respectively. Additional oxides (TiO₂, FeO and K₂O for the most important) have not been reported but are nominally present:~0-4 wt.% for TiO₂, ~10.0 wt.% FeO and ~2.0 for K₂O.

Figure 1



Figure 2

