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## Transformation of Ni-containing birnessite to tectomanganate: Influence and fate of weakly bound Ni(II) species

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1                   **Transformation of Ni-containing birnessite to**  
2                                   **tectomanganate:**  
3                   **Influence and fate of weakly bound Ni(II) species**

4  
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20

21 **Abstract**

22 The geochemical behavior of nickel, an essential trace metal element, strongly  
23 depends on its interactions with Mn oxides. Interactions between the phylломanganate  
24 birnessite and sorbed or structurally incorporated Ni have been extensively  
25 documented together with the fate of Ni along the transformation of these layered  
26 species to tunnel Mn oxides (tectomanganates). By contrast, interactions of  
27 phylломanganates with weakly bound Ni species (hydrated Ni, Ni (hydr)oxides), that  
28 possibly prevail in natural Ni-rich (>10% NiO) manganates received little attention  
29 and the influence of these Ni species on the phylломanganate-to-tectomanganate  
30 transformation remains essentially unknown. Within this framework, a set of  
31 phylломanganate precursors with contrasting contents of Ni were prepared and  
32 subjected to a reflux process mimicking the natural  
33 phylломanganate-to-tectomanganate conversion. Layered precursors and reflux  
34 products were characterized with a combination of diffractometric, spectroscopic,  
35 thermal, and chemical methods. Ni is essentially present as hydrated Ni(II) and Ni(II)  
36 (hydr)oxides in layered precursors with no detectable Ni sorbed at layer vacancy sites  
37 or structurally incorporated. Despite the high content (~1/3) of Jahn-Teller distorted  
38 Mn(III) octahedra in these layered precursors, which is known to be favorable to their  
39 conversion to tectomanganates, polymerization of Ni(OH)<sub>2</sub> in phylломanganate  
40 interlayers is kinetically favored during reflux process. Asbolane, a phylломanganate  
41 with an incomplete – island-like – octahedral layer of metal (hydr)oxides, is thus  
42 formed rather than todorokite, a common tectomanganate with a uniform 3×3 tunnel  
43 structure. A nitric acid treatment, aiming at the dissolution of the island-like interlayer  
44 Ni(OH)<sub>2</sub> layer, allows an easy and unambiguous differentiation between asbolane and  
45 todorokite, which is unaffected by the treatment. Both compounds exhibit indeed  
46 similar periodicities and can be confused when using X-ray diffraction, despite  
47 contrasting intensity ratios. Ni(OH)<sub>2</sub> polymerization hampers the formation of  
48 tectomanganates and likely contributes to the prevalence of phylломanganates over

49 tectomanganates in natural environments. Most Ni is retained during the reflux  
50 process, part of Ni (~20%) being likely structurally incorporated in the reaction  
51 products, thus enhancing the sequestration of Ni in Mn oxides.

52 *Keywords:* nickel; busserite; asbolane; todorokite; Mn oxides

53

## 1. INTRODUCTION

54

55

56 Phylломanganates and tectomanganates are the two mineral families accounting  
57 for most of Mn oxides found in terrestrial and aquatic environments (Post, 1999;  
58 Peacock and Sherman, 2007a). In the former, edge-sharing MnO<sub>6</sub> octahedra form  
59 layers whose symmetry depends on the abundance and distribution of Mn(III)  
60 octahedra, whereas in the latter, chains of edge-sharing Mn(IV)O<sub>6</sub> octahedra form the  
61 walls or floor/ceiling of tunnels. Birnessite is, together with its disordered structural  
62 analogue vernadite, the most common phylломanganate and consists of a MnO<sub>2</sub> layer,  
63 whose layer charge deficit is compensated for by the presence of cations sorbed at  
64 layer vacancy sites or present as interlayer hydrated species (7.0-7.2 Å layer-to-layer  
65 distance). Heterovalent Mn cations and/or foreign transition metal cations can be  
66 hosted in both families, either structurally substituted for Mn(IV) or as charge  
67 compensating cations located in phylломanganate interlayers or in tectomanganate  
68 tunnels (Post, 1999). The occurrence of natural phylломanganates as micro- to  
69 nano-sized cryptocrystalline minerals usually enhances their surface reactivity that  
70 originates from their high negative layer charge and from the coexistence of  
71 heterovalent Mn cations in these structures. As the result, natural phylломanganates  
72 are highly reactive with respect to both organic (Remucal and Ginder-Vogel, 2014)  
73 and inorganic compounds. Specifically, natural phylломanganates are readily  
74 associated with transition metals such as Co, Cu, Zn, or Ni (Chukhrov, 1982;  
75 Manceau et al., 1987, 1992a,b, 2002, 2007a,b, 2014; Marcus et al., 2004; Bodeř et al.,  
76 2007; Peacock and Sherman, 2007a; Lanson et al., 2008; Spinks et al., 2017;  
77 Zawadzki et al., 2018), rare earth (Kasten et al., 1998; Ohta and Kawabe, 2001;  
78 Ohnuki et al., 2015; Zawadzki et al., 2018), and radioactive elements (Duff et al.,  
79 1999; Post, 1999).

80 In an effort to disentangle the mobility, (bio)availability, and fate of these  
81 elements in different geological settings, structural interactions of phylломanganates  
82 with such foreign elements, and more especially with transition metals, have thus

83 attracted the attention of the scientific community over the last few decades (Manceau  
84 et al., 1997, 2002, 2014; Ohta and Kawabe, 2001; Lanson et al., 2002a; Marcus et al.,  
85 2004; Toner et al., 2006; Peacock and Sherman, 2007a,b; Peacock, 2009; Peña et al.,  
86 2010; Roque-Rosell et al., 2010; Sherman and Peacock, 2010; Yin et al., 2011a,b;  
87 Kwon et al., 2013; Yu et al., 2013; Burlet and Vanbrabant, 2015; Simanova et al.,  
88 2015). In most studies, foreign elements were structurally substituted for Mn(IV) in  
89 the MnO<sub>6</sub> octahedra or sorbed as inner-sphere complexes either at vacant layer sites  
90 and/or at particle edges (Manceau et al., 1997; Yin et al., 2011a, 2015; Yu et al., 2013;  
91 Ohnuki et al., 2015; Simanova and Pena, 2015; Qin et al., 2018). Similar to other  
92 transition metals, presence of Ni(II) forming inner-sphere complexes species at  
93 vacancy or edge sites and/or structurally incorporated in phyllosmanganate layers has  
94 been extensively documented (Manceau et al., 2002, 2007a,b; Peacock and Sherman,  
95 2007a,b; Grangeon et al., 2008; Peña et al., 2010; Yin et al., 2012, 2014; Kwon et al.,  
96 2013; Simanova et al., 2015). Under low pH conditions, Ni(II) occurs mainly as  
97 triple- or double-corner sharing (TCS and DCS, respectively) inner-sphere complex  
98 above/below vacancy or at edge sites of the phyllosmanganate layers (Peacock and  
99 Sherman, 2007b; Grangeon et al., 2008; Peacock, 2009; Peña et al., 2010; Simanova  
100 et al., 2015). Migration from TCS/DCS sites to octahedral layer sites is favored by  
101 increasing contact time and increasing pH (Peacock, 2009), this structural  
102 incorporation within the octahedral MnO<sub>2</sub> layer being reversible (Peacock, 2009; Peña  
103 et al., 2010) and essentially restricted to low Ni contents (Manceau et al., 2002; Bodeř  
104 et al., 2007; Peacock and Sherman, 2007a,b; Kwon et al., 2013; Simanova et al., 2015;  
105 Atkins et al., 2016), consistent with natural occurrences (Manceau et al., 2002,  
106 2007a,b; Bodeř et al., 2007; Peacock and Sherman, 2007a).

107 In addition, hydrated Ni(II) or Ni forming outer-sphere complexes (Turner and  
108 Buseck, 1981; Peacock and Sherman, 2007a) and Ni(II) (hydr)oxides (Chukhrov et al.,  
109 1987; Manceau et al., 1987, 1992a,b) may also be associated with Mn oxides. Despite  
110 their widespread natural occurrences for high Ni contents (NiO ~10-20 wt.%, – Llorca,  
111 1988; Roque-Rosell et al., 2010; Ploquin et al., 2018), these weaker associations have  
112 received little attention compared to structurally incorporated Ni or to Ni inner-sphere

113 complexes. The presence of hydrated Ni(II) species in phylломanganate interlayers  
114 leads to the presence of two planes of interlayer water molecules, thus increasing the  
115 layer-to-layer distance from  $\sim 7.2$  Å (birnessite or vernadite) to  $\sim 9.6$ - $10.2$  Å (buserite –  
116 Burns et al., 1983; Chukhrov et al., 1987; Golden et al., 1987; Manceau et al., 1997,  
117 2014; Bodeř et al., 2007; Wegorzewski et al., 2015). Such increased hydration of  
118 Ni-rich phylломanganates has for example been reported by Manceau et al. (2007b).  
119 The presence of hydrated Ni(II) in phylломanganate interlayers has been described  
120 also as favorable to the transformation of phylломanganate to tectomanganates upon  
121 hydrothermal treatment (Ching et al., 1999; Luo et al., 1999; Onda et al., 2007),  
122 similar to Mg(II) (Golden et al., 1986, 1987). In addition, Ni(II) (hydr)oxides may  
123 polymerize in phylломanganate interlayers either as a complete or as an incomplete  
124 brucite-like octahedral layer, leading to litiophorite- or asbolane-like species,  
125 respectively, both having a layer-to-layer distance of  $9.4$ - $9.8$  Å (Chukhrov et al., 1987;  
126 Manceau et al., 1987, 1992a,b; Feng et al., 2001).

127 The present article thus focuses on the association of these weakly bound Ni(II)  
128 species with phylломanganates. Synthetic samples with contrasting Ni contents were  
129 prepared and characterized by X-ray diffraction, Fourier transform infrared  
130 spectroscopy, thermogravimetric analysis, X-ray photoelectron spectroscopy,  
131 extended X-ray absorption fine structure spectroscopy (Mn and Ni K-edges). In  
132 addition, phylломanganate samples were hydrothermally treated and reaction products  
133 characterized to investigate the effect of Ni on the  
134 phylломanganate-to-tectomanganate transformation (Golden et al., 1986; Shen et al.,  
135 1993; Feng et al., 2004, 2010; Atkins et al., 2014; Grangeon et al., 2014, 2015; Zhao  
136 et al., 2015) and the fate of weakly associated Ni(II) species during this  
137 transformation. The negative impact of Ni, when present at high content, on this  
138 transformation is contrasted with that of Co. Buserite-, litiophorite-, and asbolane-like  
139 phylломanganates all have interplanar periodicities similar to those of todorokite, a  
140 common tectomanganate with a uniform  $3 \times 3$  tunnel structure (Post and Bish, 1988),  
141 however, leading to their frequent confusion (Burns et al., 1983). Special care was  
142 thus paid to allow differentiating these species using nitric acid treatment (Wu et al.,

143 2019), Fourier transform infrared spectroscopy and anomalous X-ray diffraction. An  
144 optimum procedure allowing this differentiation is proposed.

## 146 2. MATERIALS AND METHODS

### 148 2.1. Synthesis of layered precursors and reflux products

149  
150 Both initial layered precursors and reflux products described in the present study  
151 were prepared as in our previous work on their Co analogues (Wu et al., 2019).  
152 Briefly, layered precursors were prepared after Feng et al. (2004) and Song et al.  
153 (2010) by adding a concentrated NaOH solution into a Ni(II) + Mn(II) solution,  
154 prepared from their chloride salts. The resulting solution was then mixed and bubbled  
155 with oxygen gas for 5 h. Resulting black precipitates were washed and one half was  
156 dried for subsequent analyses. Initial Ni / (Ni + Mn) ratios were 0.00, 0.01, 0.05, 0.10,  
157 0.15, and 0.20. Accordingly, initial layered samples were named Bir, NiB1, NiB5,  
158 NiB10, NiB15 and NiB20, and those Ni-containing samples are collectively referred  
159 to as NiB. The other half of the precipitates was used to prepare todorokite through a  
160 reflux process following the protocol of Feng et al. (2004). Wet precipitates were thus  
161 exchanged with  $Mg^{2+}$ , rinsed, re-suspended in deionized water, and finally refluxed at  
162 100 °C for 24 h. Reflux products were finally filtered, washed, and dried.  
163 Transformation products of Bir, NiB1, NiB5, NiB10, NiB15 and NiB20 are named  
164 Tod, NiT1, NiT5, NiT10, NiT15 and NiT20, respectively; they are collectively  
165 referred to as NiT except for Tod. Both layered precursors and reflux products were  
166 dried at 60 °C for three days and ground into fine powders.

### 168 2.2. Characterization of layered precursors and reflux products

169  
170 The elemental composition of all samples was determined in triplicate using  
171 atomic absorption spectrometry (Varian AAS 240FS) from 0.1000 g of sample



172 dissolved in 25 mL solution of 0.25 mol L<sup>-1</sup> NH<sub>2</sub>OH·HCl and 5 mL of 1 mol L<sup>-1</sup>  
173 H<sub>2</sub>SO<sub>4</sub>. In addition, the specific surface area (SSA) of the reflux products was  
174 determined with the Brunauer–Emmett–Teller (BET) method and a standard  
175 adsorption analyzer (Quantachrome Autosorb-1, JEDL-6390/LV). Finally, a  
176 non-reducing nitric acid treatment (Wu et al., 2019) was performed to differentiate  
177 tectomanganates (todorokite) from phyllomanganates (buserite-, lithiophorite-, or  
178 asbolane-like phases – Wu et al., 2019). Briefly, 0.1 g sample was put in 250 mL of a  
179 1 M HNO<sub>3</sub> solution with soft stirring at ambient temperature (20 °C). At different  
180 time intervals, 5 mL of the suspension was collected and readily filtered through a  
181 0.22 μm membrane to determine Mn, Ni, and Mg release. After 1 week of this HNO<sub>3</sub>  
182 treatment, residual samples were dried at 60 °C for 3 days and analyzed with X-ray  
183 diffraction (XRD) at a scan speed of 10 ° 2θ .min<sup>-1</sup>.

184 Mineralogy of all samples was determined from powder XRD analysis using a  
185 Bruker D8 Advance diffractometer equipped with Cu Kα source (λ = 0.15418 nm)  
186 and operated at 40 kV and 40 mA. XRD patterns were collected at 0.02 ° 2θ intervals  
187 using a continuous scan mode. XRD data was collected on fresh samples (dried at  
188 60 °C for three days) at a scan speed of 10 ° 2θ .min<sup>-1</sup>. To assess the stability of their  
189 hydration, XRD data was collected on fresh samples after heating to 140 °C. Finally,  
190 XRD patterns were collected at 1 ° 2θ .min<sup>-1</sup> for aged samples that were kept dry in  
191 the dark for three months. Synchrotron XRD patterns were also collected on selected  
192 reflux products on beamline BL14B1 at the Shanghai Synchrotron Radiation Facility  
193 (SSRF) and on beamline 4B9A at the Beijing Synchrotron Radiation Facility (BSRF).  
194 At BSRF, XRD data was collected at the Mn K-edge (6.5 keV), Ni K-edge (8.3 keV),  
195 and 12 keV with a typical beam size of 2 × 1 mm<sup>2</sup>, whereas the 18 keV data was  
196 collected at SSRF with a typical beam size of ~0.2 × 0.3 mm<sup>2</sup>. In both cases, X-rays  
197 were monochromatized with a double-crystal Si (111) monochromator. Finally,  
198 high-energy X-ray scattering data was collected in transmission mode on beamline  
199 11-ID-B at the Advanced Photon Source (Argonne National Laboratory, Argonne, IL),  
200 using a ~90 keV X-ray energy. Pair distribution functions (PDF) and differential  
201 PDFs were calculated as described elsewhere (Li et al., 2011; Wang et al., 2013).

202 Infrared spectra were collected for both layered precursors and reflux products  
203 using a Bruker VERTEX 70. For each sample, 64 spectra were collected and summed  
204 over the 4000–400  $\text{cm}^{-1}$  range with a 4  $\text{cm}^{-1}$  resolution (Zhao et al., 2015).  
205 Micromorphology of the reflux products and lattice periodicities were observed a Jeol  
206 JEM2100F transmission electron microscope (TEM) operated at 200 kV. TGA was  
207 performed on a NETZSCH TG 209 instrument with a heating rate of 10  $^{\circ}\text{C min}^{-1}$  and  
208 a  $\text{N}_2$  flow of 20  $\text{mL min}^{-1}$ . Finally, X-ray photoelectron O1s spectra were collected  
209 with a VG Multilab2000. X-ray photoelectron spectroscopy (XPS) measurements  
210 used an Al  $\text{K}\alpha$  X-ray source (1486.6 eV) at a chamber base pressure of  $3 \times 10^{-9}$  Torr  
211 (Yin et al., 2011a). The O 1s spectra were collected with a 25 eV pass energy and  
212 0.1 eV energy steps, and analyzed with the Avantage software. All samples were  
213 charge-corrected to C 1s with a binding energy of 284.80 eV collected from the  
214 surface adventitious carbon (Wang et al., 2010). A Shirley-type background was  
215 subtracted before deconvolution and fitting. A Lorentzian:Gaussian ratio of 30:70 was  
216 used for all the fittings.

217 Finally, X-ray absorption spectra were collected at the Mn and Ni K-edges on the  
218 1W1B beamline of BSRF to determine the relative proportions of Mn(II), Mn(III),  
219 and Mn(IV) species and the local environments of Mn and Ni in both layered  
220 precursors and reflux products. Data was recorded at room temperature in  
221 transmission (Mn) and fluorescence (Ni) modes, and the energy was calibrated using  
222 metallic Mn/Ni foils as references. Extended X-ray absorption fine structure (EXAFS)  
223 spectra were processed using Athena and Artemis (Ravel and Newville, 2005). X-ray  
224 absorption near-edge structure (XANES) spectra of both layered precursors and reflux  
225 products were used to determine Mn average oxidation state (AOS) with a specific  
226 linear combination fitting method (the so-called Combo method). Mn(II), Mn(III),  
227 and Mn (IV) reference compounds were those of the original study (Table 1 in  
228 Manceau et al., 2012).

229

230

## 3. RESULTS

### 3.1. Elemental composition

With increasing Ni content, the Mn content decreases from ~55.4 to ~44.8 wt.% for layered precursors and from ~54.4 to ~49.2 wt. % for reflux products (Table 1). The Ni/(Mn+Ni) molar ratios are essentially similar in layered precursors and their respective reflux products, with only a very limited fraction of Ni being lost either during the Mg-for-(Na,Ni) exchange or during the subsequent transformation process. In addition, the Mg wt.% in the reflux products significantly decreases from ~3.95 wt.% in the Ni-free Tod sample to ~2.11 wt.% in NiT20, thus suggesting a competition between Mg and Ni for interlayer and/or tunnel sites of layered precursors and/or of reflux products. Specific surface area (SSA) of NiT samples, as measured with the BET method, is systematically higher than that of Ni-free Tod, but no systematic evolution of SSA is observed with the increase of Ni content.

### 3.2. Powder X-ray diffraction

X-ray diffraction (XRD) patterns collected on fresh Bir (not shown) are essentially similar to those recorded after a few months of aging and display reflections at 7.2, 3.6, ~2.45, and ~1.42 Å, typical for birnessite (Drits et al., 1997; Lanson et al., 2002b). Similar reflections are observed for fresh NiB5 and NiB10 samples (Fig. S1), in addition to a broad maximum peaking at ~4.6 Å. The intensity of this maximum increases with increasing Ni concentration in the initial solution and is thus likely related to the presence of Ni (hydr)oxide (ICDD #13-0229), whose formation was reported under similar experimental conditions (oxidation of a Ni-rich alkaline solution – Feitknecht et al., 1956). Consistently, the intensity of this maximum is increased further for NiB15 and NiB20. These two samples exhibit also reflections at ~2.45, and ~1.42 Å, typical for layered manganates, the basal reflections

260 of which indicate a 9.9-10.0 Å layer-to-layer distance (Fig. S1). Upon moderate  
261 heating (140 °C), these reflections shift to indicate a 7.2 Å layer-to-layer distance  
262 typical for birnessite, as the result of interlayer cation partial dehydration (Golden et  
263 al., 1986; Manceau et al., 1987; Drits et al., 1997). The initial presence of two planes  
264 of interlayer H<sub>2</sub>O molecules in NiB15 and NiB20 (buserite) is most likely indicative  
265 of the presence of interlayer hydrated Ni(II) cations, whose hydration is reduced upon  
266 heating. Sample heating also results in a decrease of the intensity of the ~4.6 Å  
267 maximum.

268 After aging for about three months, all layered precursors but NiB20 display over  
269 the low-angle region (5-30 °2θ) strong basal reflections at ~7.2 and ~3.6 Å typical for  
270 birnessite and indicative of the spontaneous dehydration of NiB15 interlayers (Fig. 1).  
271 Two planes of H<sub>2</sub>O molecules are still present in most NiB20 interlayers (buserite),  
272 however. Splitting of reflections at ~2.45 and ~1.42 Å, and more generally positions  
273 of the reflections over the high-angle region (30-85 °2θ – Fig. 1), unambiguously  
274 indicate that initial Bir is a triclinic birnessite (ICDD #43-1456), consistent with the  
275 synthesis protocol (Lanson et al., 2002b). With increasing Ni content, high-angle  
276 reflections broaden significantly thus indicating a decrease in the coherent scattering  
277 domain size. In addition, the resolution of these reflections decreases also with  
278 increasing Ni content, suggesting a minor increase of layer symmetry from orthogonal  
279 ( $a > b\sqrt{3}$ ,  $\gamma = 90^\circ$ ) to hexagonal ( $a = b\sqrt{3}$ ,  $\gamma = 90^\circ$  – Yin et al., 2015). Positions of  
280 birnessite *hkl* reflections remain essentially unchanged, however, indicative of the  
281 persistence of the orthogonal layer symmetry in Ni-rich layered precursors. The broad  
282 maximum at ~4.6 Å is present also in aged samples.

283 Reflux products systematically show reflections at ~9.6 Å, ~4.8 Å, ~3.2 Å,  
284 ~2.46 Å, ~2.39 Å, ~2.22 Å, ~1.95 Å, ~1.73 Å, ~1.52 Å, and ~1.42-1.41 Å, that are  
285 typical of todorokite (ICDD #38-0475 – Atkins et al., 2014, 2016). For the Ni-free  
286 Tod, relative intensity of the three low-angle reflections (at ~9.6, ~4.8, and ~3.2 Å)  
287 decreases in the order  $I_{9.6} > I_{4.8} \gg I_{3.2}$ , consistent with relative intensities of  
288 todorokite 001/100, 002/200, and 003/300 reflections (Post et al., 2003), respectively.  
289 With increasing Ni content, the positions of both ~9.6 and ~4.8 Å reflections shift

290 towards higher angles for NiT15 and NiT20, compared to Ni-free Tod (Fig. S2) and  
291 the  $\sim 4.8$  Å reflection becomes predominant, consistent with previous reports of  
292 todorokite prepared in the presence of various divalent metals (Shen et al., 1993, 1994;  
293 Ching et al., 1999; Nicolas-Tolentino et al., 1999; Fuertes et al., 2012) and  
294 asymmetric. Although not described as such in these references, this intensity ratio  
295 modification is consistent with the presence of asbolane, a phyllosilicate with an  
296 incomplete – island-like – octahedral layer of metal (hydr)oxides in the interlayer  
297 (Chukhrov et al., 1987; Golden et al., 1987; Fan and Gerson, 2015). The presence of  
298 these incomplete brucite-like octahedral layers between MnO<sub>2</sub> layers precludes the  
299 collapse of the layer-to-layer distance to  $\sim 7.2$  Å upon heating (Fig. S3)

300

### 301 3.3. Anomalous XRD patterns

302

303 To investigate further the possible origin of the observed contrasting  $I_{4,8}/I_{9,6}$   
304 intensity ratios, XRD patterns of Tod, NiT15, and NiT20 were recorded at different  
305 energies, including Mn and Ni K-edges to modify the contribution of these elements  
306 to the structure factor ( $F_{hkl}$ ). These XRD patterns logically exhibit reflections at  
307 similar d-values (Fig. 2a-d). Except for the pattern recorded at the Mn K-edge all Tod  
308 patterns also display similar low  $I_{4,8}/I_{9,6}$  intensity ratios (Fig. S4), consistent with  
309 previous reports (Shen et al., 2011). Contrastingly, the  $I_{4,8}/I_{9,6}$  intensity ratio is  
310 significantly higher for NiT15 and NiT20 compared to Tod, and varies significantly  
311 for these two samples as a function of the energy used for data collection. Specifically,  
312 NiT15 and NiT20 display highest  $I_{4,8}/I_{9,6}$  intensity ratios at the Mn K-edge and lowest  
313 ratios at the Ni K-edge, whereas similar ratios are obtained for XRD patterns recorded  
314 at energies different from the two absorption edges (Figs. 2a-d; S4). Both features  
315 indicate contrasting structures of Tod and Ni-rich NiT products depending on the  
316 initial Ni content.

317 As reported in the early descriptions of asbolane, this compound exhibits high  
318 values of the  $I_{4,8}/I_{9,6}$  intensity ratio (Chukhrov, 1982; Llorca, 1988; Fan and Gerson,  
319 2015), consistent with XRD patterns calculated for such a structure hosting an

320 incomplete Ni(OH)<sub>2</sub> layer sandwiched in between two MnO<sub>2</sub> layers (Fig. 2e). In  
321 addition, for a given composition, this ratio increases when the contribution of Mn  
322 atoms to the structure factor is reduced (XRD data recorded at the Mn K-edge – Fig.  
323 2a) and decreases when the contribution of Ni atoms to the structure factor is reduced  
324 (XRD data recorded at the Ni K-edge – Fig. 2b), consistent with the data. In addition,  
325 this ratio increases with the completeness of interlayer plane (Fig. 2e). Together with  
326 the collapse of layer-to-layer distance upon acid-dissolution of the incomplete  
327 Ni(OH)<sub>2</sub> layer (see below), these variations of the I<sub>4.8</sub>/I<sub>9.6</sub> intensity ratio can be  
328 ascribed to the presence of asbolane, rather than todorokite, in the reflux products.

329

### 330 3.4. FTIR

331

332 FTIR spectra of todorokite and of other 3×n tectomanganates display a band at  
333 ~760 cm<sup>-1</sup>, typical for the corner-sharing linkage in their structures (Fig. 3 – Golden et  
334 al., 1986; Feng et al., 2004; Julien et al., 2004; Atkins et al., 2014; Zhao et al., 2015).  
335 This band is clearly visible in Tod FTIR spectrum but steadily decreases in intensity  
336 with increasing Ni content, most likely as the result of the decreasing proportion of  
337 3×n tectomanganates in Ni-rich reflux products. Consistent with their layered  
338 structure, this band is absent in FTIR spectra of layered precursors. Rather, two lines  
339 at ~820 and ~950 cm<sup>-1</sup> are visible in the FTIR spectra of Ni-containing layered  
340 precursors. These two lines are visible also in the spectra of related reflux products,  
341 although their intensity is reduced after the reflux treatment. The peak at ~820 cm<sup>-1</sup>  
342 visible in NiB and NiT was identified as a combination lattice mode band in Ni  
343 (hydr)oxides (Hall et al., 2015), which is consistent with the XRD analysis.

344

### 345 3.5. HRTEM

346

347 Consistent with previous studies (Turner and Buseck, 1981; Golden et al., 1986;  
348 Shen et al., 1993; Feng et al., 2004; Wu et al., 2019), crystals with fibrous and  
349 plate-like morphologies are systematically present in the reflux products (Fig. 4), the

350 former dominating in Tod and NiT1 (Figs. 4a-c), whereas platy crystals dominate  
351 samples having higher Ni-contents (NiT5 and NiT15; Figs. 4f, i). Fiber-shaped  
352 crystals (Figs. 4 e, g, h) often display lattice fringes with a typical ~1 nm periodicity,  
353 corresponding to the 3×3 todorokite tunnel structure (Golden et al., 1986; Feng et al.,  
354 2004). As it is common in todorokite, fibers are often aligned at  $\pm 120^\circ$  with respect to  
355 each other, thus forming trilling patterns (Siegel and Turner, 1983; Golden et al., 1986;  
356 Feng et al., 2004; Bodei et al., 2007; Xu et al., 2010; Atkins et al., 2014). Visible  
357 lattice fringes in the plates are commonly more diverse than in the fibers (Figs. 4 b, d  
358 and j), typical for  $\alpha$ -disordered todorokite (Xu et al., 2010; Wu et al., 2019). Lattice  
359 fringes are also visible in short rods and small plates, indicating the initial  
360 development of tunnel structure therein (Figs. 4 g, h – Golden et al., 1986). Finally,  
361 smooth surfaces without visible fringes, that are most likely typical for  
362 phyllomanganate crystals (Golden et al., 1986, 1987; Wu et al., 2019), are also visible  
363 (Figs. 4 g – zone 4, 4i, j – zone 1, 4k – zone 5). Interestingly, some platy particles  
364 exhibit lattice fringes over specific zones with smooth surfaces (Figs. 4g, j). This dual  
365 character is possibly related to the initial development of tunnel structures in specific  
366 zones of the particles whereas other zones maintain their initial planar structure  
367 (Golden et al., 1987; Wu et al., 2019).

368

### 369 3.6. Nitric acid treatment

370

371 Upon equilibration in nitric acid, Mn release from layered precursors reaches  
372 equilibrium within 12 hrs, the proportion of Mn released from NiB is higher than that  
373 released from Bir, (Fig. 5a), possibly as an effect of reduced crystal size. For reflux  
374 products, the release of Mn is slower, reaching equilibrium after ~48 hrs, and is  
375 essentially the same for all NiT samples (Fig. 5b), except for NiT5. Ni release is much  
376 slower than that of Mn, reaching a plateau after ~168 hrs in both layered precursors  
377 (Fig. 5c) and reflux products (Fig. 5d). The proportion of Ni released differs however  
378 in the two series of samples as essentially all Ni is released from layered precursors  
379 whereas ~20% of Ni remains in the reflux products, possibly indicative of Ni

380 structural incorporation. Finally, Mg is systematically totally released from reflux  
381 products, possibly at a slower rate with increasing Ni content (Fig. 5f).

382 After acid treatment, the XRD pattern of Tod is essentially alike that of the initial  
383 reflux product and typical for todorokite (Fig. 5e). On the other hand, XRD patterns  
384 recorded on NiT15 and NiT20 after acid treatment exhibit, over the low-angle region,  
385 peaks at  $\sim 7.2$  and  $\sim 3.6$  Å, typical for the phyllomanganate birnessite, rather than  
386 peaks at  $\sim 9.6$ ,  $\sim 4.8$ , and  $\sim 3.2$  Å, as expected for todorokite (Fig. 5e). These two XRD  
387 patterns are actually similar to the one recorded on Bir after acid treatment and match  
388 XRD pattern of hexagonal birnessite (ICDD #23-1239), which was obtained from the  
389 low-pH equilibration of an initial triclinic birnessite (Lanson et al., 2000). The  
390 evolution of XRD patterns recorded from reflux products upon acid treatment is  
391 consistent with that observed for an asbolane-like phyllomanganate (Wu et al., 2019).  
392 The structure of these  $\sim 9.6$  Å phyllomanganates consists of an octahedral  $\text{MnO}_2$  layer  
393 similar to that of birnessite, with an incomplete octahedral  $\text{Me}(\text{OH})_2$ ,  $\text{MeOOH}$ , or  
394  $\text{MeO}_2$  sheet, possibly incommensurate with the  $\text{MnO}_2$  layer, in the interlayer  
395 (Chukhrov, 1982; Manceau et al., 1987; Fan and Gerson, 2015). This incomplete  
396 interlayer octahedral sheet is solubilized by the acid treatment leading to the observed  
397  $7.2$  Å layer-to-layer distance (Wu et al., 2019). Such a peak at  $\sim 7.2$  Å is visible in the  
398 XRD pattern of acid-treated NiT5 and NiT10, and its intensity increases with  
399 increasing Ni content, becoming the sole visible diffraction signature in acid-treated  
400 NiT15 and NiT20 (Fig. 5e).

401

### 402 3.7. Thermogravimetric analysis

403

404 Thermogravimetric analyses of layered precursors systematically show a  
405 significant weight loss at  $140$ - $150$  °C corresponding to the dehydration of interlayer  
406  $\text{Na}^+$  and  $\text{Ni}^{2+}$  cations, this weight loss decreasing with increasing Ni content (Figs. 6a,  
407 c; Table 2). A minor weight loss ( $2.4$ - $2.9\%$  in Bir, NiB5, NiB10, and NiB15) is also  
408 observed for these samples below  $100$  °C, which is most likely related to the  
409 departure of adsorbed water (Gaillot et al., 2003). This low-temperature weight loss is



410 significantly increased for NiB20 (9.25%), however, most likely as the result of a  
411 low-temperature buserite-to-birnessite transition associated with the loss of one plane  
412 of interlayer H<sub>2</sub>O molecules as observed during moderate heating of fresh NiB  
413 samples (Fig. S1). A significant weight loss (4.10 and 6.72% for NiB5 and NiB20,  
414 respectively) is observed also for NiB samples over the 170-400 °C range. This  
415 weight loss is essentially absent in the TGA curve of Bir (1.78%) and increases with  
416 Ni content. It is likely related to the presence of the Ni (hydr)oxide responsible for the  
417 broad XRD maximum peaking at ~4.6 Å (Fig. 1). Finally, a weight loss is observed at  
418 ~550 °C for all layered precursors, with no significant influence of the Ni content,  
419 probably indicating the reduction of MnO<sub>2</sub> to Mn<sub>2</sub>O<sub>3</sub> (Bish and Post, 1989).

420 TGA curves recorded for reflux products significantly differ from those of their  
421 layered precursors, with weight losses at ~150, ~285, ~440, and ~545 °C for Tod  
422 (Figs. 6b, d). The weight loss observed from ~110-240 °C is most likely related to the  
423 loss of the H<sub>2</sub>O molecules hydrating cations present in todorokite tunnels (Bish and  
424 Post, 1989). With increasing Ni content, this weight loss decreases (from 8.05 to 6.12%  
425 in Tod and NiT20, respectively) and shifts towards lower temperatures, most likely  
426 indicative of the mineralogical evolution evidenced by XRD data recorded on  
427 acid-treated reflux products (Fig. 5e). For NiT10, this low-temperature weight loss  
428 splits in a doublet owing to the coexistence of todorokite and asbolane (Fig. 5e). The  
429 weight loss occurring at ~285 °C is also ascribed to the departure of tunnel water  
430 (Bish and Post, 1989). TGA curves obtained for Tod and NiT20 also differ  
431 significantly over this range, with a shift of the weight loss towards higher  
432 temperatures in NiT20 compared to Tod. Evolution with increasing Ni content, is not  
433 as systematic as that observed for the low-temperature weight loss, however (Fig. 6d).  
434 The third weight loss occurring at ~440 °C appears as a distinct signature for the  
435 presence of todorokite, as its intensity decreases with increasing Ni content to vanish  
436 for NiT15 and NiT20 which essentially contain asbolane (Fig. 5e). A last weight loss  
437 occurs at 520-550 °C, a temperature similar to that reported for layered precursors at  
438 ~550 °C (Figs. 6b, d).

439

### 440 3.8. O1s X-ray photoelectron spectroscopy

441

442 Consistent with the systematic prevalence of  $[\text{MnO}_6]$  building octahedral in both  
443 phyllomanganates and tectomanganates, all O1s XPS spectra are dominated by a peak  
444 at  $\sim 529.6$  eV (Fig. 7), typical for oxygen ( $\text{O}^{2-}$ ) in metal oxides. This main peak is  
445 asymmetric and displays a tail on its high-energy side for both layered precursors and  
446 reflux products. This tail at  $\sim 531$ - $535$  eV can be decomposed into the contributions of  
447 oxygen atoms from hydroxyl groups ( $\text{OH}^-$ ;  $530.9$ - $531.6$  eV) and from water  
448 molecules ( $532.2$ - $533.4$  eV). For layered precursors, the intensity of this tail, and  
449 more especially that of the  $\text{OH}^-$  contribution, increases with increasing Ni content (Fig.  
450 7; Table S1), whereas the  $\text{H}_2\text{O}$  contribution is about constant for all NiB samples but  
451 NiB20 that shows a slightly higher proportion of O atoms from  $\text{H}_2\text{O}$  molecules,  
452 consistent with TGA data. In reflux products the contribution of O atoms in hydroxyl  
453 groups is significantly higher in NiT samples compared to Tod, consistent with the  
454 presence of asbolane-like phyllomanganate in Ni-rich reaction products. On the other  
455 hand, the  $\text{H}_2\text{O}$  contribution shows no systematic evolution with Ni content (Fig. 7;  
456 Table S1). Finally, relative intensities of peaks corresponding to  $\text{OH}^-$  and  $\text{H}_2\text{O}$   
457 contributions are significantly increased in NiT samples compared to their  
458 corresponding layered precursors, at least under the deep vacuum conditions  
459 prevailing during the XPS measurements.

460

### 461 3.9. X-ray absorption spectroscopy

462

#### 463 3.9.1. Mn K-edge XANES

464 Relative proportions of Mn(II), Mn(III), and Mn(IV) species and Mn average  
465 oxidation state (AOS) have been determined using the Combo method (Table 3 –  
466 Manceau et al., 2012). Results consistently indicate high ( $\sim 1/3$ ) relative proportions of  
467 Mn(III) in both layered precursors and reflux products, which is considered a key  
468 condition for the phyllomanganate-to-tectomanganate conversion (Cui et al., 2008;  
469 Atkins et al., 2014, 2016; Grangeon et al., 2014, 2015; Zhao et al., 2015; Li et al.,

470 2016). The measured proportion of Mn(III) is slightly lower in Ni-free samples  
471 compared to the Ni-bearing ones (Table 3) and no significant difference is observed  
472 between layered precursors and their reflux products. In all cases, the Mn(II) content  
473 is null within the experimental error.

474

### 475 3.9.2. Mn K-edge EXAFS

476 EXAFS spectra of both layered precursors and of reflux products show  
477 significant evolutions with increasing Ni content, thus indicating that the presence of  
478 Ni affects the local structure of manganates in the two series of samples (Fig. 8). The  
479 EXAFS spectrum collected for Ni-free layered precursor (Bir) is typical for triclinic  
480 birnessite over both the staircase and the indicator regions ( $3.5\text{-}6.0 \text{ \AA}^{-1}$  and  
481  $7.5\text{-}10.0 \text{ \AA}^{-1}$ , respectively – Marcus et al., 2004). In particular, the splitting of the  
482 features at  $\sim 7.8 \text{ \AA}^{-1}$  and  $\sim 9.0 \text{ \AA}^{-1}$  (arrows in Fig. 8a) are indicative of a high Mn(III)  
483 content, of the systematic elongation of Mn(III) octahedra along the *a* axis, and of  
484 their ordered distribution in rows parallel to the *b* axis (Marcus et al., 2004; Gaillot et  
485 al., 2007), typical features for triclinic birnessite (Webb, 2005; Atkins et al., 2014; Yin  
486 et al., 2014). With increasing Ni content, the  $\sim 7.8 \text{ \AA}^{-1}$  feature becomes less  
487 pronounced even though its position remains unchanged, consistent with the similar  
488 Mn(III) contents of all layered precursors (Table 3) and with the similar overall  
489 frequency of the EXAFS spectra of these samples (Gaillot et al., 2007). The steady  
490 lowering of the  $\sim 7.8 \text{ \AA}^{-1}$  feature resolution is thus most likely related either to disorder  
491 in the distribution of elongated Mn(III) octahedral or to some randomness in the  
492 direction of the elongation, consistent with the evolution of XRD patterns with  
493 increasing Ni content. The split  $\sim 9.0 \text{ \AA}^{-1}$  feature observed for Bir and NiB1 also turns  
494 into a broad and more symmetrical feature with increasing Ni content and the induced  
495 vanishing of the sharp high-frequency contribution.

496 In k-space Mn K-edge EXAFS spectra of Tod and NiT1 exhibit sharp features at  
497  $\sim 8.0$  and  $\sim 9.2 \text{ \AA}^{-1}$  that are typical for todorokite (Webb, 2005; Bodeř et al., 2007; Feng  
498 et al., 2010; Atkins et al., 2014). These two features essentially decrease in intensity at  
499 higher Ni content. In addition, the  $\sim 9.2 \text{ \AA}^{-1}$  feature shifts to lower wave numbers in

500 NiT samples compared to Tod and NiB1 (Fig. 8b). EXAFS spectra of Ni-rich NiT  
501 samples and their Fourier transforms (FTs) are actually alike those of their  
502 corresponding layered precursors (Figs. 9a-b), consistent with the similar structure of  
503 the MnO<sub>2</sub> layer in both birnessite and asbolane. On the contrary, spectra obtained for  
504 Ni-free Bir and Tod differ significantly, as expected for layered and tunnel manganese  
505 oxides (Fig. 9c). In particular, the FTs of these spectra show a clear shortening of the  
506 average Mn-Me<sub>E</sub> distance in Tod compared to Bir, as shown by the shift of the  
507 imaginary part of the FT (Fig. 9d, arrow). In addition, Tod FT shows a clear  
508 maximum at  $R + \Delta R \sim 3.0 \text{ \AA}$ , typical of Mn-Me<sub>C</sub> pairs, which is not observed for its  
509 Bir precursor. For both Ni-free and Ni-rich samples, Mn-O distances are similar in  
510 both layered precursors and corresponding reflux products, consistent with their  
511 similar Mn AOS.

512

### 513 3.9.3. Ni K-edge EXAFS

514 EXAFS spectra collected at the Ni K-edge are similar for both NiB samples, on  
515 the one hand, and NiT, on the other hand (Fig. 10), thus implying similar Ni  
516 speciation in each of the two sample groups. Although it was not possible to obtain a  
517 decent fit to this data using a linear combination fitting and available Ni reference  
518 spectra, similarity of the hydrated Ni(II) spectrum with collected spectra suggests that  
519 this species possibly prevails in all samples. In addition, the evolution of Ni speciation  
520 during the reflux process was assessed by fitting EXAFS spectra of reflux products as  
521 a linear combination of their respective NiB precursors and of Ni reference spectra  
522 (Fig. S4). For all NiT samples, a good fit was obtained with ~15% of a Ni(OH)<sub>2</sub>  
523 contribution in addition to the layered precursor (Table S2), consistent with the  
524 polymerization of a partial Ni(OH)<sub>2</sub> layer in birnessite interlayers to form an  
525 asbolane-like structure.

526

### 527 3.10. Pair distribution functions (G(r))

528

529 Atomic PDFs computed from Tod, NiT10, and NiT20 high-energy X-ray

530 scattering patterns show similar peak positions (Fig. 11), indicative of similar local  
531 structures (Zhu et al., 2012) as expected from the major contribution of layers/ribbons  
532 of edge-sharing  $\text{MnO}_6$  octahedra in both layered precursors and reflux products.  
533 Consistently, the maxima at  $\sim 1.90 \text{ \AA}$ ,  $\sim 2.90 \text{ \AA}$ , and  $\sim 3.50 \text{ \AA}$  essentially correspond to  
534 the first neighbor Mn-O, to edge-sharing Mn-Me, and to a combination of  
535 corner-sharing Mn-Me and of second neighbor Mn-O atomic pairs, respectively (Zhu  
536 et al., 2012). The increased intensity of the  $\sim 3.50 \text{ \AA}$  maximum and the decreased  
537 intensity of the one at  $\sim 5.00 \text{ \AA}$  in the PDF of Tod compared to NiT10 and NiT20 are  
538 typical of the prevalence of tunnel structures in the former sample and of layered  
539 structures in the latter (Grangeon et al., 2015). The  $\sim 5.00 \text{ \AA}$  maximum is mainly  
540 related indeed to second neighbor Mn-Mn pairs (Fig. S4 – Zhu et al., 2012), that are  
541 favored by the lateral extension of  $\text{MnO}_2$  layers. Finally, the peak at  $\sim 4.50 \text{ \AA}$  is most  
542 likely related to Mn2-Mn2 and Mn4-Mn4 pairs across “holes” in tunnel walls, floors,  
543 and ceilings (Fig. S6), consistent with the increased intensity of this  $\sim 4.50 \text{ \AA}$  peak  
544 with decreasing tunnel size (Zhu et al., 2012).

545 Differential PDF (d-PDF) can be calculated to assess more precisely the specific  
546 modifications affecting the reflux products, and more especially the broadening of the  
547  $\sim 1.90$  and  $2.90 \text{ \AA}$  peaks towards high d-values with increasing Ni content (Fig. 11).  
548 The d-PDF curves obtained for NiT10 and NiT20, both show a first peak at  $\sim 2.05 \text{ \AA}$   
549 typical for Ni-O pairs (Manceau et al., 2007a,b; Peacock and Sherman, 2007a,b; Yin  
550 et al., 2012; Kwon et al., 2013; Simanova et al., 2015; Atkins et al., 2016). A second  
551 peak, similar in both d-PDF curves, is visible at  $\sim 3.05 \text{ \AA}$ . This second peak is  
552 intermediate between those calculated Ni-Ni pairs in NiO and  $\text{Ni(OH)}_2$ , consistent  
553 with the presence of both compounds in the reflux products identified from XRD data  
554 (Figs. 1, 2).

555

556

## 4. DISCUSSION

557

558 **4.1. Structural transformation of layered precursors**

559

560 Addition of Ni during the synthesis of layered precursors does not impact  
561 significantly their crystal structure. In particular, the Mn(III) content of layered  
562 precursors remains about constant at  $\sim 1/3$  (Table 3) whatever the initial Ni content.  
563 Consistently, XRD results (Fig. 1) show that layer symmetry remains orthogonal ( $a >$   
564  $b\sqrt{3}$ ) for all precursors, indicative of the systematic elongation of distorted Mn(III)  
565 octahedra along the  $a$  axis and of their ordered distribution. FTs of the EXAFS spectra  
566 consistently show similar Mn-Me<sub>E</sub> distances in all layered precursors (not shown).  
567 Both XRD patterns and EXAFS spectra of layered precursors are significantly  
568 modified when increasing the Ni content, however. These changes reveal an increased  
569 disorder in the orientation of Mn(III) octahedra and/or in their distribution.

570 The increasing intensity of the  $\sim 4.6$  Å diffraction maximum with increasing Ni  
571 content indicates that part of Ni initially introduced in solution actually precipitates as  
572 Ni (hydr)oxide. The presence of Ni in the interlayers of NiB15 and NiB20 is attested  
573 by the presence of two planes of interlayer water molecules, consistent with its more  
574 negative hydration enthalpy compared to Na<sup>+</sup> (Smith, 1977; Johnson and Post, 2006),  
575 in these fresh samples. All Ni species, including the two mentioned above, are readily  
576 mobilized by the acid treatment (Fig. 5c), and it is thus reasonable to hypothesize that  
577 no Ni(II) cations are incorporated in the octahedral layer. Similarly, the orthogonal  
578 layer symmetry is typical of octahedral layers containing few vacant layer sites (Drits  
579 et al., 1998; Lanson et al., 2000), if any, and the presence of Ni(II) cations sorbed  
580 above or below such vacant sites is thus unlikely.

581 Although weakly bound to the layered precursors, the presence of Ni(II)  
582 significantly impacts their transformation during the reflux process. Bir, a typical  
583 triclinic birnessite, converts readily and completely to an  $a$ -disorder todorokite (Wu et  
584 al., 2019). With increasing Ni content, disorder increases together with the proportion  
585 of non-ideal  $3 \times n$  tunnel sizes forming trilling patterns. In addition, platy crystals are  
586 visible in all reflux products, their relative abundance increasing with increasing Ni  
587 content with only minimal fragments of tunnel oxides being detected in NiT15 and  
588 NiT20 (Fig. 4). Platy crystals are most likely asbolane, as shown by the prevalence of

589 this phyllomanganate hosting an incomplete octahedral layer of metal (hydr)oxides in  
590 their interlayers in Ni-rich reflux products. These incomplete brucite-like octahedral  
591 layers are dissolved by the acid treatment, leading to birnessite-like structures (Fig.  
592 5e). As reported in the early work of Chukhrov and coworkers on natural asbolane  
593 (Chukhrov, 1982; Chukhrov et al., 1987), XRD diffraction patterns of asbolane are  
594 characterized by high  $I_{4.8}/I_{9.6}$  intensity ratios. Such high  $I_{4.8}/I_{9.6}$  intensity ratios are  
595 commonly reported for “todorokite”, especially when synthesized in the presence of  
596 metallic elements such as Co, Ni, Cu, Zn, Er, La, etc (Shen et al., 1994; Tian et al.,  
597 1997; Ching et al., 1999; Luo et al., 1999). In the present study, the combined use of  
598 XRD simulation, anomalous XRD, HRTEM, thermogravimetric analysis, and nitric  
599 acid treatment allowed uncovering the actual nature of this seldom described  
600 phyllomanganate, but its formation may be far more common than previously  
601 reported in these experimental studies.

602 As shown in Fig. 2e, this  $I_{4.8}/I_{9.6}$  intensity ratio strongly depends on the  
603 completeness of the interlayer brucite-like octahedral layer, and the ratio determined  
604 for NiT15 and NiT20 corresponds to a ~50% completeness of the  $\text{Ni}(\text{OH})_2$  octahedral  
605 layer. This interlayer occupancy is much higher than possibly achieved with the sole  
606 presence of Ni(II) in these octahedral layers as Ni / (Ni+Mn) ratios are 8-12% in these  
607 samples, with part of the Ni present as Ni (hydr)oxide. It is thus likely that Mg cations  
608 contribute to the electron density at the interlayer mid-plane, in addition to Ni. As Mg  
609 scattering factor is lower than that of Ni, the actual completeness of the interlayer  
610 brucite-like layer is likely higher than 50%.

611

## 612 **4.2. Differentiation of asbolane and todorokite.**

613

614 Although the  $I_{4.8}/I_{9.6}$  intensity ratio strongly depends on the completeness of the  
615 interlayer brucite-like octahedral layer, a 4.8 Å peak stronger than its 9.6 Å  
616 counterpart is a possible indication of the presence of the phyllomanganate asbolane,  
617 rather than the tectomanganate todorokite, in reaction products, or in natural samples.  
618 The two compounds indeed share these two interplanar distances. With its two planes

619 of interlayer H<sub>2</sub>O molecules, busserite also exhibits similar periodicity but a gentle  
620 heating (to ~100°C) should lead to the partial dehydration of interlayer cations and to  
621 the ~7.2 Å periodicity typical of birnessite (Fig. S1 – Johnson and Post, 2006).  
622 Furthermore, busserite is unlikely to have a high enough content of interlayer cations  
623 to induce a I<sub>4.8</sub>/I<sub>9.6</sub> ratio higher than 1.0.

624 Although values of this ratio higher than 1.0 are typically observed for asbolane,  
625 the partial exchange of Mg present in todorokite tunnels by cations having a higher  
626 electronic content (such as Ni and other divalent metal cations) can also lead to such  
627 high I<sub>4.8</sub>/I<sub>9.6</sub> ratios (Fig. 11), thus not allowing an unambiguous differentiation  
628 between asbolane and todorokite. In this search for an unquestionable differentiation  
629 between asbolane and todorokite, the nitric acid treatment proposed by Wu et al.  
630 (2019) thus appears as a simple and reliable method. In the case of asbolane,  
631 dissolution of the interlayer brucite-like layer induced by the treatment leads indeed to  
632 a collapse of the layer-to-layer distance to the ~7.2 Å periodicity. On the contrary,  
633 periodicities are maintained for todorokite, the tectomanganate framework being  
634 unaltered, and positions of XRD maxima are thus unchanged.

635

### 636 **4.3. The role of Mn(III) and interlayer species during the transformation**

637

638 It is now accepted that the presence of a high Mn(III) content (>25-30%) in  
639 layered precursors is required for their conversion to tectomanganates. This pivotal  
640 role is most likely related to the steric strains induced by the presence of Jahn-Teller  
641 distorted Mn(III) octahedra in the octahedral layers. Migration of these Mn(III)  
642 cations from layer to interlayer or layer kinking at these structurally weak points  
643 contribute to building tunnel walls (Grangeon et al., 2008, 2014; Atkins et al., 2014,  
644 2016; Yang et al., 2018). In the present case, the ordered distribution of Mn(III)  
645 cations in rows parallel to the *b* axis and separated from each other along the *a* axis by  
646 two rows of Mn(IV) creates a structurally weak point favorable to the formation of  
647 tunnels walls (Atkins et al., 2014; Grangeon et al., 2014). Consistently, the initial  
648 Ni-free precursor Bir, a triclinic birnessite, was fully converted to todorokite. Similar



649 transformation of triclinic birnessite to tunnel structures was previously reported in  
650 the literature although 3×3 tunnels were not systematically obtained (Yang et al.,  
651 2018). Furthermore, the actual distribution and orientation of Mn(III) octahedra  
652 present in the MnO<sub>2</sub> layer do not appear to be key factors for the  
653 phylломanganate-to-tectomanganate conversion as tectomanganates were obtained  
654 from hexagonal birnessite precursors in which Mn(III) octahedra are distributed  
655 and/or oriented at random (Grangeon et al., 2008, 2014; Atkins et al., 2014, 2016;  
656 Zhao et al., 2015). The absence of transformation of NiB precursors to tunnel  
657 structures thus does not appear to be related to their Mn(III) content, which is similar  
658 to or higher than that of the Ni-free Bir.

659 As layers of all NiB precursors are essentially devoid of Ni, both in the octahedral  
660 layer and sorbed above/below vacant layer sites, the absence of transformation of  
661 these precursors to tunnel structures is thus likely due to kinetic effects favoring the  
662 formation of asbolane. Specifically, polymerization of Ni as Ni(OH)<sub>2</sub> fragments in the  
663 interlayers of initial birnessite/buserite (Fig. S5, Table S2) is favored over the  
664 conversion to tectomanganates. In turn, the presence of interlayer Ni(OH)<sub>2</sub> fragments  
665 likely impedes the migration of Mn(III) from the layer to the interlayer thus  
666 preventing the conversion to tectomanganates. Such competing effects are supported  
667 also by the increase of (Mg + Ni)/Mn ratios with increasing Ni content (Table 1) and  
668 are likely to occur with a variety of other hydrolysable metal cations. Consistently,  
669 “todorokite” containing divalent metal cations and displaying high I<sub>4.8</sub>/I<sub>9.6</sub> ratios  
670 possibly indicate the presence of asbolane-like structure. Such competing effects  
671 might contribute also to the prevalence of phylломanganates over tectomanganates in  
672 natural environments (Wu et al., 2019).

673

#### 674 **4.4. Negative influence of Ni and Co on the phyllo-to-tectomanganate** 675 **transformation: Contrasting mechanisms**

676

677 Both in the present work investigating the possible transformation of  
678 Ni-containing layered precursors to tectomanganates and in the related study

679 investigating a similar process for Co-containing layered precursors, tectomanganate  
680 formation was inhibited, totally (Ni) or partially (Co) for Me-rich precursors. The  
681 origin of these similar negative impacts differs however for the two metals. In the  
682 present case, layered precursors contain sufficient amounts of Mn(III) (~1/3) to  
683 trigger the transformation, as Ni(II) is not structurally incorporated in the precursors.  
684 As discussed above, polymerization of Ni(OH)<sub>2</sub> fragments in birnessite interlayers is  
685 kinetically favored over migration of Mn(III) from the layer to the interlayer, or layer  
686 kinking, however, leading to the formation of asbolane rather than tectomanganates.  
687 In contrast and consistent with previous reports (Manceau et al., 1997; Yu et al., 2012;  
688 Kwon et al., 2013; Simanova and Pena, 2015; Yin et al., 2015), Co(II) is readily  
689 oxidized by birnessite during the formation of initial layered precursors with Co(III)  
690 being structurally incorporated in these precursors owing to the similar ionic radii of  
691 Co(III) and Mn(IV) (Wu et al., 2019). The symmetry of layered precursors is  
692 increased from orthogonal to hexagonal. As a consequence, the content of Mn(III) in  
693 Co-containing layered precursors is lowered and does not allow triggering the  
694 transformation to tunnel structures (Grangeon et al., 2008, 2014; Atkins et al., 2014,  
695 2016). In both cases, little Me is released to solution during the reflux process and Me  
696 cations are structurally incorporated in the reflux products (~20% and ~80% for Ni  
697 and Co, from the nitric acid experiments and EXAFS simulation, respectively) thus  
698 enhancing their sequestration in both tecto- and phyllomanganates.

699

#### 700 **4.5. The fate of Nickel during the transformation of layered precursors**

701

702 Previous studies have consistently shown that Ni can be incorporated in the  
703 birnessite structure, either within the octahedral layer or sorbed at vacant layers sites  
704 (Peacock and Sherman, 2007b; Grangeon et al., 2008, 2017; Peacock, 2009; Peña et  
705 al., 2010; Yin et al., 2012; Kwon et al., 2013; Yin et al., 2014; Simanova et al., 2015;  
706 Atkins et al., 2016; Lefkowitz and Elzinga, 2017), consistent with natural occurrences  
707 (Bodeř et al., 2007; Manceau et al., 2007a,b; Peacock and Sherman, 2007a). In their  
708 literature review, Grangeon et al. (2017) showed that the fraction of Ni structurally

709 incorporated in birnessite octahedral layers may be high (up to ~50%) only for low Ni  
710 content (~1%), as reported for natural samples (Bodeř et al., 2007; Manceau et al.,  
711 2007a,b; Peacock and Sherman, 2007a). Structural incorporation of Ni(II) in the  
712 octahedral layer appears to be enhanced by circumneutral pH conditions and time  
713 (Manceau et al., 2007b; Peacock and Sherman, 2007b; Peacock, 2009; Peña et al.,  
714 2010). Competitive sorption at TCS/DCS sites has been proposed also as a  
715 mechanism favoring structural incorporation in the octahedral layer (Lefkowitz and  
716 Elzinga, 2017). Some studies report the prevalence of Ni sorbed at TCS/DCS sites  
717 even for Ni/Mn ratios higher than 10%, however (Manceau et al., 2007a,b; Peña et al.,  
718 2010; Yin et al., 2012; Simanova et al., 2015; Grangeon et al., 2017). In such cases,  
719 sorption of Ni(II) as inner sphere complexes is favored by the presence of vacant  
720 octahedral sites and is thus negatively correlated with the Mn(III) content.  
721 Consistently, in our study structural incorporation of Ni in vacancy-free NiB  
722 precursors appears very limited. In NiB layered precursors, Ni is rather present both  
723 as Ni (hydr)oxide and as hydrated cations in birnessite/buserite interlayers, as shown  
724 by the increased hydration of Ni-rich precursors (Figs. 1, S1). The influence of Ni on  
725 phylломanganate hydration has been reported also for natural samples in which  
726 buserite (or 10 Å vernadite) abundance increases with increasing Ni content (Bodeř et  
727 al., 2007; Manceau et al., 2007a).

728 As discussed above, hydrated Ni(II) initially present in birnessite interlayers are  
729 readily stabilized during the reflux products by forming Ni(OH)<sub>2</sub> fragments (Fig. S5,  
730 Table S2), consistent with previous experiments with vacancy-free phylломanganate  
731 precursors (Feng et al., 2001). Polymerization of Ni(OH)<sub>2</sub> brucite-like octahedral  
732 layers in phylломanganate interlayers is consistent also with the natural predominance  
733 of asbolane in Ni-rich environments (NiO ~10-20 wt.%, – Chukhrov, 1982; Chukhrov  
734 et al., 1987; Roque-Rosell et al., 2010; Ploquin et al., 2018) . Although the Mn AOS  
735 of his synthetic samples (~4.0) was favorable for Ni sorption at layer vacancy sites,  
736 Manceau et al. (2007b) also reported the overwhelming presence of Ni as Ni(OH)<sub>2</sub> at  
737 high Ni content (Ni/Mn ratio of ~0.10 and ~0.18 at pH 4 and 7, respectively), despite  
738 experimental conditions supposed to prevent Ni(OH)<sub>2</sub> precipitation. The reflux

739 process appears to have little influence on the Ni (hydr)oxide formed together with  
740 the layered precursors, as evidenced by the stability of the corresponding maximum at  
741  $\sim 4.6 \text{ \AA}$  in the reflux products (Fig. 1b). Finally, part of Ni ( $\sim 20 \%$ ) is stabilized during  
742 the reflux process, as shown by the incomplete Ni release during acid treatment (Fig.  
743 5d). Stabilized Ni is possibly structurally incorporated in the reflux products. On the  
744 contrary, Atkins et al. (2016) reported no structural incorporation of Ni(II) during the  
745 formation of todorokite from a Ni-bearing birnessite precursor ( $\sim 1 \text{ wt\% Ni}$  in  
746 hexagonal birnessite). They reported also a decrease in the Ni content of their reflux  
747 products, whereas in our experimental conditions stability of Ni/(Mn+Ni) ratios  
748 (Table 1) indicates that all Ni initially present in layered precursors is retained during  
749 the reflux process.

750

## 751 5. CONCLUSION

752

753 The present study reports on the association of weakly bound Ni species  
754 (hydrated Ni, Ni (hydr)oxides) with phylломanganates, that possibly prevail in natural  
755 Ni-rich ( $>10\% \text{ NiO}$ ) manganates. Whatever their Ni content, prepared  
756 phylломanganates systematically exhibit an orthogonal layer symmetry, indicative of  
757 a high Mn(III) content and of a minimal number of octahedral layer vacancies. Ni is  
758 essentially present as charge-compensating hydrated Ni(II) and as Ni (hydr)oxide,  
759 with no detectable Ni(II) sorbed or structurally incorporated in the phylломanganate  
760 structure.

761 Despite their high content of Mn(III), a favorable condition for their conversion  
762 to tectomanganates, asbolane rather than tectomanganates forms during the reflux  
763 processing of these Ni-rich phylломanganates. Asbolane formation results from the  
764 kinetically favored polymerization of fragments of Ni(OH)<sub>2</sub> octahedral layers in  
765 birnessite interlayers. The structure of the initial manganate layers is essentially  
766 unaffected by this transformation.

767 Formation of asbolane, rather than todorokite, may actually be more frequent than

768 reported in the literature owing to the similar periodicities in both compounds. High  
769 values ( $> 1$ ) of the  $I_{4.8}/I_{9.6}$  intensity ratio may suggest the presence of this seldom  
770 described phyllomanganate. Acid nitric treatment, aiming at the dissolution of the  
771 island-like interlayer  $\text{Ni}(\text{OH})_2$  layer, represents an easy and unambiguous way to  
772 differentiate asbolane from todorokite which is unaffected by the treatment.

773 Both the presence of Co and Ni in synthesized phyllomanganates impede their  
774 transformation to tectomanganates when refluxed, the origin of this similar impact  
775 being different for the two metals, however. Oxidation of Co(II) sorbed on the  $\text{MnO}_2$   
776 layer and the subsequent structural incorporation of Co(III) in the octahedral layer  
777 releases the steric strains related to the presence of Jahn-Teller distorted Mn(III)  
778 octahedra, which is required for the phyllomanganate-to-tectomanganate conversion.  
779 In contrast, the presence of hydrated Ni(II) favors the polymerization of fragments of  
780  $\text{Ni}(\text{OH})_2$  octahedral layers, that in turn prevents the formation of tectomanganates  
781 despite the favorable presence of Mn(III) in the octahedral  $\text{MnO}_2$  layer. In both cases,  
782 “foreign” metal cations are retained in the refluxed products, with contrasting ratios of  
783 structural incorporation.

784

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1040

1041

**Table 1**

1042

Chemical composition and specific surface area of layered precursors and reflux products

1043

Samples	Mn wt. %	Ni wt. %	Ni/(Mn+Ni) mol%	Mg wt. %	(Mg+Ni)/Mn mol%	SSA (m <sup>2</sup> /g)
Bir	55.4(1.0)	--	--	--	--	--
NiB1	54.1(1.1)	0.39(1)	0.67	--	--	--
NiB5	53.7(1.0)	1.86(4)	3.18	--	--	--
NiB10	51.2(1.0)	3.89(7)	5.98	--	--	--
NiB15	47.3(1.7)	5.32(3)	8.71	--	--	--
NiB20	44.8(1.8)	7.14(7)	11.50	--	--	--
Tod	54.4(2.1)	--	--	3.95(8)	16.4	90.7
NiT1	54.3(8)	0.33(1)	0.57	3.51(12)	16.2	--
NiT5	54.2(6)	1.57(2)	2.57	3.12(2)	15.6	93.7
NiT10	52.7(6)	3.79(1)	5.94	2.61(2)	17.5	119.7
NiT15	52.4(1.2)	5.66(9)	8.41	2.59(6)	20.4	109.0
NiT20	49.2(9)	7.74(2)	11.38	2.11(2)	22.5	106.8

1044

1045 **Table 2.**

1046 Weight loss of layered precursors and reflux products over specific temperature  
1047 intervals

Temperature (°C)	Weight loss/%				
	Bir	NiB5	NiB10	NiB15	NiB20
<b>RT-100</b>	2.45	2.87	2.64	2.84	9.25
<b>100-170</b>	8.43	7.14	6.54	5.73	4.87
<b>170-400</b>	1.78	4.10	4.74	5.78	6.72
<b>Total</b>	17.38	19.20	18.85	19.15	25.31
	Tod	NiT5	NiT10	NiT15	NiT20
<b>RT-110</b>	2.75	2.93	3.39	3.90	4.12
<b>110-240</b>	8.05	7.68	7.43	6.57	6.12
<b>240-380</b>	5.47	5.32	5.04	5.34	5.97
<b>380-500</b>	2.69	2.33	2.08	1.93	1.79
<b>500-650</b>	4.50	4.35	4.06	3.75	3.61
<b>Total</b>	23.47	22.61	22.01	21.48	21.60

1048

1049 **Table 3.**

1050 Fractional and average valence states of Mn obtained for layered precursors and  
1051 reflux products from the Combo fit of XANES spectra and 1<sup>st</sup> derivatives. The  
1052 estimated error for Combo method is  $\pm 4\%$  (Manceau et al., 2012; Yin et al., 2015)

Sample	Mn <sup>2+</sup> at. %	Mn <sup>3+</sup> at. %	Mn <sup>4+</sup> at. %	Mn-AOS
Bir	3	26	72	3.69
NiB1	2	25	74	3.71
NiB5	2	30	67	3.64
NiB10	2	31	68	3.66
NiB15	0	31	69	3.69
NiB20	0	35	65	3.65
Tod	2	25	74	3.72
NiT1	1	35	65	3.64
NiT5	3	32	66	3.63
NiT10	2	35	64	3.62
NiT15	1	34	65	3.64
NiT20	0	33	67	3.67

1053

1054 **Figure Captions**

1055

1056 **Fig. 1.** XRD patterns of layered precursors (top) and reflux products (bottom). Colors  
1057 correspond to the Ni/(Ni+Mn) ratio of the samples. Green ticks at the bottom of upper  
1058 and lower figures indicate the reflections of triclinic birnessite (ICDD #43-1456) and  
1059 todorokite (ICDD #38-0475), respectively.

1060

1061 **Fig. 2.** XRD patterns collected for NiT20, NiT15, and Tod at (a) Mn K-edge (6.5 keV),  
1062 (b) Ni K-edge (8.3 eV), (c) 12.0 keV, and (d) 18.0 keV. Intensities were systematically  
1063 normalized to the most intense reflection. Colors as in **Fig. 1.** (e) XRD patterns  
1064 calculated for an asbolane containing 0.525 Ni(OH)<sub>2</sub> in MnO<sub>2</sub> interlayers. Contents of  
1065 Ni and Mn atoms are reduced by 20% at their respective absorption edges to simulate  
1066 the anomalous effect.

1067

1068 **Fig. 3.** FTIR spectra of layered precursors and reflux products (top and bottom,  
1069 respectively). Colors as in **Fig. 1.**

1070

1071 **Fig. 4.** HRTEM images of (a-b): Tod, (c-e): NiT1, (f-h): NiT5, (i-m): NiT1. Numbers  
1072 in (b) and (d) indicate the number of MnO<sub>6</sub> octahedra defining tunnel width. Arrows  
1073 in (i) and (l) indicate the direction of fringes bending.

1074

1075 **Fig. 5.** Proportion of metal cations released during the nitric acid treatment: (a-b): Mn,  
1076 (c-d): Ni, and (f): Mg. The amount of metal released is normalized to the sample  
1077 overall metal content (see text for details). (e): XRD patterns of samples treated with  
1078 nitric acid. Colors as in **Fig. 1** except for Bir in (e).

1079

1080 **Fig. 6.** Thermogravimetric analysis traces obtained for (a) layered precursors and (b)  
1081 reflux products, and their 1<sup>st</sup> derivatives (c and d, respectively). 1<sup>st</sup> derivative curves  
1082 were smoothed using a Savitzky-Golay filter and a 15 point window. Colors as in **Fig.**  
1083 **1.**

1084  
1085 **Fig. 7.** XPS O1s spectra obtained for layered precursors and reflux products (left and  
1086 right, respectively). The thick solid curve represents the best fit to the data (open red  
1087 squares). Peaks corresponding to O<sup>2-</sup>, OH<sup>-</sup>, and H<sub>2</sub>O species are shown in the order of  
1088 increasing binding energy as dashed purple, dotted-dashed green, and dotted blue  
1089 lines, respectively. The green dotted line is the difference curve.

1090  
1091 **Fig. 8.** Mn K-edge EXAFS spectra obtained for (a) layered precursors and (b) reflux  
1092 products. Colors as in **Fig. 1.**

1093  
1094 **Fig. 9.** Comparison of (a, c) Mn K-edge EXAFS and (b, d) of their FTs (showing both  
1095 real and imaginary parts) obtained for Ni-rich (Top: NiB15 and NiT15) and Ni-free  
1096 (bottom: Bir and Tod) layered precursors and corresponding reflux products.

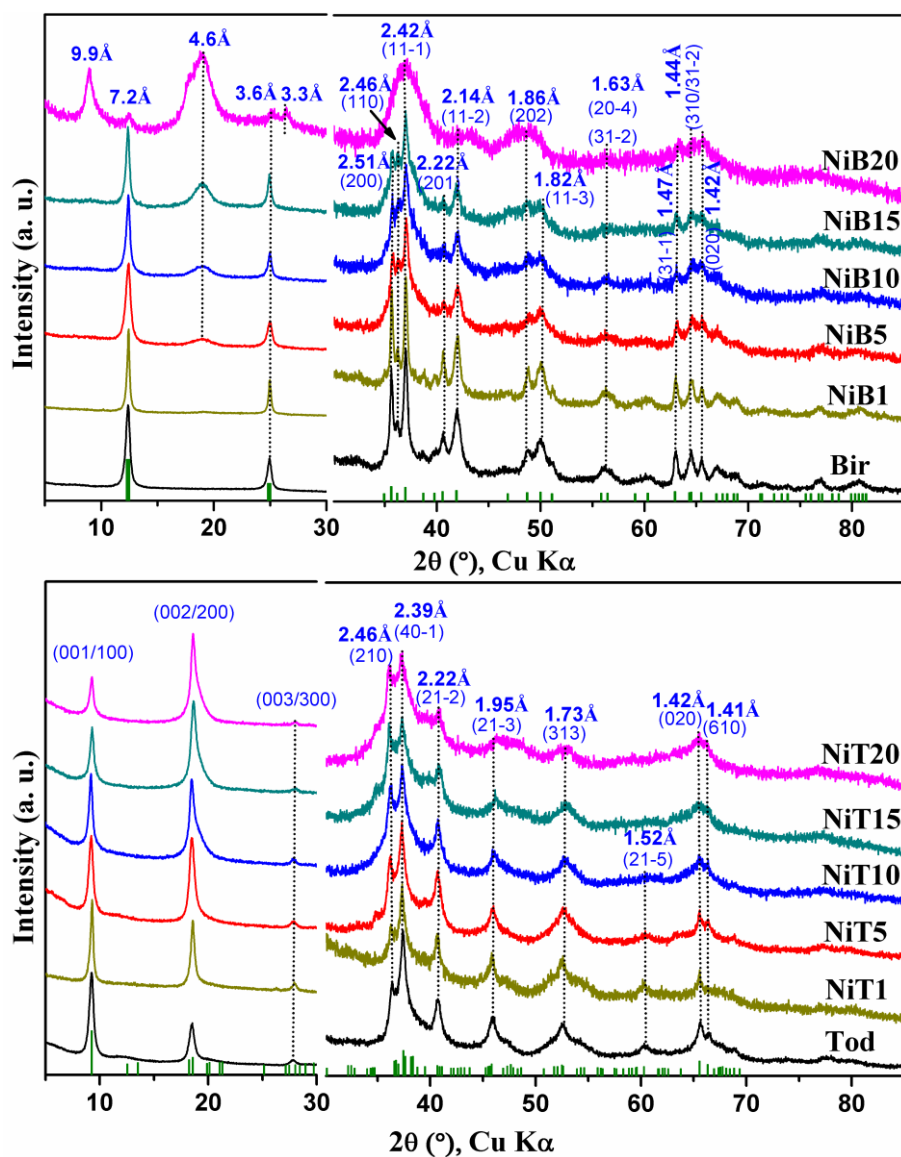
1097  
1098 **Fig. 10.** Comparison of Ni K-edge EXAFS spectra obtained for layered precursors  
1099 and reflux products with reference spectra. Sample colors as in **Fig. 1.**

1100  
1101 **Fig. 11.** Pair distribution functions obtained for Ni-rich (NiT10 and NiT20) and  
1102 Ni-free (Tod) reflux products (top). Middle: d-PDFs relative to Tod PDF data; bottom:  
1103 PDFs calculated for Ni (hydr)oxides (ICSD #76670 and #28101 for NiO and Ni(OH)<sub>2</sub>,  
1104 respectively). Intensities of experimental PDFs and of difference PDFs were  
1105 normalized to the 2.9-3.1 Å peak.

1106



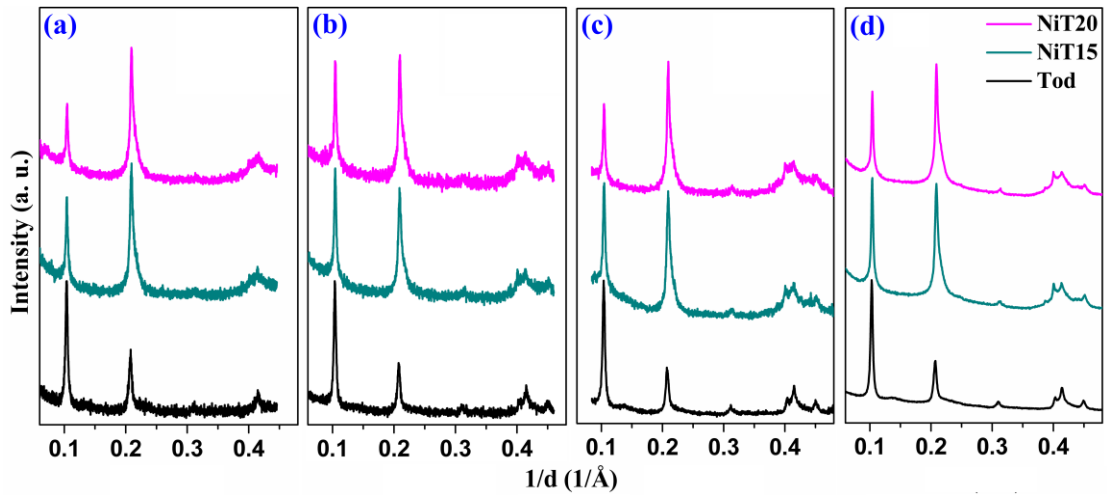
1107 **Fig. 12.** Intensity distributions calculated for ideal todorokite (ICDD #38-0475) with  
1108 (a) only Mg in todorokite tunnels and (b) with Mg and Ni present in a 1:1 ratio in  
1109 todorokite tunnels. In both cases, the model of Post et al. (2003) was modified  
1110 assuming the complete filling of the O8 site by H<sub>2</sub>O molecules (no cations) and the  
1111 presence of 1.0 cation (Mg or Mg+Ni) in the Mg site.



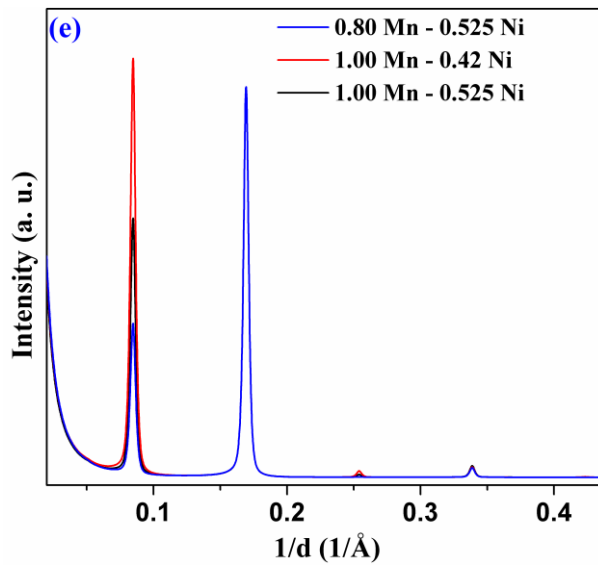
1

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6



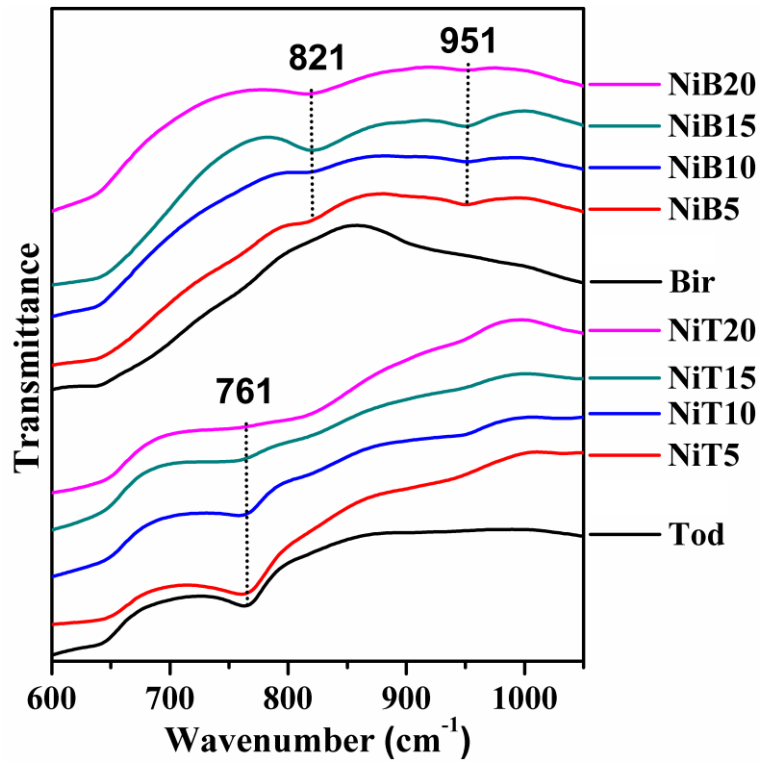
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9 **Fig. 2.** XRD patterns collected for NiT20, NiT15, and Tod at (a) Mn K-edge (6.5 keV),  
 10 (b) Ni K-edge (8.3 eV), (c) 12.0 keV, and (d) 18.0 keV. Intensities were systematically  
 11 normalized to the most intense reflection. Colors as in **Fig. 1**. (e) XRD patterns  
 12 calculated for an asbolane containing 0.525  $\text{Ni}(\text{OH})_2$  in  $\text{MnO}_2$  interlayers. Contents of  
 13 Ni and Mn atoms are reduced by 20% at their respective absorption edges to simulate  
 14 the anomalous effect.

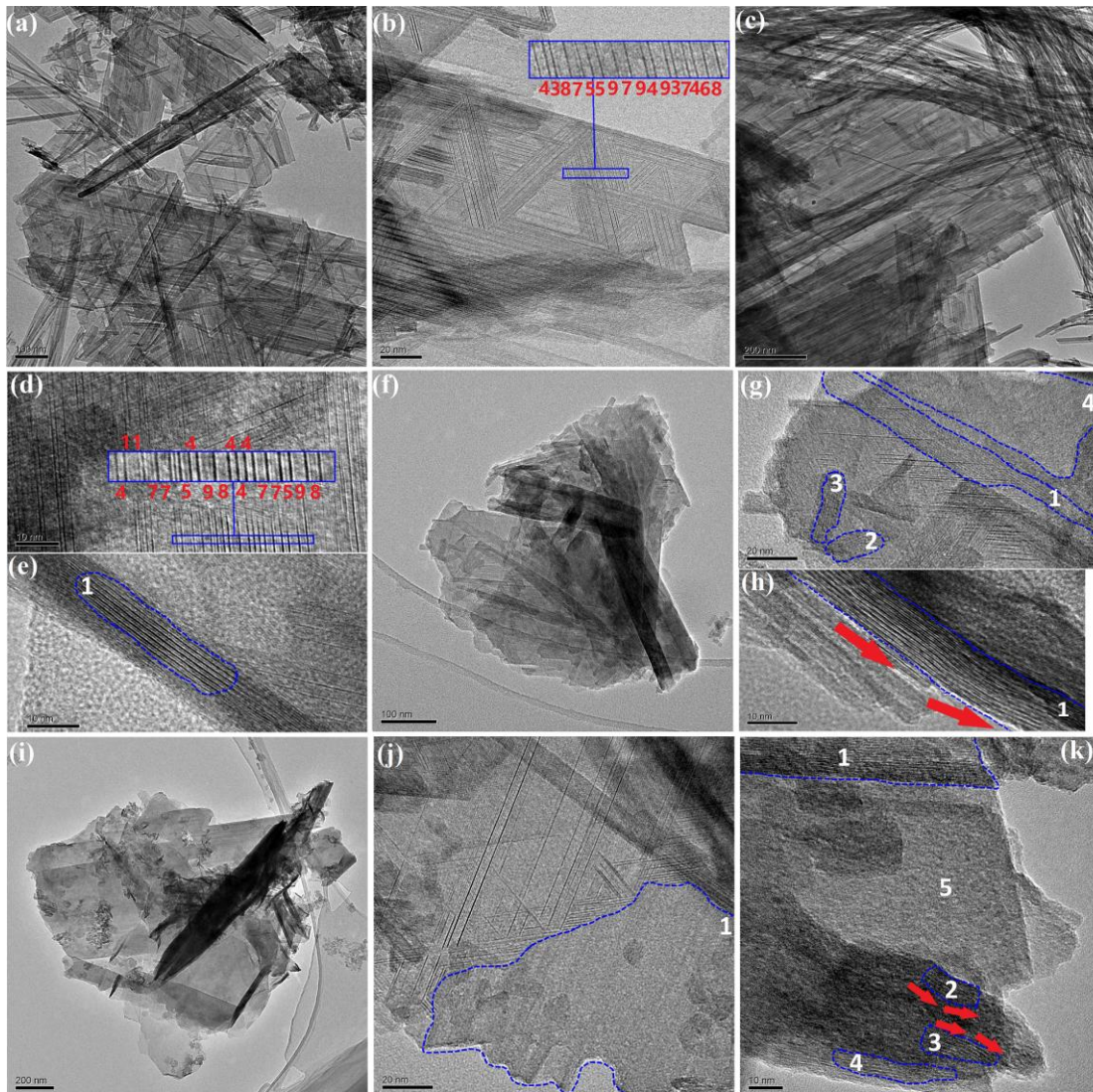
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16

17 **Fig. 3.** FTIR spectra of layered precursors and reflux products (top and bottom,  
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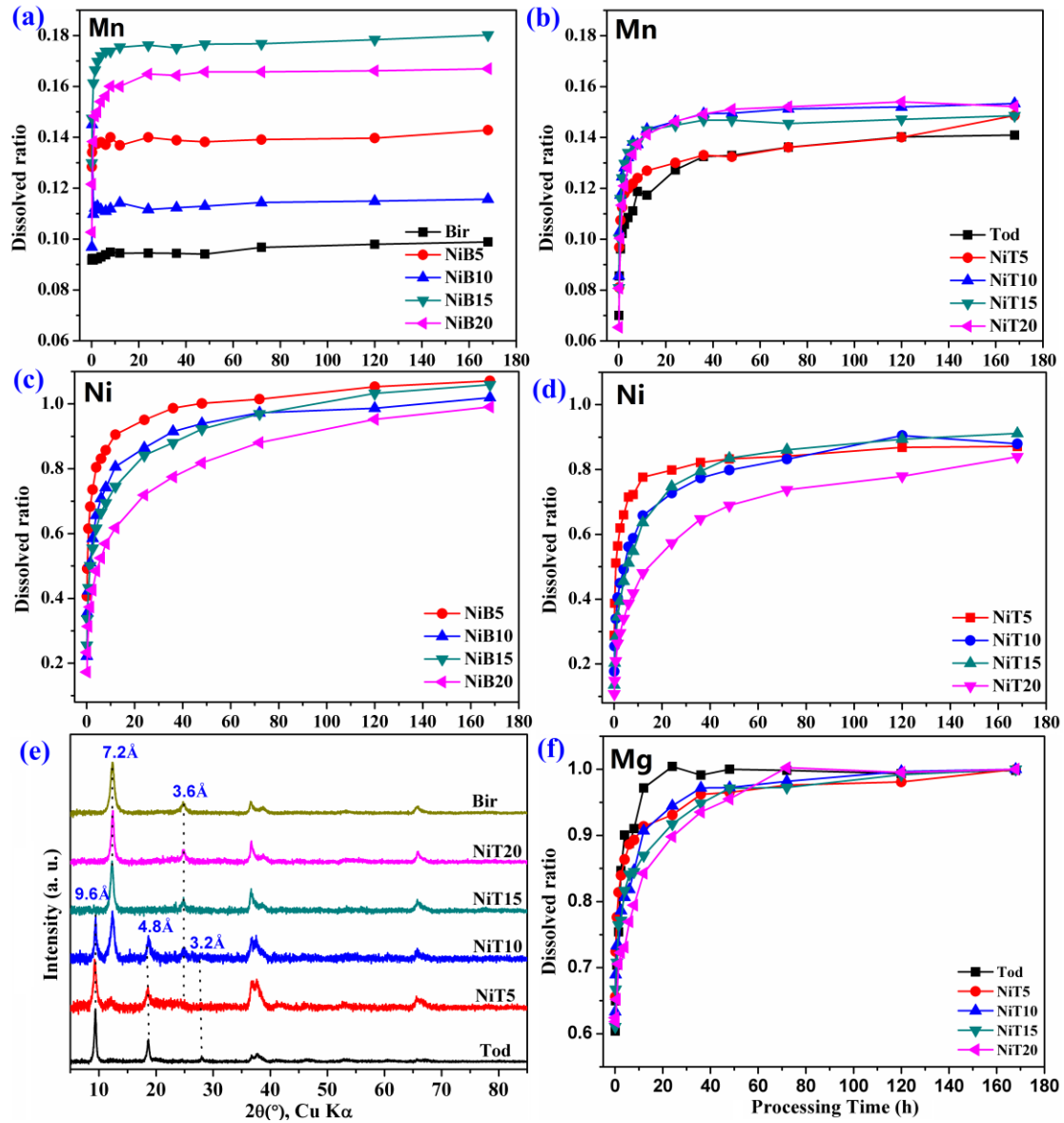
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20

21 **Fig. 4.** HRTEM images of (a-b): Tod, (c-e): NiT1, (f-h): NiT5, (i-m): NiT1. Numbers  
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 23 in (i) and (l) indicate the direction of fringes bending.

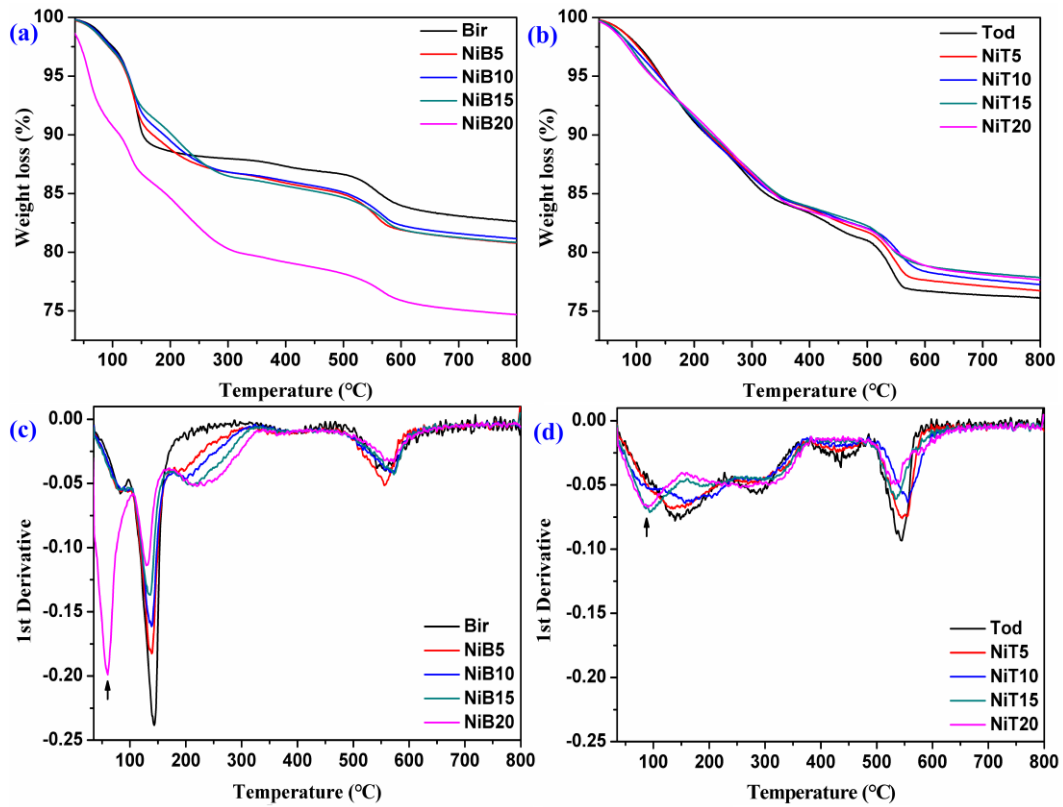
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25

26 **Fig. 5.** Proportion of metal cations released during the nitric acid treatment: (a-b): Mn,  
 27 (c-d): Ni, and (f): Mg. The amount of metal released is normalized to the sample  
 28 overall metal content (see text for details). (e): XRD patterns of samples treated with  
 29 nitric acid. Colors as in **Fig. 1** except for Bir in (e).

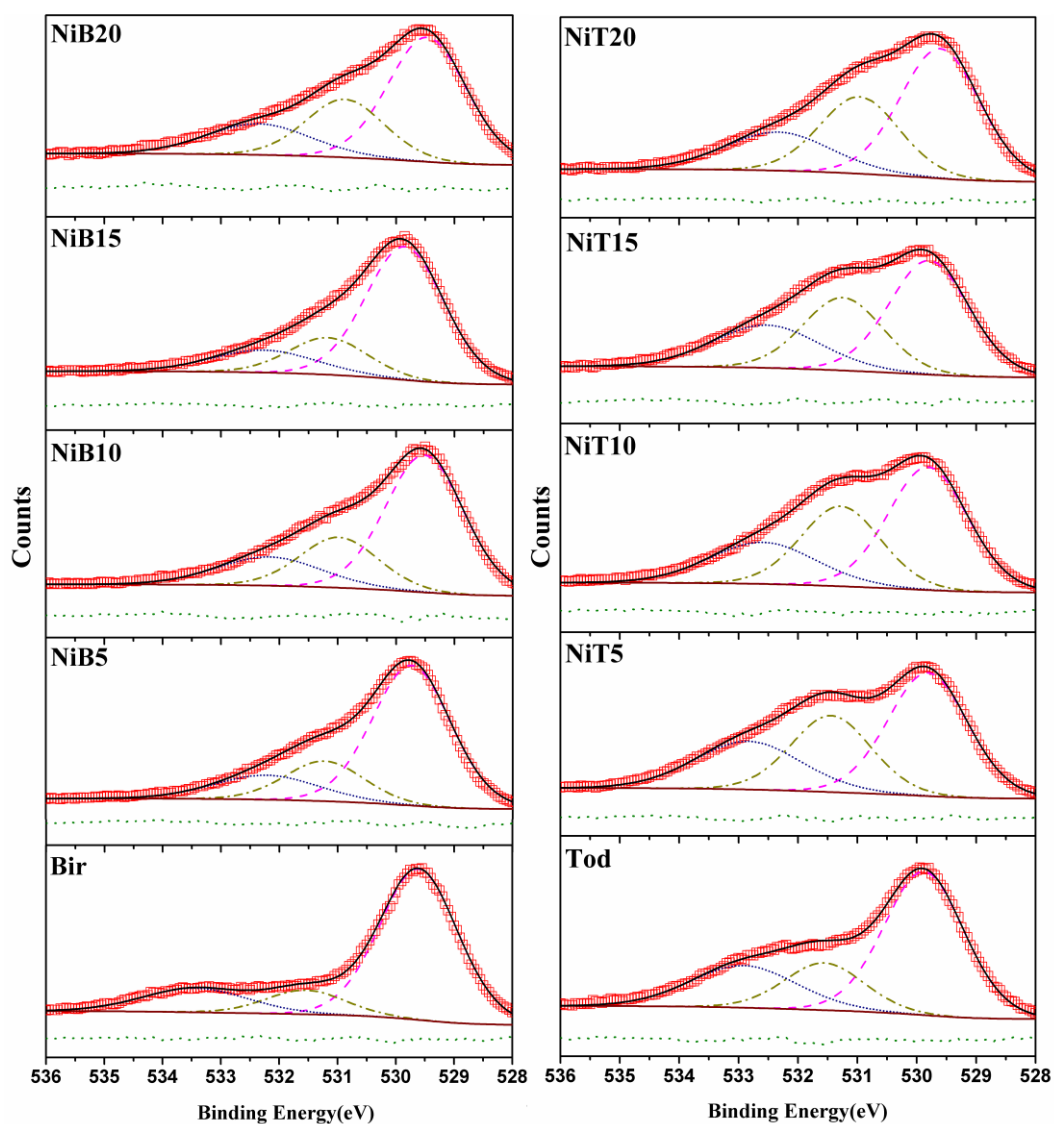
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31

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 33 reflux products, and their 1<sup>st</sup> derivatives (c and d, respectively). 1<sup>st</sup> derivative curves  
 34 were smoothed using a Savitzky-Golay filter and a 15 points window. Colors as in Fig.  
 35 1.

36

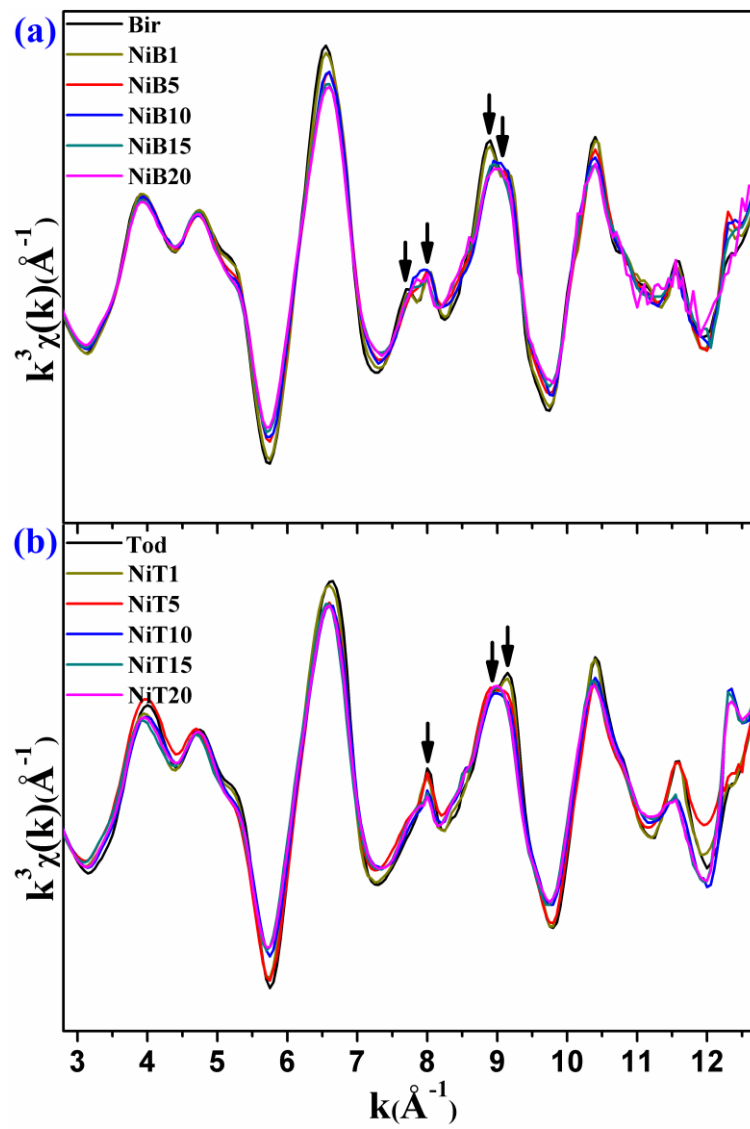


37

38 **Fig. 7.** XPS O1s spectra obtained for layered precursors and reflux products (left and  
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 40 squares). Peaks corresponding to  $O^{2-}$ ,  $OH^-$ , and  $H_2O$  species are shown in the order of  
 41 increasing binding energy as dashed purple, dotted-dashed green, and dotted blue  
 42 lines, respectively. The green dotted line is the difference curve.

43

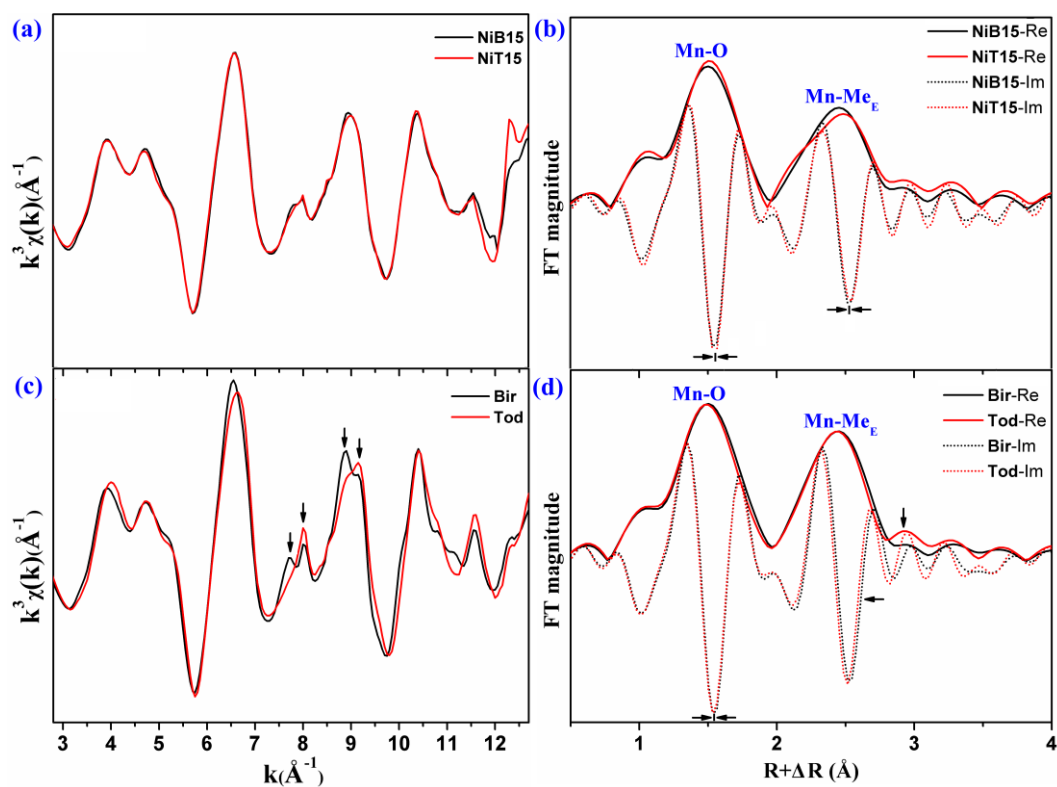




44

45 **Fig. 8.** Mn K-edge EXAFS spectra obtained for (a) layered precursors and (b) reflux  
 46 products. Colors as in **Fig. 1.**

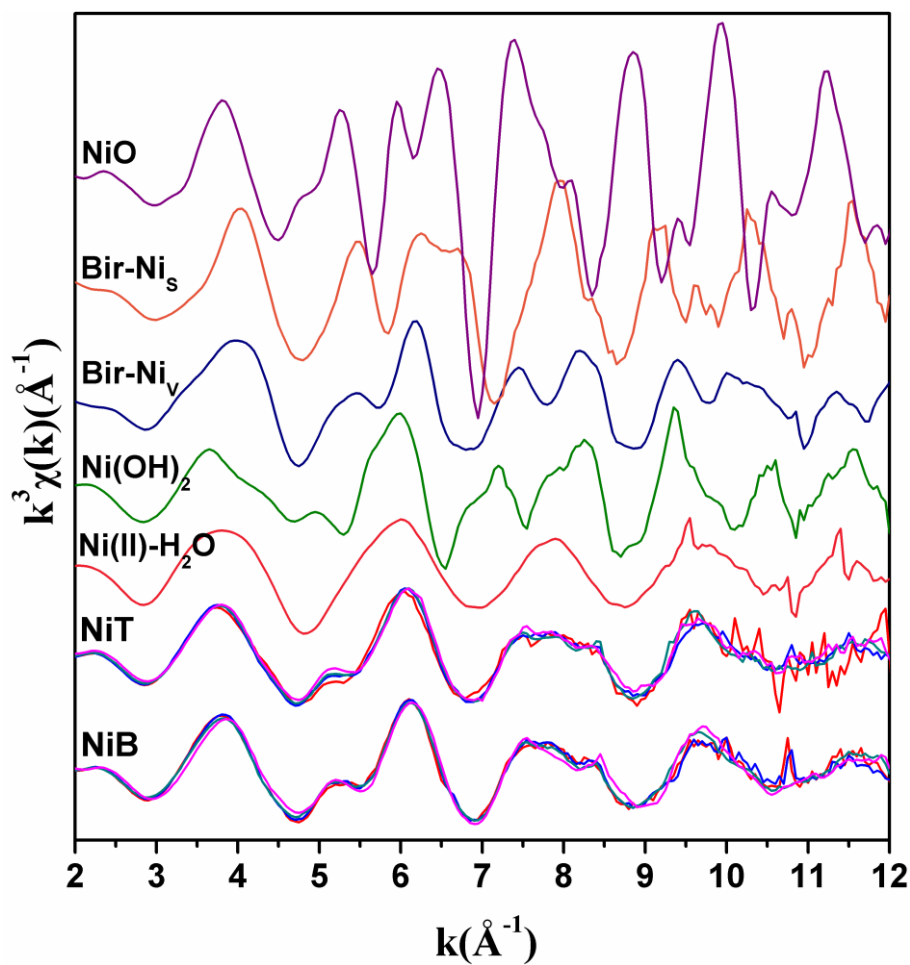
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48

49 **Fig. 9.** Comparison of (a, c) Mn K-edge EXAFS and (b, d) of their FTs (showing both  
 50 real and imaginary parts) obtained for Ni-rich (Top: NiB15 and NiT15) and Ni-free  
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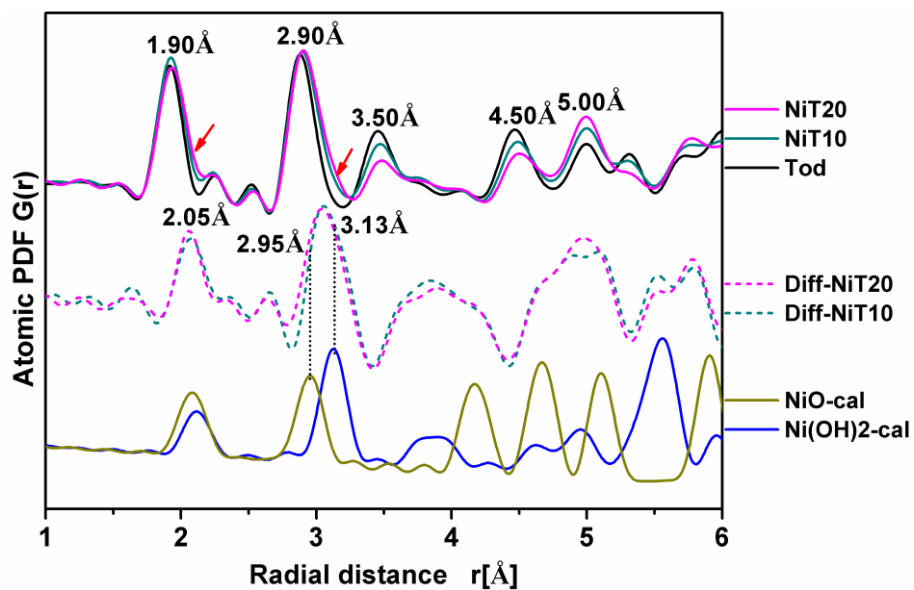
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53

54 **Fig. 10.** Comparison of Ni K-edge EXAFS spectra obtained for layered precursors  
 55 and reflux products with reference spectra. Sample colors as in **Fig. 1.**

56

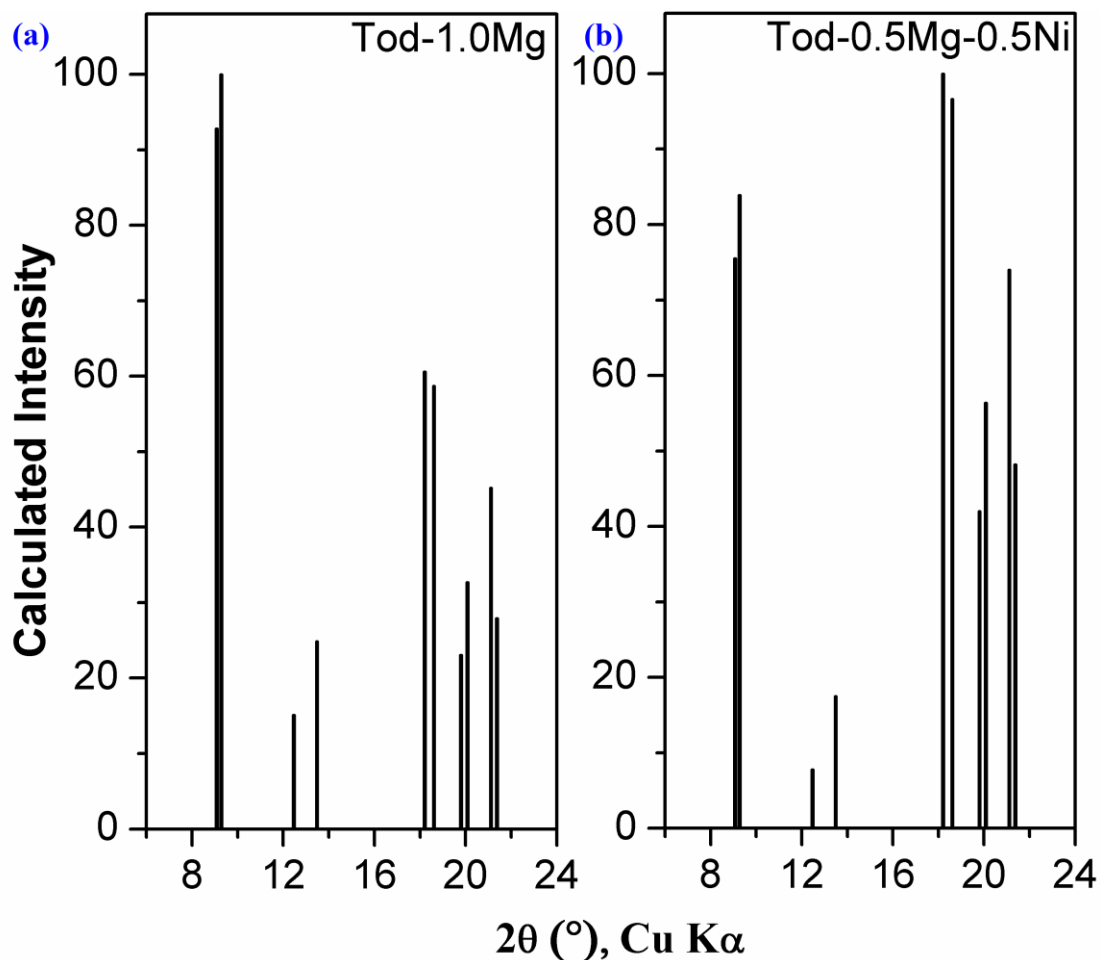


57

58 **Fig. 11.** Pair distribution functions obtained for Ni-rich (NiT10 and NiT20) and

59 Ni-free (Tod) reflux products (top). Middle: d-PDFs relative to Tod PDF data; bottom:  
60 PDFs calculated for Ni (hydr)oxides (ICSD #76670 and #28101 for NiO and Ni(OH)<sub>2</sub>,  
61 respectively). Intensities of experimental PDFs and of difference PDFs were  
62 normalized to the 2.9-3.1 Å peak.

63



64

65 **Fig. 12.** Intensity distributions calculated for ideal todorokite (ICDD #38-0475) with  
66 (a) only Mg in todorokite tunnels and (b) with Mg and Ni present in a 1:1 ratio in  
67 todorokite tunnels. In both cases, the model of Post et al.(2003) was modified  
68 assuming the complete filling of the O8 site by H<sub>2</sub>O molecules (no cations) and the  
69 presence of 1.0 cation (Mg or Mg+Ni) in the Mg site.

70

**Transformation of Ni-containing birnessite to  
tectomanganate:  
Influence and fate of weakly bound Ni(II) species**

**Supplementary material**

**Geochimica Et Cosmochimica Acta**

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Table S1

Relative proportions of O species in the near-surface of layered precursors and reflux products calculated by fitting XPS O1s data

Sample	O1s (BE)	State	BE(eV)	FWHM(eV)	At. %
<b>Bir</b>	529.35	O <sup>2-</sup>	529.6	1.59	72.9
		OH <sup>-</sup>	531.5	1.59	11.6
		H <sub>2</sub> O	533.4	2.12	15.5
<b>NiB5</b>	529.72	O <sup>2-</sup>	529.7	1.59	65.9
		OH <sup>-</sup>	531.2	1.59	18.8
		H <sub>2</sub> O	532.2	2.12	15.3
<b>NiB10</b>	529.72	O <sup>2-</sup>	529.5	1.59	60.8
		OH <sup>-</sup>	531.0	1.59	22.3
		H <sub>2</sub> O	532.2	2.12	16.9
<b>NiB15</b>	530.06	O <sup>2-</sup>	529.8	1.59	66.6
		OH <sup>-</sup>	531.2	1.59	18.5
		H <sub>2</sub> O	532.3	2.12	15.0
<b>NiB20</b>	530.21	O <sup>2-</sup>	529.5	1.59	55.6
		OH <sup>-</sup>	530.9	1.59	25.9
		H <sub>2</sub> O	532.4	2.12	18.5
<b>Tod</b>	529.89	O <sup>2-</sup>	529.9	1.59	58.1
		OH <sup>-</sup>	531.6	1.59	20.9
		H <sub>2</sub> O	533.0	2.12	21.0
<b>NiT5</b>	529.84	O <sup>2-</sup>	529.8	1.59	46.7
		OH <sup>-</sup>	531.4	1.59	29.0
		H <sub>2</sub> O	532.8	2.12	24.4
<b>NiT10</b>	530.08	O <sup>2-</sup>	529.8	1.59	47.6
		OH <sup>-</sup>	531.3	1.59	31.0
		H <sub>2</sub> O	532.6	2.12	21.5
<b>NiT15</b>	530.15	O <sup>2-</sup>	529.8	1.59	46.8

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		OH <sup>-</sup>	531.2	1.59	28.9
		H <sub>2</sub> O	532.5	2.12	23.4
		O <sup>2-</sup>	529.6	1.59	50.3
<b>NiT20</b>	529.59	OH <sup>-</sup>	531.0	1.59	29.8
		H <sub>2</sub> O	532.3	2.12	19.9

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Table S2

Relative proportions of NiB and Ni(OH)<sub>2</sub> species obtained from linear combination fitting of NiT K-edge EXAFS spectra using Ni(OH)<sub>2</sub> and corresponding NiB samples as standard references (Fig. S5).

Samples	NiBn /%	Ni(OH) <sub>2</sub> /%	Error /%
NiT5	82.0	18.0	2.1
NiT10	86.5	13.5	1.0
NiT15	87.1	12.9	0.9
NiT20	85.3	14.7	1.1



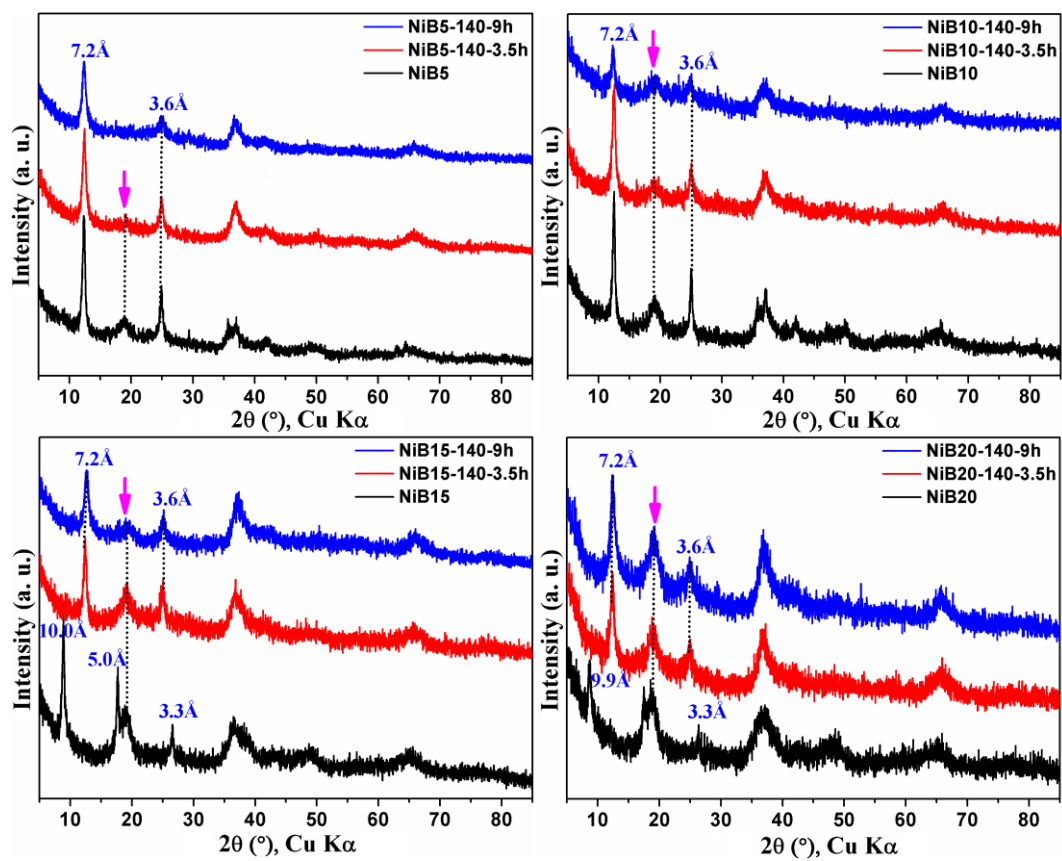


Fig. S1. XRD patterns of fresh layered precursors after heating to 140 °C for different durations

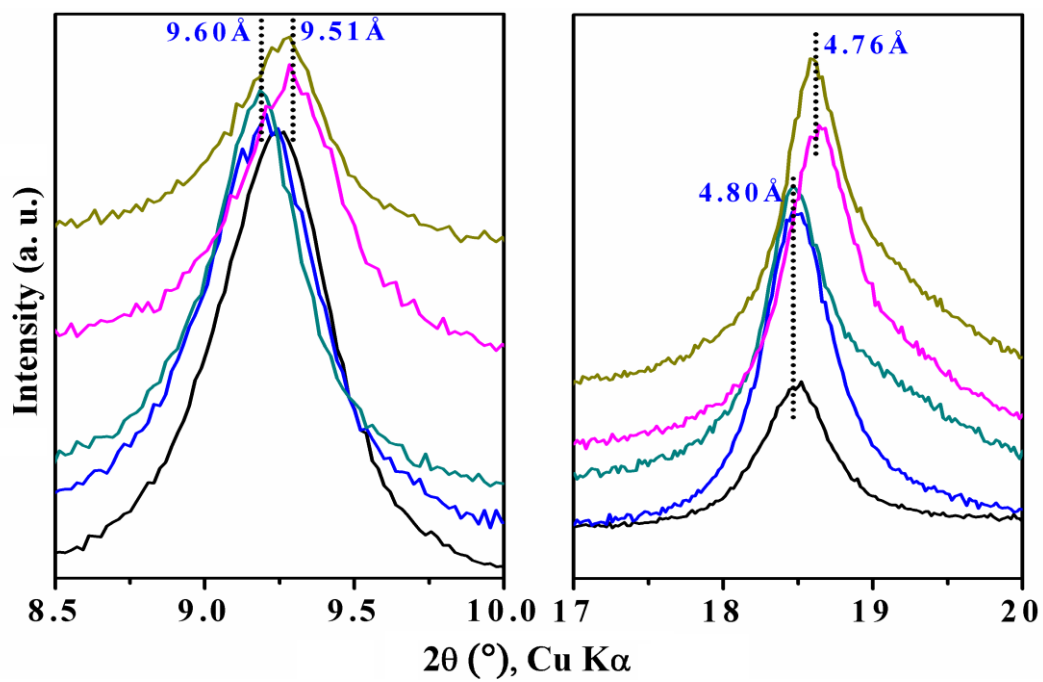


Fig. S2. XRD patterns of reflux products (zoom on low-angle regions). Colors as in Fig. 1.

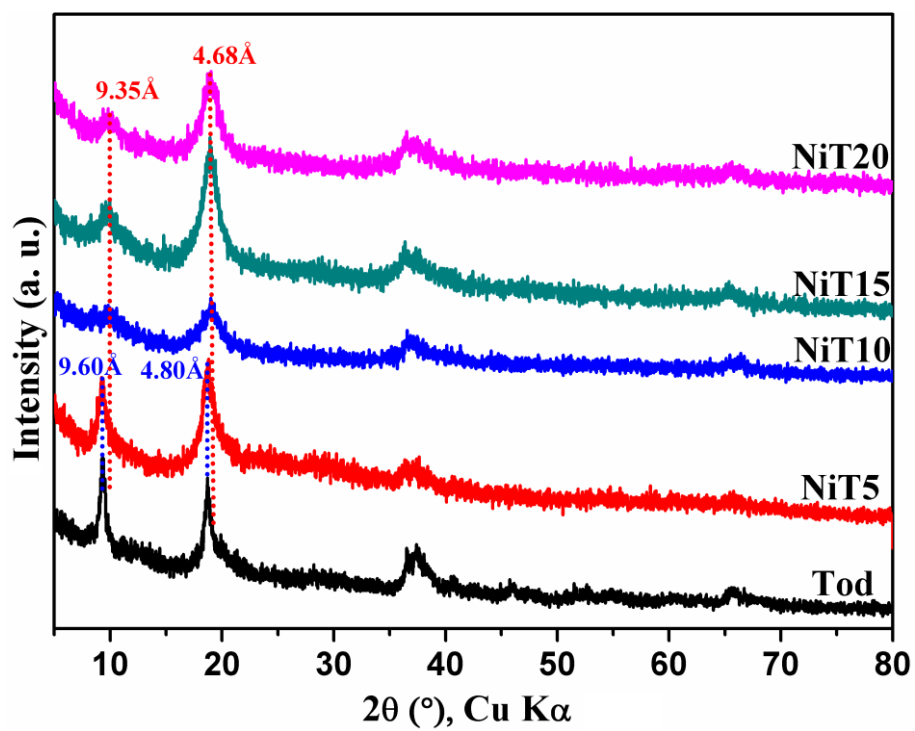


Fig. S3. XRD patterns of reflux products after heating to 140 °C for 12 hours. Colors as in Fig. 1.

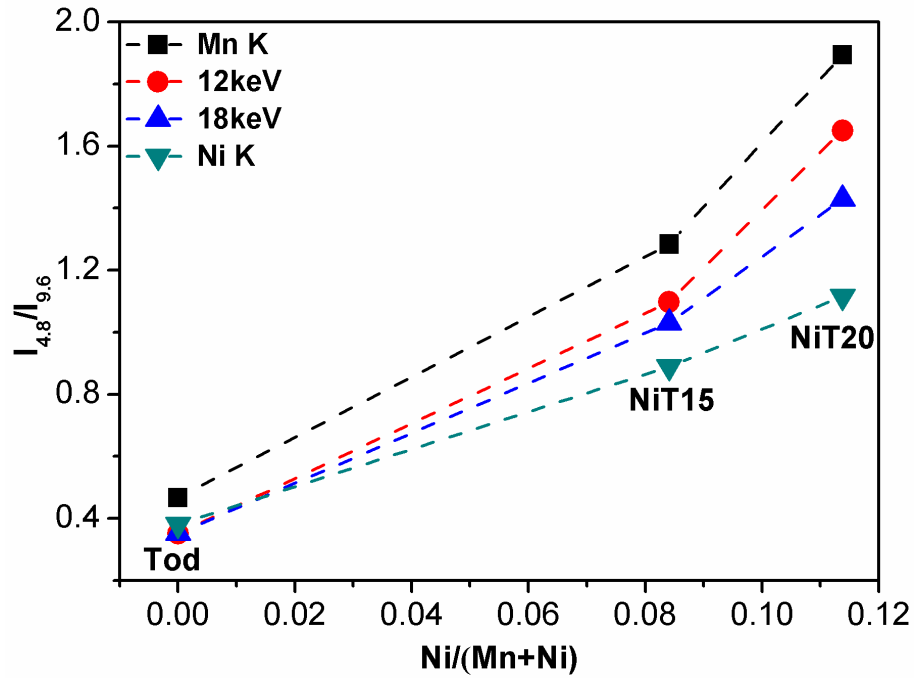


Fig. S4. Intensity ratio of the intensity of 4.8 Å and 9.6 Å peaks ( $I_{4.8}/I_{9.6}$ ) in reflux products as a function of Ni content and X-ray energy (see Fig. 2 for the data).

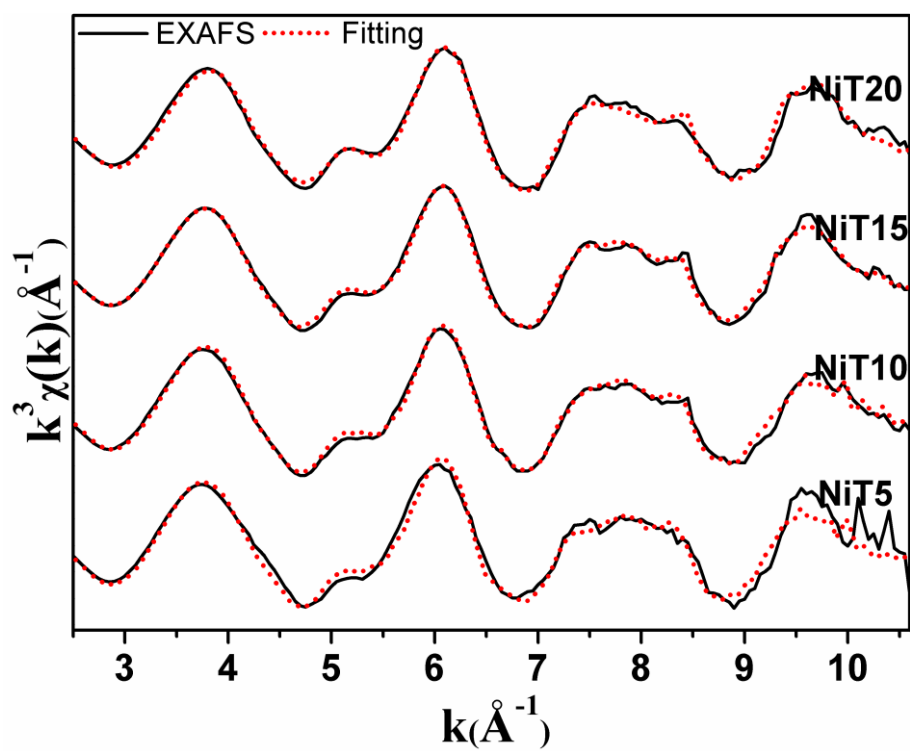


Fig. S5. Linear combination fitting of NiT K-edge EXAFS spectra using  $\text{Ni}(\text{OH})_2$  and corresponding NiB samples as standard references.

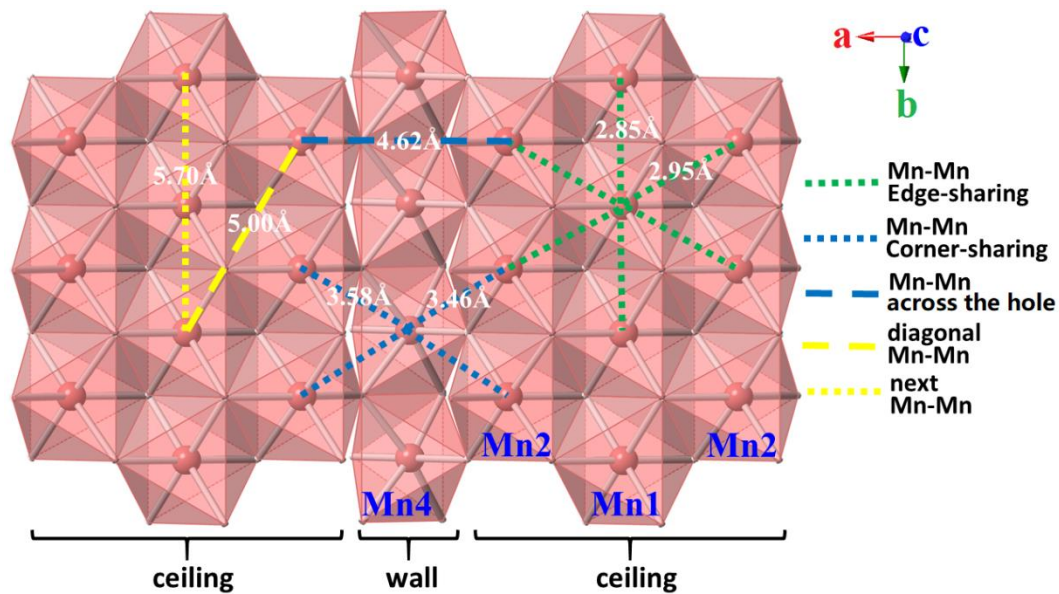


Fig. S6. Interatomic distances calculated for a variety of Mn-Mn pairs and ideal todorokite structure (ICDD #38-0475). The Mn4-Mn4 distance (4.48 Å) which is not reported in the figure is similar to the Mn2-Mn2 distance. Figure modified from Fig. 1(b) in Wu et al. (2019).

## REFERENCES

- Wu Z., Peacock C.L., Lanson B., Yin H., Zheng L., Chen Z., Tan W., Qiu G., Liu F. and Feng X. (2019) Transformation of Co-containing birnessite to todorokite: Effect of Co on the transformation and implications for Co mobility. *Geochim. Cosmochim. Acta* **246**, 21-40.