

Transformation of Ni-containing birnessite to tectomanganate: Influence and fate of weakly bound Ni(II) species

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Zhongkuan Wu, Bruno Lanson, Xionghan Feng, Hui Yin, Zhangjie Qin, et al.. Transformation of Nicontaining birnessite to tectomanganate: Influence and fate of weakly bound Ni(II) species. Geochimica et Cosmochimica Acta, 2020, 271, pp.96-115. 10.1016/j.gca.2019.12.023 . insu-02984573

HAL Id: insu-02984573 https://insu.hal.science/insu-02984573

Submitted on 25 Nov 2020

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1	Transformation of Ni-containing birnessite to
2	tectomanganate:
3	Influence and fate of weakly bound Ni(II) species
4	
5	Zhongkuan Wu ^{a,b} , Bruno Lanson ^{b,*} , Xionghan Feng ^a , Hui Yin ^a , Zhangjie Qin ^a ,
6Xiac	oming Wang, ^a Wenfeng Tan ^a , Zhongjun Chen ^c , Wen Wen ^d , Fan Liu ^{a,*}
7	
8	^a Key Laboratory of Arable Land Conservation (Middle and Lower Reaches of
9	Yangtse River) Ministry of Agriculture, College of Resources and Environment,
10	Huazhong Agricultural University, Wuhan 430070, China
11	^b Univ. Grenoble Alpes, Univ. Savoie Mont Blanc, CNRS, IRD, IFSTTAR,
12	ISTerre, F-38000 Grenoble, France
13	^c Beijing Synchrotron Radiation Facility, Institute of High Energy Physics,
14	Chinese Academy of Sciences, Beijing 100039, China
15	^d Shanghai Synchrotron Radiation Facility, Shanghai Advanced Research
16	Institute Chinese Academy of Sciences, Shanghai 201204, China
17	
18	* Corresponding author: <u>bruno.lanson@univ-grenoble-alpes.fr;</u>
19	liufan@mail.hzau.edu.cn
20	

21 Abstract

22 The geochemical behavior of nickel, an essential trace metal element, strongly 23 depends on its interactions with Mn oxides. Interactions between the phyllomanganate 24 birnessite and sorbed or structurally incorporated Ni have been extensively documented together with the fate of Ni along the transformation of these layered 25 species to tunnel Mn oxides (tectomanganates). By contrast, interactions of 26 27 phyllomanganates with weakly bound Ni species (hydrated Ni, Ni (hydr)oxides), that 28 possibly prevail in natural Ni-rich (>10% NiO) manganates received little attention 29 and the influence of these Ni species on the phyllomanganate-to-tectomanganate 30 transformation remains essentially unknown. Within this framework, a set of phyllomanganate precursors with contrasting contents of Ni were prepared and 31 32 subjected to a reflux process mimicking the natural phyllomanganate-to-tectomanganate conversion. Layered precursors and reflux 33 products were characterized with a combination of diffractometric, spectroscopic, 34 thermal, and chemical methods. Ni is essentially present as hydrated Ni(II) and Ni(II) 35 36 (hydr)oxides in layered precursors with no detectable Ni sorbed at layer vacancy sites 37 or structurally incorporated. Despite the high content ($\sim 1/3$) of Jahn-Teller distorted Mn(III) octahedra in these layered precursors, which is known to be favorable to their 38 39 conversion to tectomanganates, polymerization of Ni(OH)₂ in phyllomanganate 40 interlayers is kinetically favored during reflux process. Asbolane, a phyllomanganate 41 with an incomplete - island-like - octahedral layer of metal (hydr)oxides, is thus formed rather than todorokite, a common tectomanganate with a uniform 3×3 tunnel 42 43 structure. A nitric acid treatment, aiming at the dissolution of the island-like interlayer Ni(OH)₂ layer, allows an easy and unambiguous differentiation between asbolane and 44 45 todorokite, which is unaffected by the treatment. Both compounds exhibit indeed similar periodicities and can be confused when using X-ray diffraction, despite 46 contrasting intensity ratios. Ni(OH)₂ polymerization hampers the formation of 47 48 tectomanganates and likely contributes to the prevalence of phyllomanganates over

- tectomanganates in natural environments. Most Ni is retained during the reflux process, part of Ni (~20%) being likely structurally incorporated in the reaction products, thus enhancing the sequestration of Ni in Mn oxides.
- 52 *Keywords:* nickel; buserite; asbolane; todorokite; Mn oxides

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1. INTRODUCTION

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Phyllomanganates and tectomanganates are the two mineral families accounting 56 57 for most of Mn oxides found in terrestrial and aquatic environments (Post, 1999; Peacock and Sherman, 2007a). In the former, edge-sharing MnO_6 octahedra form 58 layers whose symmetry depends on the abundance and distribution of Mn(III) 59 octahedra, whereas in the latter, chains of edge-sharing Mn(IV)O₆ octahedra form the 60 walls or floor/ceiling of tunnels. Birnessite is, together with its disordered structural 61 analogue vernadite, the most common phyllomanganate and consists of a MnO₂ layer, 62 63 whose layer charge deficit is compensated for by the presence of cations sorbed at layer vacancy sites or present as interlayer hydrated species (7.0-7.2 Å layer-to-layer 64 65 distance). Heterovalent Mn cations and/or foreign transition metal cations can be hosted in both families, either structurally substituted for Mn(IV) or as charge 66 compensating cations located in phyllomanganate interlayers or in tectomanganate 67 tunnels (Post, 1999). The occurrence of natural phyllomanganates as micro- to 68 69 nano-sized cryptocrystalline minerals usually enhances their surface reactivity that 70 originates from their high negative layer charge and from the coexistence of 71 heterovalent Mn cations in these structures. As the result, natural phyllomanganates 72 are highly reactive with respect to both organic (Remucal and Ginder-Vogel, 2014) 73 and inorganic compounds. Specifically, natural phyllomanganates are readily 74 associated with transition metals such as Co, Cu, Zn, or Ni (Chukhrov, 1982; Manceau et al., 1987, 1992a,b, 2002, 2007a,b, 2014; Marcus et al., 2004; Bodeï et al., 75 76 2007; Peacock and Sherman, 2007a; Lanson et al., 2008; Spinks et al., 2017; Zawadzki et al., 2018), rare earth (Kasten et al., 1998; Ohta and Kawabe, 2001; 77 Ohnuki et al., 2015; Zawadzki et al., 2018), and radioactive elements (Duff et al., 78 1999; Post, 1999). 79

In an effort to disentangle the mobility, (bio)availability, and fate of these elements in different geological settings, structural interactions of phyllomanganates with such foreign elements, and more especially with transition metals, have thus

83 attracted the attention of the scientific community over the last few decades (Manceau 84 et al., 1997, 2002, 2014; Ohta and Kawabe, 2001; Lanson et al., 2002a; Marcus et al., 2004; Toner et al., 2006; Peacock and Sherman, 2007a,b; Peacock, 2009; Peña et al., 85 2010; Roque-Rosell et al., 2010; Sherman and Peacock, 2010; Yin et al., 2011a,b; 86 Kwon et al., 2013; Yu et al., 2013; Burlet and Vanbrabant, 2015; Simanova et al., 87 2015). In most studies, foreign elements were structurally substituted for Mn(IV) in 88 the MnO₆ octahedra or sorbed as inner-sphere complexes either at vacant layer sites 89 90 and/or at particle edges (Manceau et al., 1997; Yin et al., 2011a, 2015; Yu et al., 2013; Ohnuki et al., 2015; Simanova and Pena, 2015; Qin et al., 2018). Similar to other 91 92 transition metals, presence of Ni(II) forming inner-sphere complexes species at vacancy or edge sites and/or structurally incorporated in phyllomanganate layers has 93 94 been extensively documented (Manceau et al., 2002, 2007a,b; Peacock and Sherman, 2007a,b; Grangeon et al., 2008; Peña et al., 2010; Yin et al., 2012, 2014; Kwon et al., 95 2013; Simanova et al., 2015). Under low pH conditions, Ni(II) occurs mainly as 96 triple- or double-corner sharing (TCS and DCS, respectively) inner-sphere complex 97 98 above/below vacancy or at edge sites of the phyllomanganate layers (Peacock and Sherman, 2007b; Grangeon et al., 2008; Peacock, 2009; Peña et al., 2010; Simanova 99 100 et al., 2015). Migration from TCS/DCS sites to octahedral layer sites is favored by 101 increasing contact time and increasing pH (Peacock, 2009), this structural 102 incorporation within the octahedral MnO₂ layer being reversible (Peacock, 2009; Peña 103 et al., 2010) and essentially restricted to low Ni contents (Manceau et al., 2002; Bodeï et al., 2007; Peacock and Sherman, 2007a,b; Kwon et al., 2013; Simanova et al., 2015; 104 Atkins et al., 2016), consistent with natural occurrences (Manceau et al., 2002, 105 106 2007a,b; Bodeï et al., 2007; Peacock and Sherman, 2007a).

In addition, hydrated Ni(II) or Ni forming outer-sphere complexes (Turner and Buseck, 1981; Peacock and Sherman, 2007a) and Ni(II) (hydr)oxides (Chukhrov et al., 1987; Manceau et al., 1987, 1992a,b) may also be associated with Mn oxides. Despite their widespread natural occurrences for high Ni contents (NiO ~10-20 wt.%, – Llorca, 1988; Roque-Rosell et al., 2010; Ploquin et al., 2018), these weaker associations have received little attention compared to structurally incorporated Ni or to Ni inner-sphere

complexes. The presence of hydrated Ni(II) species in phyllomanganate interlayers 113 114 leads to the presence of two planes of interlayer water molecules, thus increasing the layer-to-layer distance from ~7.2 Å (birnessite or vernadite) to ~9.6-10.2 Å (buserite – 115 Burns et al., 1983; Chukhrov et al., 1987; Golden et al., 1987; Manceau et al., 1997, 116 117 2014; Bodeï et al., 2007; Wegorzewski et al., 2015). Such increased hydration of Ni-rich phyllomanganates has for example been reported by Manceau et al. (2007b). 118 The presence of hydrated Ni(II) in phyllomanganate interlayers has been described 119 120 also as favorable to the transformation of phyllomanganate to tectomanganates upon hydrothermal treatment (Ching et al., 1999; Luo et al., 1999; Onda et al., 2007), 121 122 similar to Mg(II) (Golden et al., 1986, 1987). In addition, Ni(II) (hydr)oxides may polymerize in phyllomanganate interlayers either as a complete or as an incomplete 123 brucite-like octahedral layer, leading to litiophorite- or asbolane-like species, 124 respectively, both having a layer-to-layer distance of 9.4-9.8 Å (Chukhrov et al., 1987; 125Manceau et al., 1987, 1992a,b; Feng et al., 2001). 126

The present article thus focuses on the association of these weakly bound Ni(II) 127 128 species with phyllomanganates. Synthetic samples with contrasting Ni contents were prepared and characterized by X-ray diffraction, Fourier transform infrared 129 spectroscopy, thermogravimetric analysis, X-ray photoelectron spectroscopy, 130 extended X-ray absorption fine structure spectroscopy (Mn and Ni K-edges). In 131 132addition, phyllomanganate samples were hydrothermally treated and reaction products characterized investigate effect of Ni 133to the on the phyllomanganate-to-tectomanganate transformation (Golden et al., 1986; Shen et al., 134 1993; Feng et al., 2004, 2010; Atkins et al., 2014; Grangeon et al., 2014, 2015; Zhao 135 et al., 2015) and the fate of weakly associated Ni(II) species during this 136 transformation. The negative impact of Ni, when present at high content, on this 137 transformation is contrasted with that of Co. Buserite-, litiophorite-, and asbolane-like 138phyllomanganates all have interplanar periodicities similar to those of todorokite, a 139 common tectomanganate with a uniform 3×3 tunnel structure (Post and Bish, 1988), 140 141 however, leading to their frequent confusion (Burns et al., 1983). Special care was thus paid to allow differentiating these species using nitric acid treatment (Wu et al., 142

143 2019), Fourier transform infrared spectroscopy and anomalous X-ray diffraction. An
144 optimum procedure allowing this differentiation is proposed.

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2. MATERIALS AND METHODS

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148 **2.1. Synthesis of layered precursors and reflux products**

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Both initial layered precursors and reflux products described in the present study 150 were prepared as in our previous work on their Co analogues (Wu et al., 2019). 151 Briefly, layered precursors were prepared after Feng et al. (2004) and Song et al. 152(2010) by adding a concentrated NaOH solution into a Ni(II) + Mn(II) solution, 153 154prepared from their chloride salts. The resulting solution was then mixed and bubbled 155with oxygen gas for 5 h. Resulting black precipitates were washed and one half was dried for subsequent analyses. Initial Ni / (Ni + Mn) ratios were 0.00, 0.01, 0.05, 0.10, 156 0.15, and 0.20. Accordingly, initial layered samples were named Bir, NiB1, NiB5, 157 158NiB10, NiB15 and NiB20, and those Ni-containing samples are collectively referred 159to as NiB. The other half of the precipitates was used to prepare todorokite through a reflux process following the protocol of Feng et al. (2004). Wet precipitates were thus 160 exchanged with Mg^{2+} , rinsed, re-suspended in deionized water, and finally refluxed at 161 100 °C for 24 h. Reflux products were finally filtered, washed, and dried. 162 Transformation products of Bir, NiB1, NiB5, NiB10, NiB15 and NiB20 are named 163 Tod, NiT1, NiT5, NiT10, NiT15 and NiT20, respectively; they are collectively 164 165 referred to as NiT except for Tod. Both layered precursors and reflux products were 166 dried at 60 °C for three days and ground into fine powders.

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168 **2.2. Characterization of layered precursors and reflux products**

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170 The elemental composition of all samples was determined in triplicate using 171 atomic absorption spectrometry (Varian AAS 240FS) from 0.1000 g of sample

dissolved in 25 mL solution of 0.25 mol L^{-1} NH₂OH·HCl and 5 mL of 1 mol L^{-1} 172H₂SO₄. In addition, the specific surface area (SSA) of the reflux products was 173determined with the Brunauer-Emmett-Teller (BET) method and a standard 174 adsorption analyzer (Quantachrome Autosorb-1, JEDL-6390/LV). Finally, a 175non-reducing nitric acid treatment (Wu et al., 2019) was performed to differentiate 176tectomanganates (todorokite) from phyllomanganates (buserite-, lithiophorite-, or 177asbolane-like phases – Wu et al., 2019). Briefly, 0.1 g sample was put in 250 mL of a 178 1 M HNO₃ solution with soft stirring at ambient temperature (20 °C). At different 179 time intervals, 5 mL of the suspension was collected and readily filtered through a 180 0.22 µm membrane to determine Mn, Ni, and Mg release. After 1 week of this HNO₃ 181 treatment, residual samples were dried at 60 °C for 3 days and analyzed with X-ray 182diffraction (XRD) at a scan speed of $10 \circ 2\theta$.min⁻¹. 183

Mineralogy of all samples was determined from powder XRD analysis using a 184 Bruker D8 Advance diffractometer equipped with Cu K α source ($\lambda = 0.15418$ nm) 185 and operated at 40 kV and 40 mA. XRD patterns were collected at 0.02 $^{\circ}2\theta$ intervals 186 187 using a continuous scan mode. XRD data was collected on fresh samples (dried at 60 °C for three days) at a scan speed of 10 ° 2θ .min⁻¹. To assess the stability of their 188 hydration, XRD data was collected on fresh samples after heating to 140 °C. Finally, 189 XRD patterns were collected at $1^{\circ} 2\theta$.min⁻¹ for aged samples that were kept dry in 190 191 the dark for three months. Synchrotron XRD patterns were also collected on selected reflux products on beamline BL14B1 at the Shanghai Synchrotron Radiation Facility 192 (SSRF) and on beamline 4B9A at the Beijing Synchrotron Radiation Facility (BSRF). 193 At BSRF, XRD data was collected at the Mn K-edge (6.5 keV), Ni K-edge (8.3 keV), 194 and 12 keV with a typical beam size of $2 \times 1 \text{ mm}^2$, whereas the 18 keV data was 195 collected at SSRF with a typical beam size of $\sim 0.2 \times 0.3 \text{ mm}^2$. In both cases, X-rays 196 were monochromatized with a double-crystal Si (111) monochromator. Finally, 197 high-energy X-ray scattering data was collected in transmission mode on beamline 198 11-ID-B at the Advanced Photon Source (Argonne National Laboratory, Argonne, IL), 199 using a ~90 keV X-ray energy. Pair distribution functions (PDF) and differential 200 PDFs were calculated as described elsewhere (Li et al., 2011; Wang et al., 2013). 201

202 Infrared spectra were collected for both layered precursors and reflux products using a Bruker VERTEX 70. For each sample, 64 spectra were collected and summed 203 over the 4000–400 cm^{-1} range with a 4 cm^{-1} resolution (Zhao et al., 2015). 204 Micromorphology of the reflux products and lattice periodicities were observed a Jeol 205 206 JEM2100F transmission electron microscope (TEM) operated at 200 kV. TGA was performed on a NETZSCH TG 209 instrument with a heating rate of 10 °C min⁻¹ and 207 a N₂ flow of 20 mL min⁻¹. Finally, X-ray photoelectron O1s spectra were collected 208 with a VG Multilab2000. X-ray photoelectron spectroscopy (XPS) measurements 209 used an Al K α X-ray source (1486.6 eV) at a chamber base pressure of 3×10^{-9} Torr 210 (Yin et al., 2011a). The O 1s spectra were collected with a 25 eV pass energy and 211 0.1 eV energy steps, and analyzed with the Avantage software. All samples were 212 213 charge-corrected to C 1s with a binding energy of 284.80 eV collected from the surface adventitious carbon (Wang et al., 2010). A Shirley-type background was 214 215 subtracted before deconvolution and fitting. A Lorentzian:Gaussian ratio of 30:70 was used for all the fittings. 216

217 Finally, X-ray absorption spectra were collected at the Mn and Ni K-edges on the 1W1B beamline of BSRF to determine the relative proportions of Mn(II), Mn(III), 218 and Mn(IV) species and the local environments of Mn and Ni in both layered 219 220 precursors and reflux products. Data was recorded at room temperature in 221 transmission (Mn) and fluorescence (Ni) modes, and the energy was calibrated using 222 metallic Mn/Ni foils as references. Extended X-ray absorption fine structure (EXAFS) spectra were processed using Athena and Artemis (Ravel and Newville, 2005). X-ray 223 224 absorption near-edge structure (XANES) spectra of both layered precursors and reflux 225 products were used to determine Mn average oxidation state (AOS) with a specific linear combination fitting method (the so-called Combo method). Mn(II), Mn(III), 226 and Mn (IV) reference compounds were those of the original study (Table 1 in 227 228 Manceau et al., 2012).

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231	3. RESULTS
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233	3.1. Elemental composition
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235	With increasing Ni content, the Mn content decreases from ~55.4 to ~44.8 wt.%
236	for layered precursors and from \sim 54.4 to \sim 49.2 wt. % for reflux products (Table 1).
237	The Ni/(Mn+Ni) molar ratios are essentially similar in layered precursors and their
238	respective reflux products, with only a very limited fraction of Ni being lost either
239	during the Mg-for-(Na,Ni) exchange or during the subsequent transformation process.
240	In addition, the Mg wt.% in the reflux products significantly decreases from ~3.95 wt.%
241	in the Ni-free Tod sample to ~2.11 wt.% in NiT20, thus suggesting a competition
242	between Mg and Ni for interlayer and/or tunnel sites of layered precursors and/or of
243	reflux products. Specific surface area (SSA) of NiT samples, as measured with the
244	BET method, is systematically higher than that of Ni-free Tod, but no systematic
245	evolution of SSA is observed with the increase of Ni content.
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247	3.2. Powder X-ray diffraction
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249	X-ray diffraction (XRD) patterns collected on fresh Bir (not shown) are

essentially similar to those recorded after a few months of aging and display 250reflections at 7.2, 3.6, ~2.45, and ~1.42 Å, typical for birnessite (Drits et al., 1997; 251Lanson et al., 2002b). Similar reflections are observed for fresh NiB5 and NiB10 252samples (Fig. S1), in addition to a broad maximum peaking at ~4.6 Å. The intensity of 253 this maximum increases with increasing Ni concentration in the initial solution and is 254 thus likely related to the presence of Ni (hydr)oxide (ICDD #13-0229), whose 255 formation was reported under similar experimental conditions (oxidation of a Ni-rich 256 alkaline solution - Feitknecht et al., 1956). Consistently, the intensity of this 257 258maximum is increased further for NiB15 and NiB20. These two samples exhibit also reflections at ~2.45, and ~1.42 Å, typical for layered manganates, the basal reflections 259

of which indicate a 9.9-10.0 Å layer-to-layer distance (Fig. S1). Upon moderate 260 heating (140 °C), these reflections shift to indicate a 7.2 Å layer-to-layer distance 261 typical for birnessite, as the result of interlayer cation partial dehydration (Golden et 262 al., 1986; Manceau et al., 1987; Drits et al., 1997). The initial presence of two planes 263 of interlayer H₂O molecules in NiB15 and NiB20 (buserite) is most likely indicative 264 of the presence of interlayer hydrated Ni(II) cations, whose hydration is reduced upon 265heating. Sample heating also results in a decrease of the intensity of the ~4.6 Å 266 267 maximum.

After aging for about three months, all layered precursors but NiB20 display over 268 the low-angle region (5-30 $^{\circ}2\theta$) strong basal reflections at ~7.2 and ~3.6 Å typical for 269 birnessite and indicative of the spontaneous dehydration of NiB15 interlayers (Fig. 1). 270Two planes of H₂O molecules are still present in most NiB20 interlayers (buserite), 271however. Splitting of reflections at ~2.45 and ~1.42 Å, and more generally positions 272of the reflections over the high-angle region (30-85 $^{\circ}2\theta$ – Fig. 1), unambiguously 273 indicate that initial Bir is a triclinic birnessite (ICDD #43-1456), consistent with the 274275synthesis protocol (Lanson et al., 2002b). With increasing Ni content, high-angle reflections broaden significantly thus indicating a decrease in the coherent scattering 276 277 domain size. In addition, the resolution of these reflections decreases also with increasing Ni content, suggesting a minor increase of layer symmetry from orthogonal 278 $(a > b\sqrt{3}, \gamma = 90^{\circ})$ to hexagonal $(a = b\sqrt{3}, \gamma = 90^{\circ} - Yin$ et al., 2015). Positions of 279280 birnessite hkl reflections remain essentially unchanged, however, indicative of the persistence of the orthogonal layer symmetry in Ni-rich layered precursors. The broad 281 282 maximum at \sim 4.6 Å is present also in aged samples.

Reflux products systematically show reflections at ~9.6 Å, ~4.8 Å, ~3.2 Å, ~2.46 Å, ~2.39 Å, ~2.22 Å, ~1.95 Å, ~1.73 Å, ~1.52 Å, and ~1.42-1.41 Å, that are typical of todorokite (ICDD #38-0475 – Atkins et al., 2014, 2016). For the Ni-free Tod, relative intensity of the three low-angle reflections (at ~9.6, ~4.8, and ~3.2 Å) decreases in the order $I_{9.6} > I_{4.8} >> I_{3.2}$, consistent with relative intensities of todorokite 001/100, 002/200, and 003/300 reflections (Post et al., 2003), respectively. With increasing Ni content, the positions of both ~9.6 and ~4.8 Å reflections shift

towards higher angles for NiT15 and NiT20, compared to Ni-free Tod (Fig. S2) and 290 the ~4.8 Å reflection becomes predominant, consistent with previous reports of 291 todorokite prepared in the presence of various divalent metals (Shen et al., 1993, 1994; 292 Ching et al., 1999; Nicolas-Tolentino et al., 1999; Fuertes et al., 2012) and 293 asymmetric. Although not described as such in these references, this intensity ratio 294 modification is consistent with the presence of asbolane, a phyllomanganate with an 295 incomplete - island-like - octahedral layer of metal (hydr)oxides in the interlayer 296 297 (Chukhrov et al., 1987; Golden et al., 1987; Fan and Gerson, 2015). The presence of these incomplete brucite-like octahedral layers between MnO₂ layers precludes the 298 collapse of the layer-to-layer distance to ~ 7.2 Å upon heating (Fig. S3) 299

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- 301 **3.3. Anomalous XRD patterns**
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To investigate further the possible origin of the observed contrasting $I_{4.8}/I_{9.6}$ 303 intensity ratios, XRD patterns of Tod, NiT15, and NiT20 were recorded at different 304 305 energies, including Mn and Ni K-edges to modify the contribution of these elements to the structure factor (F_{hkl}) . These XRD patterns logically exhibit reflections at 306 similar d-values (Fig. 2a-d). Except for the pattern recorded at the Mn K-edge all Tod 307 patterns also display similar low I_{4.8}/I_{9.6} intensity ratios (Fig. S4), consistent with 308 309 previous reports (Shen et al., 2011). Contrastingly, the I_{4.8}/I_{9.6} intensity ratio is significantly higher for NiT15 and NiT20 compared to Tod, and varies significantly 310 311 for these two samples as a function of the energy used for data collection. Specifically, 312 NiT15 and NiT20 display highest I4.8/I9.6 intensity ratios at the Mn K-edge and lowest 313 ratios at the Ni K-edge, whereas similar ratios are obtained for XRD patterns recorded at energies different from the two absorption edges (Figs. 2a-d; S4). Both features 314 indicate contrasting structures of Tod and Ni-rich NiT products depending on the 315 initial Ni content. 316

As reported in the early descriptions of asbolane, this compound exhibits high values of the $I_{4.8}/I_{9.6}$ intensity ratio (Chukhrov, 1982; Llorca, 1988; Fan and Gerson, 2015), consistent with XRD patterns calculated for such a structure hosting an

incomplete Ni(OH)₂ layer sandwiched in between two MnO₂ layers (Fig. 2e). In 320 321 addition, for a given composition, this ratio increases when the contribution of Mn atoms to the structure factor is reduced (XRD data recorded at the Mn K-edge – Fig. 322 2a) and decreases when the contribution of Ni atoms to the structure factor is reduced 323 (XRD data recorded at the Ni K-edge – Fig. 2b), consistent with the data. In addition, 324 this ratio increases with the completeness of interlayer plane (Fig. 2e). Together with 325 the collapse of layer-to-layer distance upon acid-dissolution of the incomplete 326 327 Ni(OH)₂ layer (see below), these variations of the I_{4.8}/I_{9.6} intensity ratio can be ascribed to the presence of asbolane, rather than todorokite, in the reflux products. 328

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330 **3.4. FTIR**

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FTIR spectra of todorokite and of other 3×n tectomanganates display a band at 332 ~760 cm⁻¹, typical for the corner-sharing linkage in their structures (Fig. 3 – Golden et 333 al., 1986; Feng et al., 2004; Julien et al., 2004; Atkins et al., 2014; Zhao et al., 2015). 334 335 This band is clearly visible in Tod FTIR spectrum but steadily decreases in intensity with increasing Ni content, most likely as the result of the decreasing proportion of 336 3×n tectomanganates in Ni-rich reflux products. Consistent with their layered 337 structure, this band is absent in FTIR spectra of layered precursors. Rather, two lines 338 at ~820 and ~950 cm⁻¹ are visible in the FTIR spectra of Ni-containing layered 339 340 precursors. These two lines are visible also in the spectra of related reflux products, although their intensity is reduced after the reflux treatment. The peak at ~ 820 cm⁻¹ 341 visible in NiB and NiT was identified as a combination lattice mode band in Ni 342 343 (hydr)oxides (Hall et al., 2015), which is consistent with the XRD analysis.

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345 **3.5. HRTEM**

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Consistent with previous studies (Turner and Buseck, 1981; Golden et al., 1986; Shen et al., 1993; Feng et al., 2004; Wu et al., 2019), crystals with fibrous and plate-like morphologies are systematically present in the reflux products (Fig. 4), the 350 former dominating in Tod and NiT1 (Figs. 4a-c), whereas platy crystals dominate samples having higher Ni-contents (NiT5 and NiT15; Figs. 4f, i). Fiber-shaped 351 crystals (Figs. 4 e, g, h) often display lattice fringes with a typical ~ 1 nm periodicity, 352 353 corresponding to the 3×3 todorokite tunnel structure (Golden et al., 1986; Feng et al., 354 2004). As it is common in todorokite, fibers are often aligned at $\pm 120^{\circ}$ with respect to each other, thus forming trilling patterns (Siegel and Turner, 1983; Golden et al., 1986; 355 Feng et al., 2004; Bodeï et al., 2007; Xu et al., 2010; Atkins et al., 2014). Visible 356 lattice fringes in the plates are commonly more diverse than in the fibers (Figs. 4 b, d 357 and i), typical for *a*-disordered todorokite (Xu et al., 2010; Wu et al., 2019). Lattice 358 fringes are also visible in short rods and small plates, indicating the initial 359 development of tunnel structure therein (Figs. 4 g, h – Golden et al., 1986). Finally, 360 361 smooth surfaces without visible fringes, that are most likely typical for phyllomanganate crystals (Golden et al., 1986, 1987; Wu et al., 2019), are also visible 362 (Figs. 4 g – zone 4, 4i, j – zone 1, 4k – zone 5). Interestingly, some platy particles 363 exhibit lattice fringes over specific zones with smooth surfaces (Figs. 4g, i). This dual 364 365 character is possibly related to the initial development of tunnel structures in specific zones of the particles whereas other zones maintain their initial planar structure 366 (Golden et al., 1987; Wu et al., 2019). 367

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- 369 **3.6. Nitric acid treatment**
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Upon equilibration in nitric acid, Mn release from layered precursors reaches 371 equilibrium within 12 hrs, the proportion of Mn released from NiB is higher than that 372 released from Bir, (Fig. 5a), possibly as an effect of reduced crystal size. For reflux 373 products, the release of Mn is slower, reaching equilibrium after ~48 hrs, and is 374 essentially the same for all NiT samples (Fig. 5b), except for NiT5. Ni release is much 375 slower than that of Mn, reaching a plateau after ~168 hrs in both layered precursors 376 (Fig. 5c) and reflux products (Fig. 5d). The proportion of Ni released differs however 377 in the two series of samples as essentially all Ni is released from layered precursors 378 whereas ~20% of Ni remains in the reflux products, possibly indicative of Ni 379

structural incorporation. Finally, Mg is systematically totally released from reflux
 products, possibly at a slower rate with increasing Ni content (Fig. 5f).

After acid treatment, the XRD pattern of Tod is essentially alike that of the initial 382 reflux product and typical for todorokite (Fig. 5e). On the other hand, XRD patterns 383 recorded on NiT15 and NiT20 after acid treatment exhibit, over the low-angle region, 384 peaks at ~ 7.2 and ~ 3.6 Å, typical for the phyllomanganate birnessite, rather than 385 peaks at ~9.6, ~4.8, and ~3.2 Å, as expected for todorokite (Fig. 5e). These two XRD 386 patterns are actually similar to the one recorded on Bir after acid treatment and match 387 XRD pattern of hexagonal birnessite (ICDD #23-1239), which was obtained from the 388 low-pH equilibration of an initial triclinic birnessite (Lanson et al., 2000). The 389 evolution of XRD patterns recorded from reflux products upon acid treatment is 390 391 consistent with that observed for an asbolane-like phyllomanganate (Wu et al., 2019). The structure of these ~ 9.6 Å phyllomanganates consists of an octahedral MnO₂ layer 392 similar to that of birnessite, with an incomplete octahedral Me(OH)₂, MeOOH, or 393 MeO₂ sheet, possibly incommensurate with the MnO₂ layer, in the interlayer 394 (Chukhrov, 1982; Manceau et al., 1987; Fan and Gerson, 2015). This incomplete 395 interlayer octahedral sheet is solubilized by the acid treatment leading to the observed 396 7.2 Å layer-to-layer distance (Wu et al., 2019). Such a peak at ~7.2 Å is visible in the 397 XRD pattern of acid-treated NiT5 and NiT10, and its intensity increases with 398 399 increasing Ni content, becoming the sole visible diffraction signature in acid-treated NiT15 and NiT20 (Fig. 5e). 400

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402 **3.7. Thermogravimetric analysis**

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Thermogravimetric analyses of layered precursors systematically show a significant weight loss at 140-150 °C corresponding to the dehydration of interlayer Na⁺ and Ni²⁺ cations, this weight loss decreasing with increasing Ni content (Figs. 6a, c; Table 2). A minor weight loss (2.4-2.9% in Bir, NiB5, NiB10, and NiB15) is also observed for these samples below 100 °C, which is most likely related to the departure of adsorbed water (Gaillot et al., 2003). This low-temperature weight loss is

significantly increased for NiB20 (9.25%), however, most likely as the result of a 410 411 low-temperature buserite-to-birnessite transition associated with the loss of one plane of interlayer H₂O molecules as observed during moderate heating of fresh NiB 412 samples (Fig. S1). A significant weight loss (4.10 and 6.72% for NiB5 and NiB20, 413 respectively) is observed also for NiB samples over the 170-400 °C range. This 414 weight loss is essentially absent in the TGA curve of Bir (1.78%) and increases with 415 Ni content. It is likely related to the presence of the Ni (hydr)oxide responsible for the 416 broad XRD maximum peaking at ~4.6 Å (Fig. 1). Finally, a weight loss is observed at 417 ~550 °C for all layered precursors, with no significant influence of the Ni content, 418 probably indicating the reduction of MnO₂ to Mn₂O₃ (Bish and Post, 1989). 419

420 TGA curves recorded for reflux products significantly differ from those of their 421 layered precursors, with weight losses at ~150, ~285, ~440, and ~545 °C for Tod (Figs. 6b, d). The weight loss observed from ~110-240 °C is most likely related to the 422 loss of the H₂O molecules hydrating cations present in todorokite tunnels (Bish and 423 424 Post, 1989). With increasing Ni content, this weight loss decreases (from 8.05 to 6.12% 425 in Tod and NiT20, respectively) and shifts towards lower temperatures, most likely 426 indicative of the mineralogical evolution evidenced by XRD data recorded on acid-treated reflux products (Fig. 5e). For NiT10, this low-temperature weight loss 427 splits in a doublet owing to the coexistence of todorokite and asbolane (Fig. 5e). The 428 429 weight loss occurring at ~285 °C is also ascribed to the departure of tunnel water (Bish and Post, 1989). TGA curves obtained for Tod and NiT20 also differ 430 significantly over this range, with a shift of the weight loss towards higher 431 temperatures in NiT20 compared to Tod. Evolution with increasing Ni content, is not 432 433 as systematic as that observed for the low-temperature weight loss, however (Fig. 6d). The third weight loss occurring at ~440 °C appears as a distinct signature for the 434 presence of todorokite, as its intensity decreases with increasing Ni content to vanish 435 for NiT15 and NiT20 which essentially contain asbolane (Fig. 5e). A last weight loss 436 occurs at 520-550 °C, a temperature similar to that reported for layered precursors at 437 438 ~550 °C (Figs. 6b, d).

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Consistent with the systematic prevalence of [MnO₆] building octahedral in both 442 phyllomanganates and tectomanganates, all O1s XPS spectra are dominated by a peak 443 at ~529.6 eV (Fig. 7), typical for oxygen (O^{2-}) in metal oxides. This main peak is 444 asymmetric and displays a tail on its high-energy side for both layered precursors and 445 reflux products. This tail at ~531-535 eV can be decomposed into the contributions of 446 447 oxygen atoms from hydroxyl groups (OH; 530.9-531.6 eV) and from water molecules (532.2-533.4 eV). For layered precursors, the intensity of this tail, and 448 more especially that of the OH⁻ contribution, increases with increasing Ni content (Fig. 449 7; Table S1), whereas the H_2O contribution is about constant for all NiB samples but 450 451 NiB20 that shows a slightly higher proportion of O atoms from H₂O molecules, consistent with TGA data. In reflux products the contribution of O atoms in hydroxyl 452 groups is significantly higher in NiT samples compared to Tod, consistent with the 453 presence of asbolane-like phyllomanganate in Ni-rich reaction products. On the other 454 hand, the H_2O contribution shows no systematic evolution with Ni content (Fig. 7; 455 Table S1). Finally, relative intensities of peaks corresponding to OH^{-} and H_2O 456 contributions are significantly increased in NiT samples compared to their 457 corresponding layered precursors, at least under the deep vacuum conditions 458 459 prevailing during the XPS measurements.

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461 **3.9. X-ray absorption spectroscopy**

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463 3.9.1. Mn K-edge XANES

Relative proportions of Mn(II), Mn(III), and Mn(IV) species and Mn average oxidation state (AOS) have been determined using the Combo method (Table 3 – Manceau et al., 2012). Results consistently indicate high (~1/3) relative proportions of Mn(III) in both layered precursors and reflux products, which is considered a key condition for the phyllomanganate-to-tectomanganate conversion (Cui et al., 2008; Atkins et al., 2014, 2016; Grangeon et al., 2014, 2015; Zhao et al., 2015; Li et al., 2016). The measured proportion of Mn(III) is slightly lower in Ni-free samples
compared to the Ni-bearing ones (Table 3) and no significant difference is observed
between layered precursors and their reflux products. In all cases, the Mn(II) content
is null within the experimental error.

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475 *3.9.2. Mn K-edge EXAFS*

EXAFS spectra of both layered precursors and of reflux products show 476 significant evolutions with increasing Ni content, thus indicating that the presence of 477 Ni affects the local structure of manganates in the two series of samples (Fig. 8). The 478 EXAFS spectrum collected for Ni-free layered precursor (Bir) is typical for triclinic 479 birnessite over both the staircase and the indicator regions $(3.5-6.0 \text{ Å}^{-1} \text{ and}$ 480 7.5-10.0 Å⁻¹, respectively – Marcus et al., 2004). In particular, the splitting of the 481 features at ~7.8 Å⁻¹ and ~9.0 Å⁻¹ (arrows in Fig. 8a) are indicative of a high Mn(III) 482 483 content, of the systematic elongation of Mn(III) octahedra along the a axis, and of their ordered distribution in rows parallel to the b axis (Marcus et al., 2004; Gaillot et 484 al., 2007), typical features for triclinic birnessite (Webb, 2005; Atkins et al., 2014; Yin 485 et al., 2014). With increasing Ni content, the ~7.8 Å⁻¹ feature becomes less 486 pronounced even though its position remains unchanged, consistent with the similar 487 Mn(III) contents of all layered precursors (Table 3) and with the similar overall 488 frequency of the EXAFS spectra of these samples (Gaillot et al., 2007). The steady 489 lowering of the ~7.8 Å⁻¹ feature resolution is thus most likely related either to disorder 490 in the distribution of elongated Mn(III) octahedral or to some randomness in the 491 direction of the elongation, consistent with the evolution of XRD patterns with 492 increasing Ni content. The split ~9.0 \AA^{-1} feature observed for Bir and NiB1 also turns 493 494 into a broad and more symmetrical feature with increasing Ni content and the induced vanishing of the sharp high-frequency contribution. 495

In k-space Mn K-edge EXAFS spectra of Tod and NiT1 exhibit sharp features at ~ 8.0 and ~ 9.2 Å⁻¹ that are typical for todorokite (Webb, 2005; Bodeï et al., 2007; Feng et al., 2010; Atkins et al., 2014). These two features essentially decrease in intensity at higher Ni content. In addition, the ~ 9.2 Å⁻¹ feature shifts to lower wave numbers in 500 NiT samples compared to Tod and NiB1 (Fig. 8b). EXAFS spectra of Ni-rich NiT 501 samples and their Fourier transforms (FTs) are actually alike those of their corresponding layered precursors (Figs. 9a-b), consistent with the similar structure of 502 the MnO₂ layer in both birnessite and asbolane. On the contrary, spectra obtained for 503 Ni-free Bir and Tod differ significantly, as expected for layered and tunnel manganese 504 oxides (Fig. 9c). In particular, the FTs of these spectra show a clear shortening of the 505 average Mn-Me_E distance in Tod compared to Bir, as shown by the shift of the 506 imaginary part of the FT (Fig. 9d, arrow). In addition, Tod FT shows a clear 507 maximum at $R + \Delta R \sim 3.0$ Å, typical of Mn-Me_C pairs, which is not observed for its 508 Bir precursor. For both Ni-free and Ni-rich samples, Mn-O distances are similar in 509 both layered precursors and corresponding reflux products, consistent with their 510 511 similar Mn AOS.

512

513 3.9.3. Ni K-edge EXAFS

EXAFS spectra collected at the Ni K-edge are similar for both NiB samples, on 514 515 the one hand, and NiT, on the other hand (Fig. 10), thus implying similar Ni speciation in each of the two sample groups. Although it was not possible to obtain a 516 517 decent fit to this data using a linear combination fitting and available Ni reference spectra, similarity of the hydrated Ni(II) spectrum with collected spectra suggests that 518 519 this species possibly prevails in all samples. In addition, the evolution of Ni speciation during the reflux process was assessed by fitting EXAFS spectra of reflux products as 520 a linear combination of their respective NiB precursors and of Ni reference spectra 521 (Fig. S4). For all NiT samples, a good fit was obtained with $\sim 15\%$ of a Ni(OH)₂ 522 contribution in addition to the layered precursor (Table S2), consistent with the 523 polymerization of a partial Ni(OH)₂ layer in birnessite interlayers to form an 524 525asbolane-like structure.

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527 **3.10.** Pair distribution functions (G(r))

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529 Atomic PDFs computed from Tod, NiT10, and NiT20 high-energy X-ray

scattering patterns show similar peak positions (Fig. 11), indicative of similar local 530 structures (Zhu et al., 2012) as expected from the major contribution of layers/ribbons 531 of edge-sharing MnO₆ octahedra in both layered precursors and reflux products. 532 Consistently, the maxima at ~1.90 Å, ~2.90 Å, and ~3.50 Å essentially correspond to 533 the first neighbor Mn-O, to edge-sharing Mn-Me, and to a combination of 534 corner-sharing Mn-Me and of second neighbor Mn-O atomic pairs, respectively (Zhu 535 et al., 2012). The increased intensity of the ~3.50 Å maximum and the decreased 536 intensity of the one at ~5.00 Å in the PDF of Tod compared to NiT10 and NiT20 are 537 typical of the prevalence of tunnel structures in the former sample and of layered 538 structures in the latter (Grangeon et al., 2015). The ~5.00 Å maximum is mainly 539 related indeed to second neighbor Mn-Mn pairs ($\overline{Fig. S4} - Zhu$ et al., 2012), that are 540 favored by the lateral extension of MnO_2 layers. Finally, the peak at ~4.50 Å is most 541 likely related to Mn2-Mn2 and Mn4-Mn4 pairs across "holes" in tunnel walls, floors, 542 and ceilings (Fig. S6), consistent with the increased intensity of this ~4.50 Å peak 543 with decreasing tunnel size (Zhu et al., 2012). 544

545 Differential PDF (d-PDF) can be calculated to assess more precisely the specific modifications affecting the reflux products, and more especially the broadening of the 546 ~1.90 and 2.90 Å peaks towards high d-values with increasing Ni content (Fig. 11). 547 The d-PDF curves obtained for NiT10 and NiT20, both show a first peak at ~2.05 Å 548 typical for Ni-O pairs (Manceau et al., 2007a,b; Peacock and Sherman, 2007a,b; Yin 549et al., 2012; Kwon et al., 2013; Simanova et al., 2015; Atkins et al., 2016). A second 550 peak, similar in both d-PDF curves, is visible at ~3.05 Å. This second peak is 551 intermediate between those calculated Ni-Ni pairs in NiO and Ni(OH)₂, consistent 552 with the presence of both compounds in the reflux products identified from XRD data 553(Figs. 1, 2). 554

4. DISCUSSION

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4.1. Structural transformation of layered precursors

560 Addition of Ni during the synthesis of layered precursors does not impact significantly their crystal structure. In particular, the Mn(III) content of layered 561 precursors remains about constant at $\sim 1/3$ (Table 3) whatever the initial Ni content. 562 Consistently, XRD results (Fig. 1) show that layer symmetry remains orthogonal (a > a563 $b\sqrt{3}$) for all precursors, indicative of the systematic elongation of distorted Mn(III) 564 octahedra along the a axis and of their ordered distribution. FTs of the EXAFS spectra 565 566 consistently show similar Mn-Me_E distances in all layered precursors (not shown). Both XRD patterns and EXAFS spectra of layered precursors are significantly 567 modified when increasing the Ni content, however. These changes reveal an increased 568 disorder in the orientation of Mn(III) octahedra and/or in their distribution. 569

The increasing intensity of the ~4.6 Å diffraction maximum with increasing Ni 570 content indicates that part of Ni initially introduced in solution actually precipitates as 571 Ni (hydr)oxide. The presence of Ni in the interlayers of NiB15 and NiB20 is attested 572 by the presence of two planes of interlayer water molecules, consistent with its more 573 negative hydration enthalpy compared to Na⁺ (Smith, 1977; Johnson and Post, 2006), 574 in these fresh samples. All Ni species, including the two mentioned above, are readily 575 mobilized by the acid treatment (Fig. 5c), and it is thus reasonable to hypothesize that 576 no Ni(II) cations are incorporated in the octahedral layer. Similarly, the orthogonal 577 layer symmetry is typical of octahedral layers containing few vacant layer sites (Drits 578 et al., 1998; Lanson et al., 2000), if any, and the presence of Ni(II) cations sorbed 579 580 above or below such vacant sites is thus unlikely.

Although weakly bound to the layered precursors, the presence of Ni(II) 581 582 significantly impacts their transformation during the reflux process. Bir, a typical 583 triclinic birnessite, converts readily and completely to an *a*-disorder todorokite (Wu et al., 2019). With increasing Ni content, disorder increases together with the proportion 584 of non-ideal 3×n tunnel sizes forming trilling patterns. In addition, platy crystals are 585visible in all reflux products, their relative abundance increasing with increasing Ni 586 content with only minimal fragments of tunnel oxides being detected in NiT15 and 587 NiT20 (Fig. 4). Platy crystals are most likely asbolane, as shown by the prevalence of 588

589 this phyllomanganate hosting an incomplete octahedral layer of metal (hydr)oxides in their interlayers in Ni-rich reflux products. These incomplete brucite-like octahedral 590 layers are dissolved by the acid treatment, leading to birnessite-like structures (Fig. 591 5e). As reported in the early work of Chukhrov and coworkers on natural asbolane 592 (Chukhrov, 1982; Chukhrov et al., 1987), XRD diffraction patterns of asbolane are 593 characterized by high I_{4.8}/I_{9.6} intensity ratios. Such high I_{4.8}/I_{9.6} intensity ratios are 594 commonly reported for "todorokite", especially when synthesized in the presence of 595 596 metallic elements such as Co, Ni, Cu, Zn, Er, La, etc (Shen et al., 1994; Tian et al., 1997; Ching et al., 1999; Luo et al., 1999). In the present study, the combined use of 597 XRD simulation, anomalous XRD, HRTEM, thermogravimetric analysis, and nitric 598 acid treatment allowed uncovering the actual nature of this seldom described 599 phyllomanganate, but its formation may be far more common than previously 600 reported in these experimental studies. 601

As shown in Fig. 2e, this $I_{4,8}/I_{9,6}$ intensity ratio strongly depends on the 602 completeness of the interlayer brucite-like octahedral layer, and the ratio determined 603 604 for NiT15 and NiT20 corresponds to a ~50% completeness of the Ni(OH)₂ octahedral layer. This interlayer occupancy is much higher than possibly achieved with the sole 605 606 presence of Ni(II) in these octahedral layers as Ni / (Ni+Mn) ratios are 8-12% in these samples, with part of the Ni present as Ni (hydr)oxide. It is thus likely that Mg cations 607 608 contribute to the electron density at the interlayer mid-plane, in addition to Ni. As Mg 609 scattering factor is lower than that of Ni, the actual completeness of the interlayer 610 brucite-like layer is likely higher than 50%.

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612 **4.2. Differentiation of asbolane and todorokite.**

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Although the $I_{4.8}/I_{9.6}$ intensity ratio strongly depends on the completeness of the interlayer brucite-like octahedral layer, a 4.8 Å peak stronger than its 9.6 Å counterpart is a possible indication of the presence of the phyllomanganate asbolane, rather than the tectomanganate todorokite, in reaction products, or in natural samples. The two compounds indeed share these two interplanar distances. With its two planes of interlayer H₂O molecules, buserite also exhibits similar periodicity but a gentle heating (to ~100°C) should lead to the partial dehydration of interlayer cations and to the ~7.2 Å periodicity typical of birnessite (Fig. S1 – Johnson and Post, 2006). Furthermore, buserite is unlikely to have a high enough content of interlayer cations to induce a $I_{4.8}/I_{9.6}$ ratio higher than 1.0.

Although values of this ratio higher than 1.0 are typically observed for asbolane, 624 the partial exchange of Mg present in todorokite tunnels by cations having a higher 625 626 electronic content (such as Ni and other divalent metal cations) can also lead to such high $I_{4,8}/I_{9,6}$ ratios (Fig. 11), thus not allowing an unambiguous differentiation 627 between asbolane and todorokite. In this search for an unquestionable differentiation 628 between asbolane and todorokite, the nitric acid treatment proposed by Wu et al. 629 630 (2019) thus appears as a simple and reliable method. In the case of asbolane, dissolution of the interlayer brucite-like layer induced by the treatment leads indeed to 631 a collapse of the layer-to-layer distance to the ~7.2 Å periodicity. On the contrary, 632 periodicities are maintained for todorokite, the tectomanganate framework being 633 634 unaltered, and positions of XRD maxima are thus unchanged.

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4.3. The role of Mn(III) and interlayer species during the transformation

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It is now accepted that the presence of a high Mn(III) content (>25-30%) in 638 layered precursors is required for their conversion to tectomanganates. This pivotal 639 640 role is most likely related to the steric strains induced by the presence of Jahn-Teller distorted Mn(III) octahedra in the octahedral layers. Migration of these Mn(III) 641 642 cations from layer to interlayer or layer kinking at these structurally weak points contribute to building tunnel walls (Grangeon et al., 2008, 2014; Atkins et al., 2014, 643 2016; Yang et al., 2018). In the present case, the ordered distribution of Mn(III) 644 cations in rows parallel to the b axis and separated from each other along the a axis by 645 two rows of Mn(IV) creates a structurally weak point favorable to the formation of 646 tunnels walls (Atkins et al., 2014; Grangeon et al., 2014). Consistently, the initial 647 Ni-free precursor Bir, a triclinic birnessite, was fully converted to todorokite. Similar 648

649 transformation of triclinic birnessite to tunnel structures was previously reported in the literature although 3×3 tunnels were not systematically obtained (Yang et al., 650 2018). Furthermore, the actual distribution and orientation of Mn(III) octahedra 651 present in the MnO₂ layer do not appear to be key factors for the 652 phyllomanganate-to-tectomanganate conversion as tectomanganates were obtained 653 from hexagonal birnessite precursors in which Mn(III) octahedra are distributed 654 and/or oriented at random (Grangeon et al., 2008, 2014; Atkins et al., 2014, 2016; 655 656 Zhao et al., 2015). The absence of transformation of NiB precursors to tunnel structures thus does not appear to be related to their Mn(III) content, which is similar 657 658 to or higher than that of the Ni-free Bir.

As layers of all NiB precursors are essentially devoid of Ni, both in the octahedral 659 660 layer and sorbed above/below vacant layer sites, the absence of transformation of these precursors to tunnel structures is thus likely due to kinetic effects favoring the 661 formation of asbolane. Specifically, polymerization of Ni as Ni(OH)₂ fragments in the 662 interlayers of initial birnessite/buserite (Fig. S5, Table S2) is favored over the 663 664 conversion to tectomanganates. In turn, the presence of interlayer Ni(OH)₂ fragments likely impedes the migration of Mn(III) from the layer to the interlayer thus 665 preventing the conversion to tectomanganates. Such competing effects are supported 666 also by the increase of (Mg + Ni)/Mn ratios with increasing Ni content (Table 1) and 667 are likely to occur with a variety of other hydrolysable metal cations. Consistently, 668 "todorokite" containing divalent metal cations and displaying high I4.8/I9.6 ratios 669 possibly indicate the presence of asbolane-like structure. Such competing effects 670 671 might contribute also to the prevalence of phyllomanganates over tectomanganates in 672 natural environments (Wu et al., 2019).

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4.4. Negative influence of Ni and Co on the phyllo-to-tectomanganate transformation: Contrasting mechanisms

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677 Both in the present work investigating the possible transformation of 678 Ni-containing layered precursors to tectomanganates and in the related study 679 investigating a similar process for Co-containing layered precursors, tectomanganate 680 formation was inhibited, totally (Ni) or partially (Co) for Me-rich precursors. The origin of these similar negative impacts differs however for the two metals. In the 681 present case, layered precursors contain sufficient amounts of Mn(III) (~1/3) to 682 683 trigger the transformation, as Ni(II) is not structurally incorporated in the precursors. As discussed above, polymerization of Ni(OH)₂ fragments in birnessite interlayers is 684 kinetically favored over migration of Mn(III) from the layer to the interlayer, or layer 685 686 kinking, however, leading to the formation of asbolane rather than tectomanganates. 687 In contrast and consistent with previous reports (Manceau et al., 1997; Yu et al., 2012; Kwon et al., 2013; Simanova and Pena, 2015; Yin et al., 2015), Co(II) is readily 688 oxidized by birnessite during the formation of initial layered precursors with Co(III) 689 690 being structurally incorporated in these precursors owing to the similar ionic radii of Co(III) and Mn(IV) (Wu et al., 2019). The symmetry of layered precursors is 691 increased from orthogonal to hexagonal. As a consequence, the content of Mn(III) in 692 Co-containing layered precursors is lowered and does not allow triggering the 693 694 transformation to tunnel structures (Grangeon et al., 2008, 2014; Atkins et al., 2014, 2016). In both cases, little Me is released to solution during the reflux process and Me 695 696 cations are structurally incorporated in the reflux products (~20% and ~80% for Ni and Co, from the nitric acid experiments and EXAFS simulation, respectively) thus 697 enhancing their sequestration in both tecto- and phyllomanganates. 698

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4.5. The fate of Nickel during the transformation of layered precursors

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Previous studies have consistently shown that Ni can be incorporated in the birnessite structure, either within the octahedral layer or sorbed at vacant layers sites (Peacock and Sherman, 2007b; Grangeon et al., 2008, 2017; Peacock, 2009; Peña et al., 2010; Yin et al., 2012; Kwon et al., 2013; Yin et al., 2014; Simanova et al., 2015; Atkins et al., 2016; Lefkowitz and Elzinga, 2017), consistent with natural occurrences (Bodeï et al., 2007; Manceau et al., 2007a,b; Peacock and Sherman, 2007a). In their literature review, Grangeon et al. (2017) showed that the fraction of Ni structurally 709 incorporated in birnessite octahedral layers may be high (up to $\sim 50\%$) only for low Ni 710 content (~1%), as reported for natural samples (Bodeï et al., 2007; Manceau et al., 2007a,b; Peacock and Sherman, 2007a). Structural incorporation of Ni(II) in the 711 712 octahedral layer appears to be enhanced by circumneutral pH conditions and time 713 (Manceau et al., 2007b; Peacock and Sherman, 2007b; Peacock, 2009; Peña et al., 2010). Competitive sorption at TCS/DCS sites has been proposed also as a 714 715 mechanism favoring structural incorporation in the octahedral layer (Lefkowitz and 716 Elzinga, 2017). Some studies report the prevalence of Ni sorbed at TCS/DCS sites even for Ni/Mn ratios higher than 10%, however (Manceau et al., 2007a,b; Peña et al., 717 718 2010; Yin et al., 2012; Simanova et al., 2015; Grangeon et al., 2017). In such cases, sorption of Ni(II) as inner sphere complexes is favored by the presence of vacant 719 720 octahedral sites and is thus negatively correlated with the Mn(III) content. 721 Consistently, in our study structural incorporation of Ni in vacancy-free NiB precursors appears very limited. In NiB layered precursors, Ni is rather present both 722 as Ni (hydr)oxide and as hydrated cations in birnessite/buserite interlayers, as shown 723 by the increased hydration of Ni-rich precursors (Figs. 1, S1). The influence of Ni on 724 phyllomanganate hydration has been reported also for natural samples in which 725 buserite (or 10 Å vernadite) abundance increases with increasing Ni content (Bodeï et 726 al., 2007; Manceau et al., 2007a). 727

728 As discussed above, hydrated Ni(II) initially present in birnessite interlayers are readily stabilized during the reflux products by forming $Ni(OH)_2$ fragments (Fig. S5, 729 Table S2), consistent with previous experiments with vacancy-free phyllomanganate 730 precursors (Feng et al., 2001). Polymerization of Ni(OH)₂ brucite-like octahedral 731 732 layers in phyllomanganate interlayers is consistent also with the natural predominance of asbolane in Ni-rich environments (NiO ~10-20 wt.%, - Chukhrov, 1982; Chukhrov 733 et al., 1987; Roque-Rosell et al., 2010; Ploquin et al., 2018) . Although the Mn AOS 734of his synthetic samples (~4.0) was favorable for Ni sorption at layer vacancy sites, 735 Manceau et al. (2007b) also reported the overwhelming presence of Ni as Ni(OH)₂ at 736 737 high Ni content (Ni/Mn ratio of ~0.10 and ~0.18 at pH 4 and 7, respectively), despite experimental conditions supposed to prevent Ni(OH)₂ precipitation. The reflux 738

739 process appears to have little influence on the Ni (hydr)oxide formed together with 740 the layered precursors, as evidenced by the stability of the corresponding maximum at ~4.6 Å in the reflux products (Fig. 1b). Finally, part of Ni (~20 %) is stabilized during 741 742 the reflux process, as shown by the incomplete Ni release during acid treatment (Fig. 743 5d). Stabilized Ni is possibly structurally incorporated in the reflux products. On the contrary, Atkins et al. (2016) reported no structural incorporation of Ni(II) during the 744 745 formation of todorokite from a Ni-bearing birnessite precursor (~1 wt% Ni in 746 hexagonal birnessite). They reported also a decrease in the Ni content of their reflux 747 products, whereas in our experimental conditions stability of Ni/(Mn+Ni) ratios (Table 1) indicates that all Ni initially present in layered precursors is retained during 748 the reflux process. 749

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5. CONCLUSION

753 The present study reports on the association of weakly bound Ni species 754 (hydrated Ni, Ni (hydr)oxides) with phyllomanganates, that possibly prevail in natural NiO) manganates. Whatever their Ni content, prepared 755 Ni-rich (>10% phyllomanganates systematically exhibit an orthogonal layer symmetry, indicative of 756 a high Mn(III) content and of a minimal number of octahedral layer vacancies. Ni is 757 758 essentially present as charge-compensating hydrated Ni(II) and as Ni (hydr)oxide, 759 with no detectable Ni(II) sorbed or structurally incorporated in the phyllomanganate 760 structure.

Despite their high content of Mn(III), a favorable condition for their conversion to tectomanganates, absolane rather than tectomanganates forms during the reflux processing of these Ni-rich phyllomanganates. Asbolane formation results from the kinetically favored polymerization of fragments of $Ni(OH)_2$ octahedral layers in birnessite interlayers. The structure of the initial manganate layers is essentially unaffected by this transformation.

Formation of asbolane, rather than todorokite, may actually be more frequent than

reported in the literature owing to the similar periodicities in both compounds. High values (> 1) of the $I_{4.8}/I_{9.6}$ intensity ratio may suggest the presence of this seldom described phyllomanganate. Acid nitric treatment, aiming at the dissolution of the island-like interlayer Ni(OH)₂ layer, represents an easy and unambiguous way to differentiate asbolane from todorokite which is unaffected by the treatment.

Both the presence of Co and Ni in synthesized phyllomanganates impede their 773 transformation to tectomanganates when refluxed, the origin of this similar impact 774 775 being different for the two metals, however. Oxidation of Co(II) sorbed on the MnO₂ 776 layer and the subsequent structural incorporation of Co(III) in the octahedral layer 777 releases the steric strains related to the presence of Jahn-Teller distorted Mn(III) octahedra, which is required for the phyllomanganate-to-tectomanganate conversion. 778 In contrast, the presence of hydrated Ni(II) favors the polymerization of fragments of 779 Ni(OH)₂ octahedral layers, that in turn prevents the formation of tectomanganates 780 despite the favorable presence of Mn(III) in the octahedral MnO₂ layer. In both cases, 781 "foreign" metal cations are retained in the refluxed products, with contrasting ratios of 782 783 structural incorporation.

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ACKNOWLEDGMENTS

787 Caroline Peacock (Leeds University, UK) is thanked for providing Ni K-edge 788 EXAFS data of Bir-Ni_s and Bir-Ni_v references. The authors are grateful to 789 Mengqiang Zhu (University of Wyoming, Laramie, Wy, USA) for helping collect 790 high-energy X-ray scattering data and providing Ni(II)-H₂O EXAFS reference. This 791 work is supported by the National Key Research and Development Program of China (No. 2016YFD0800400) and the Natural Science Foundations of China (No. 792 41571448). Xionghan Feng, Bruno Lanson, and Zhongkuan Wu are supported by a 793 Cai Yuanpei program between the Key Laboratory of Arable Land Conservation 794 795 (Wuhan, China) and ISTerre (Grenoble, France)

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¹⁰¹⁵ application for removal of lead and arsenite. J. Hazard. Mater. 188, 341-349.

1041 Table 1

1042 Chemical composition and specific surface area of layered precursors and reflux
 1043 products

Samples	Mn wt. %	Ni wt. %	Ni/(Mn+Ni) mol%	Mg wt. %	(Mg+Ni)/Mn mol%	SSA (m^2/g)
Bir	55.4(1.0)					
NiB1	54.1(1.1)	0.39(1)	0.67			
NiB5	53.7(1.0)	1.86(4)	3.18			
NiB10	51.2(1.0)	3.89(7)	5.98			
NiB15	47.3(1.7)	5.32(3)	8.71			
NiB20	44.8(1.8)	7.14(7)	11.50			
Tod	54.4(2.1)			3.95(8)	16.4	90.7
NiT1	54.3(8)	0.33(1)	0.57	3.51(12)	16.2	
NiT5	54.2(6)	1.57(2)	2.57	3.12(2)	15.6	93.7
NiT10	52.7(6)	3.79(1)	5.94	2.61(2)	17.5	119.7
NiT15	52.4(1.2)	5.66(9)	8.41	2.59(6)	20.4	109.0
NiT20	49.2(9)	7.74(2)	11.38	2.11(2)	22.5	106.8

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1045 Table 2.

1046 Weight loss of layered precursors and reflux products over specific temperature1047 intervals

Tommorotuno (°C)			Weight loss/	%	
Temperature (°C)	Bir	NiB5	NiB10	NiB15	NiB20
RT-100	2.45	2.87	2.64	2.84	9.25
100-170	8.43	7.14	6.54	5.73	4.87
170-400	1.78	4.10	4.74	5.78	6.72
Total	17.38	19.20	18.85	19.15	25.31
	Tod	NiT5	NiT10	NiT15	NiT20
RT-110	2.75	2.93	3.39	3.90	4.12
110-240	8.05	7.68	7.43	6.57	6.12
240-380	5.47	5.32	5.04	5.34	5.97
380-500	2.69	2.33	2.08	1.93	1.79
500-650	4.50	4.35	4.06	3.75	3.61
Total	23.47	22.61	22.01	21.48	21.60

1049 **Table 3**.

Fractional and average valence states of Mn obtained for layered precursors and reflux products from the Combo fit of XANES spectra and 1^{st} derivatives. The estimated error for Combo method is ±4% (Manceau et al., 2012; Yin et al., 2015)

Comula	Mn ²⁺	Mn ³⁺	Mn ⁴⁺	Ma AOS
Sample	at. %	at. %	at. %	Mn-AOS
Bir	3	26	72	3.69
NiB1	2	25	74	3.71
NiB5	2	30	67	3.64
NiB10	2	31	68	3.66
NiB15	0	31	69	3.69
NiB20	0	35	65	3.65
Tod	2	25	74	3.72
NiT1	1	35	65	3.64
NiT5	3	32	66	3.63
NiT10	2	35	64	3.62
NiT15	1	34	65	3.64
NiT20	0	33	67	3.67

1054 **Figure Captions**

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Fig. 1. XRD patterns of layered precursors (top) and reflux products (bottom). Colors correspond to the Ni/(Ni+Mn) ratio of the samples. Green ticks at the bottom of upper and lower figures indicate the reflections of triclinic birnessite (ICDD #43-1456) and todorokite (ICDD #38-0475), respectively.

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Fig. 2. XRD patterns collected for NiT20, NiT15, and Tod at (a) Mn K-edge (6.5 keV),
(b) Ni K-edge (8.3 eV), (c) 12.0 keV, and (d) 18.0 keV. Intensities were systematically
normalized to the most intense reflection. Colors as in Fig. 1. (e) XRD patterns
calculated for an asbolane containing 0.525 Ni(OH)₂ in MnO₂ interlayers. Contents of
Ni and Mn atoms are reduced by 20% at their respective absorption edges to simulate
the anomalous effect.

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Fig. 3. FTIR spectra of layered precursors and reflux products (top and bottom,
respectively). Colors as in Fig. 1.

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Fig. 4. HRTEM images of (a-b): Tod, (c-e): NiT1, (f-h): NiT5, (i-m): NiT1. Numbers in (b) and (d) indicate the number of MnO_6 octahedra defining tunnel width. Arrows in (i) and (l) indicate the direction of fringes bending.

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Fig. 5. Proportion of metal cations released during the nitric acid treatment: (a-b): Mn, (c-d): Ni, and (f): Mg. The amount of metal released is normalized to the sample overall metal content (see text for details). (e): XRD patterns of samples treated with nitric acid. Colors as in Fig. 1 except for Bir in (e).

Fig. 6. Thermogravimetric analysis traces obtained for (a) layered precursors and (b)
 reflux products, and their 1st derivatives (c and d, respectively). 1st derivative curves
 were smoothed using a Savitzky-Golay filter and a 15 point window. Colors as in Fig.
 1.

1084

Fig. 7. XPS O1s spectra obtained for layered precursors and reflux products (left and right, respectively). The thick solid curve represents the best fit to the data (open red squares). Peaks corresponding to O^{2-} , OH^{-} , and H_2O species are shown in the order of increasing binding energy as dashed purple, dotted-dashed green, and dotted blue lines, respectively. The green dotted line is the difference curve.

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Fig. 8. Mn K-edge EXAFS spectra obtained for (a) layered precursors and (b) reflux
products. Colors as in Fig. 1.

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Fig. 9. Comparison of (a, c) Mn K-edge EXAFS and (b, d) of their FTs (showing both real and imaginary parts) obtained for Ni-rich (Top: NiB15 and NiT15) and Ni-free (bottom: Bir and Tod) layered precursors and corresponding reflux products.

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Fig. 10. Comparison of Ni K-edge EXAFS spectra obtained for layered precursors
and reflux products with reference spectra. Sample colors as in Fig. 1.

1100

Fig. 11. Pair distribution functions obtained for Ni-rich (NiT10 and NiT20) and Ni-free (Tod) reflux products (top). Middle: d-PDFs relative to Tod PDF data; bottom: PDFs calculated for Ni (hydr)oxides (ICSD #76670 and #28101 for NiO and Ni(OH)₂, respectively). Intensities of experimental PDFs and of difference PDFs were normalized to the 2.9-3.1 Å peak.

Fig. 12. Intensity distributions calculated for ideal todorokite (ICDD #38-0475) with (a) only Mg in todorokite tunnels and (b) with Mg and Ni present in a 1:1 ratio in todorokite tunnels. In both cases, the model of Post et al. (2003) was modified assuming the complete filling of the O8 site by H₂O molecules (no cations) and the presence of 1.0 cation (Mg or Mg+Ni) in the Mg site.

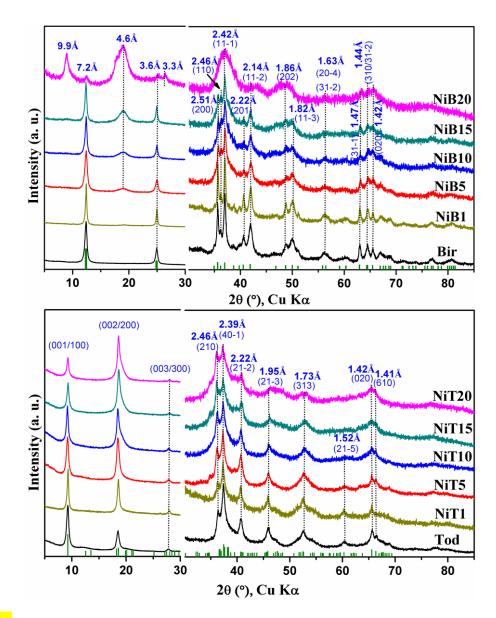


Fig. 1. XRD patterns of layered precursors (top) and reflux products (bottom). Colors
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and lower figures indicate the reflections of triclinic birnessite (ICDD #43-1456) and
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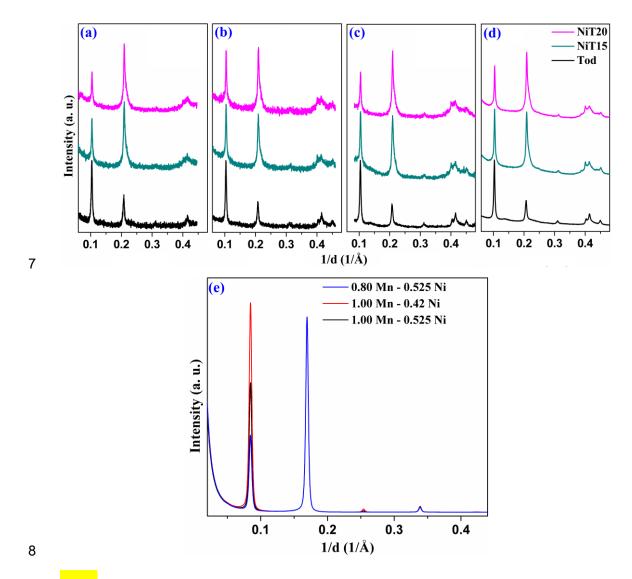


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(b) Ni K-edge (8.3 eV), (c) 12.0 keV, and (d) 18.0 keV. Intensities were systematically
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calculated for an asbolane containing 0.525 Ni(OH)₂ in MnO₂ interlayers. Contents of
Ni and Mn atoms are reduced by 20% at their respective absorption edges to simulate
the anomalous effect.

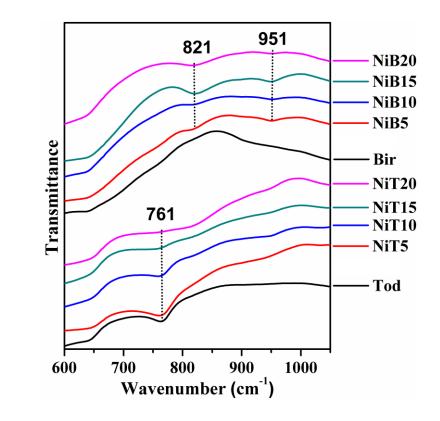




Fig. 3. FTIR spectra of layered precursors and reflux products (top and bottom,
respectively). Colors as in Fig. 1.

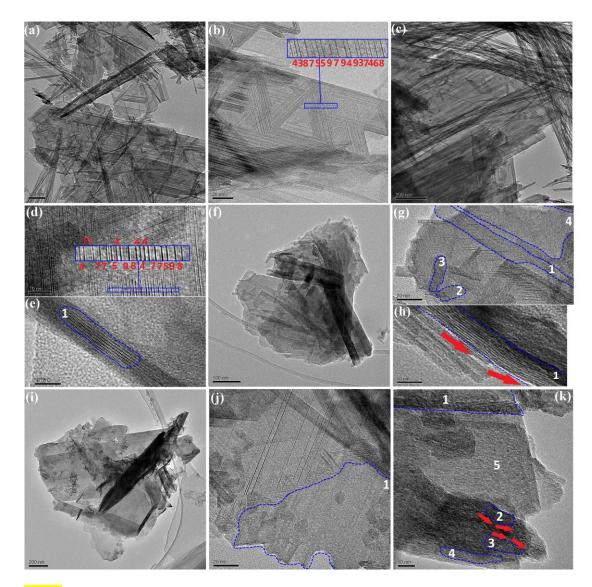


Fig. 4. HRTEM images of (a-b): Tod, (c-e): NiT1, (f-h): NiT5, (i-m): NiT1. Numbers
in (b) and (d) indicate the number of MnO₆ octahedra defining tunnel width. Arrows
in (i) and (l) indicate the direction of fringes bending.

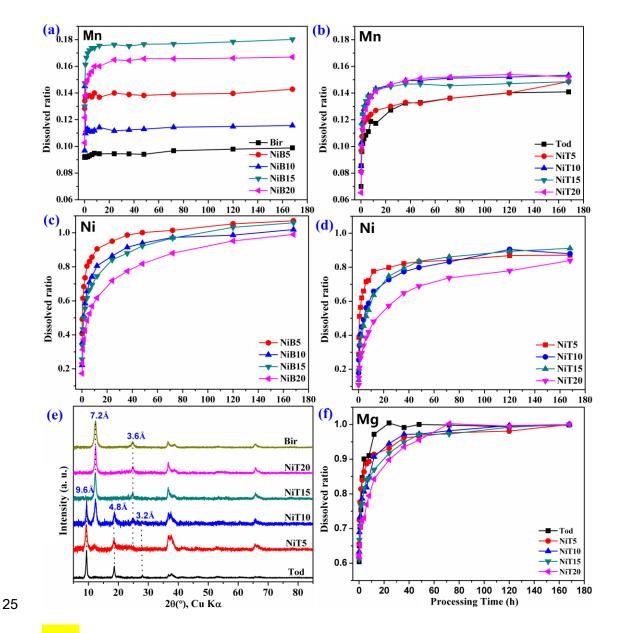


Fig. 5. Proportion of metal cations released during the nitric acid treatment: (a-b): Mn, (c-d): Ni, and (f): Mg. The amount of metal released is normalized to the sample overall metal content (see text for details). (e): XRD patterns of samples treated with nitric acid. Colors as in Fig. 1 except for Bir in (e).

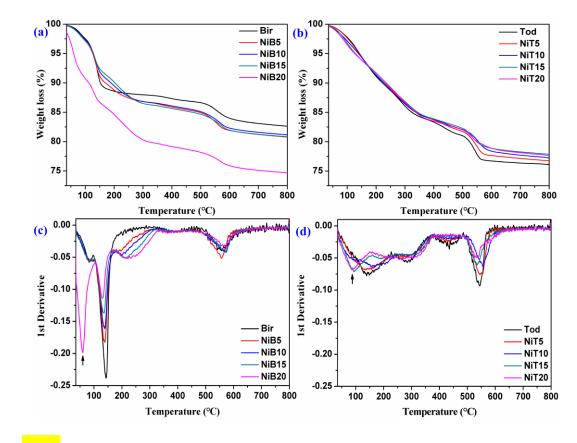
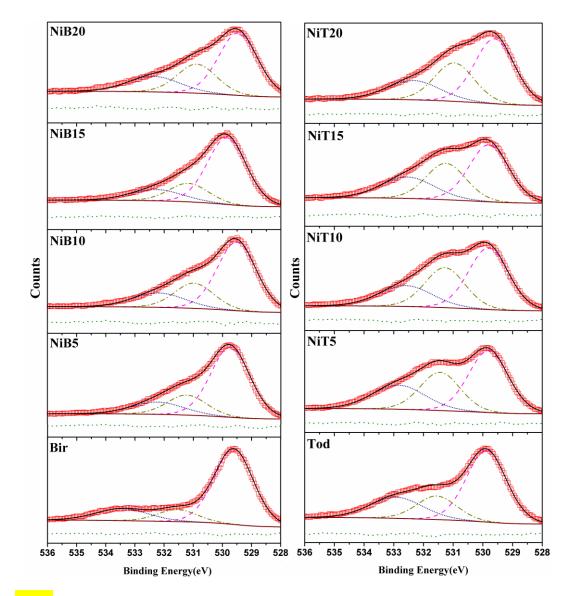


Fig. 6. Thermogravimetric analysis traces obtained for (a) layered precursors and (b)
reflux products, and their 1st derivatives (c and d, respectively). 1st derivative curves
were smoothed using a Savitzky-Golay filter and a 15 points window. Colors as in Fig.
1.



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Fig. 7. XPS O1s spectra obtained for layered precursors and reflux products (left and right, respectively). The thick solid curve represents the best fit to the data (open red squares). Peaks corresponding to O^{2-} , OH^- , and H_2O species are shown in the order of increasing binding energy as dashed purple, dotted-dashed green, and dotted blue lines, respectively. The green dotted line is the difference curve.

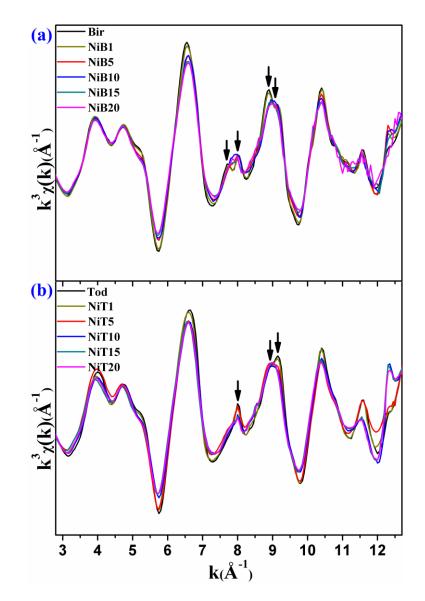


Fig. 8. Mn K-edge EXAFS spectra obtained for (a) layered precursors and (b) reflux
products. Colors as in Fig. 1.

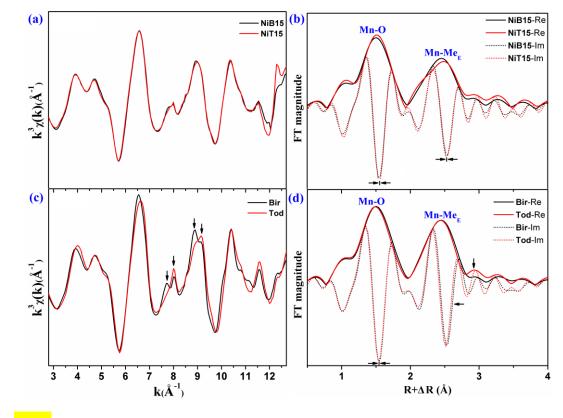


Fig. 9. Comparison of (a, c) Mn K-edge EXAFS and (b, d) of their FTs (showing both
real and imaginary parts) obtained for Ni-rich (Top: NiB15 and NiT15) and Ni-free
(bottom: Bir and Tod) layered precursors and corresponding reflux products.

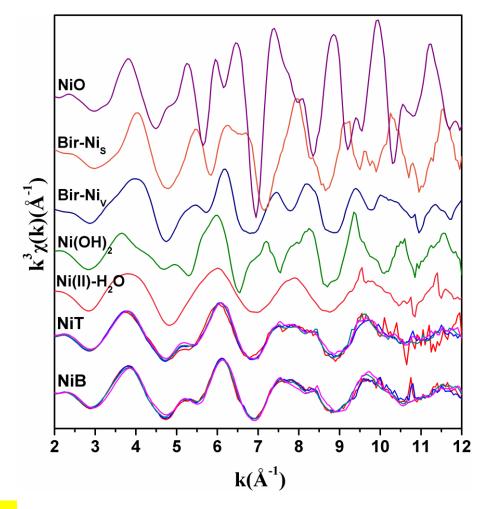
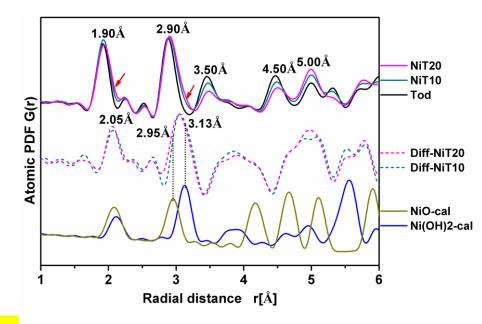
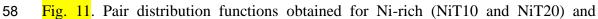


Fig. 10. Comparison of Ni K-edge EXAFS spectra obtained for layered precursors
and reflux products with reference spectra. Sample colors as in Fig. 1.





Ni-free (Tod) reflux products (top). Middle: d-PDFs relative to Tod PDF data; bottom:
PDFs calculated for Ni (hydr)oxides (ICSD #76670 and #28101 for NiO and Ni(OH)₂,
respectively). Intensities of experimental PDFs and of difference PDFs were
normalized to the 2.9-3.1 Å peak.

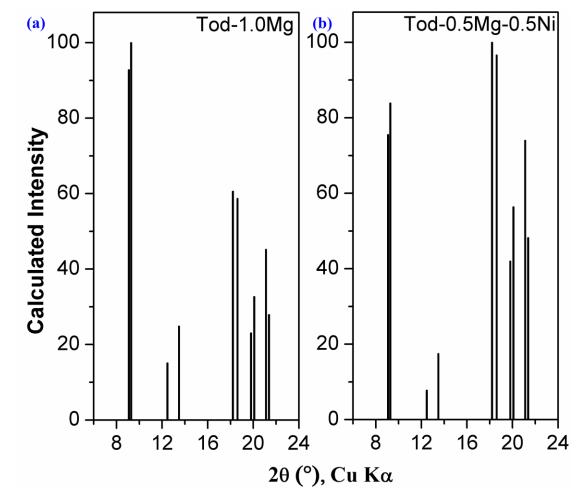


Fig. 12. Intensity distributions calculated for ideal todorokite (ICDD #38-0475) with (a) only Mg in todorokite tunnels and (b) with Mg and Ni present in a 1:1 ratio in todorokite tunnels. In both cases, the model of Post et al.(2003) was modified assuming the complete filling of the O8 site by H_2O molecules (no cations) and the presence of 1.0 cation (Mg or Mg+Ni) in the Mg site.

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Transformation of Ni-containing birnessite to tectomanganate:

Influence and fate of weakly bound Ni(II) species

Supplementary material

Geochimica Et Cosmochimica Acta

Zhongkuan Wu^{a,b}, Bruno Lanson^{b,*}, Xionghan Feng^a, Hui Yin^a, Zhangjie Qin^a, Xiaoming Wang^a, Wenfeng Tan^a, Zhongjun Chen^c, Wen Wen^d, Fan Liu^{a,*}

^a Key Laboratory of Arable Land Conservation (Middle and Lower Reaches of Yangtse River) Ministry of Agriculture, College of Resources and Environment, Huazhong Agricultural University, Wuhan 430070, China

^b Univ. Grenoble Alpes, Univ. Savoie Mont Blanc, CNRS, IRD, IFSTTAR, ISTerre, F-38000 Grenoble, France

^c Beijing Synchrotron Radiation Facility, Institute of High Energy Physics, Chinese Academy of Sciences, Beijing 100039, China

^d Shanghai Synchrotron Radiation Facility, Shanghai Advanced Research Institute Chinese Academy of Sciences, Shanghai 201204, China

* Corresponding author: <u>bruno.lanson@univ-grenoble-alpes.fr;</u> liufan@mail.hzau.edu.cn

Table S1

Sample	O1s (BE)	State	BE(eV)	FWHM(eV)	At.%
		O ²⁻	529.6	1.59	72.9
Bir	529.35	OH-	531.5	1.59	11.6
		H ₂ O	533.4	2.12	15.5
	-	O ²⁻	529.7	1.59	65.9
NiB5	529.72	OH-	531.2	1.59	18.8
		H ₂ O	532.2	2.12	15.3
	-	O ²⁻	529.5	1.59	60.8
NiB10	529.72	OH-	531.0	1.59	22.3
		H ₂ O	532.2	2.12	16.9
	-	O ²⁻	529.8	1.59	66.6
NiB15	530.06	OH-	531.2	1.59	18.5
		H_2O	532.3	2.12	15.0
	-	O ²⁻	529.5	1.59	55.6
NiB20	530.21	OH-	530.9	1.59	25.9
		H ₂ O	532.4	2.12	18.5
	-	O ²⁻	529.9	1.59	58.1
Tod	529.89	OH-	531.6	1.59	20.9
		H ₂ O	533.0	2.12	21.0
	-	O ²⁻	529.8	1.59	46.7
NiT5	529.84	OH-	531.4	1.59	29.0
		H ₂ O	532.8	2.12	24.4
	-	O ²⁻	529.8	1.59	47.6
NiT10	530.08	OH-	531.3	1.59	31.0
		H ₂ O	532.6	2.12	21.5
NiT15	530.15	O ²⁻	529.8	1.59	46.8

Relative proportions of O species in the near-surface of layered precursors and reflux products calculated by fitting XPS O1s data

	OH-	531.2	1.59	28.9
	H ₂ O	532.5	2.12	23.4
	O ²⁻	529.6	1.59	50.3
529.59	OH-	531.0	1.59	29.8
	H ₂ O	532.3	2.12	19.9
	529.59	H ₂ O O ²⁻ 529.59 OH ⁻	$\begin{array}{c c} H_2O & 532.5 \\ \hline O^{2-} & 529.6 \\ 529.59 & OH^- & 531.0 \end{array}$	$\begin{array}{c ccccccccccccccccccccccccccccccccccc$

Table S2

Relative proportions of NiB and Ni(OH)₂ species obtained from linear combination fitting of NiT K-edge EXAFS spectra using Ni(OH)₂ and corresponding NiB samples as standard references (Fig. S5).

Samples	NiBn /%	Ni(OH) ₂ /%	Error /%
NiT5	82.0	18.0	2.1
NiT10	86.5	13.5	1.0
NiT15	87.1	12.9	0.9
NiT20	85.3	14.7	1.1

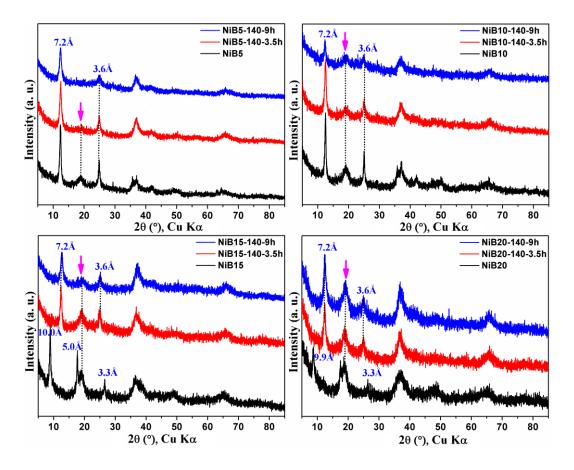


Fig. S1. XRD patterns of fresh layered precursors after heating to 140 °C for different durations

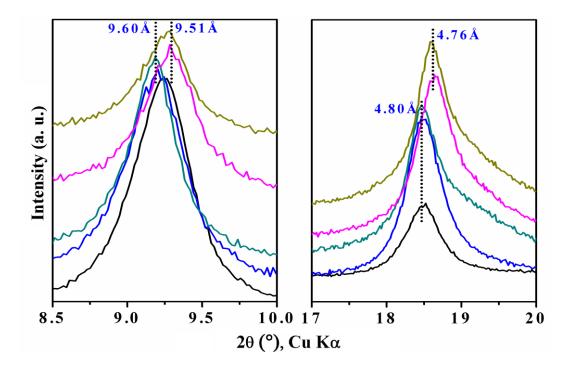


Fig. S2. XRD patterns of reflux products (zoom on low-angle regions). Colors as in Fig. 1.

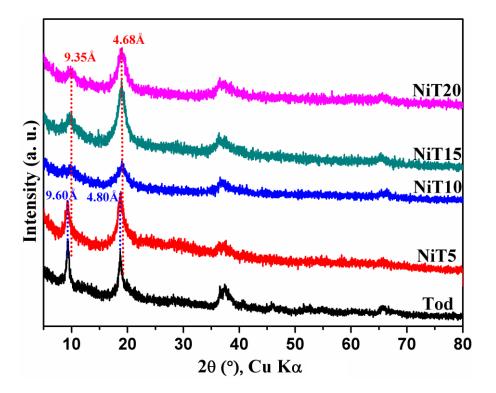


Fig. S3. XRD patterns of reflux products after heating to 140 °C for 12 hours. Colors as in Fig. 1.

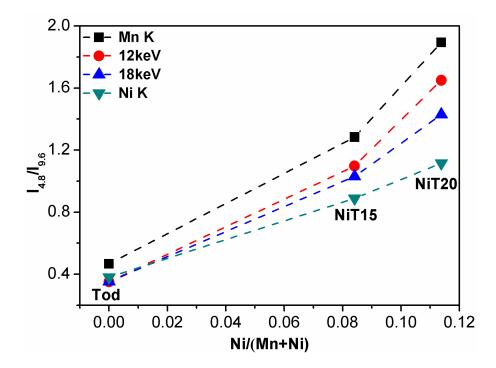


Fig. S4. Intensity ratio of the intensity of 4.8 Å and 9.6 Å peaks ($I_{4.8}/I_{9.6}$) in reflux products as a function of Ni content and X-ray energy (see Fig. 2 for the data).

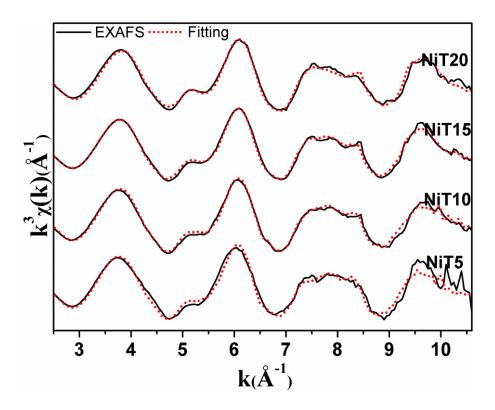


Fig. S5. Linear combination fitting of NiT K-edge EXAFS spectra using Ni(OH)₂ and corresponding NiB samples as standard references.

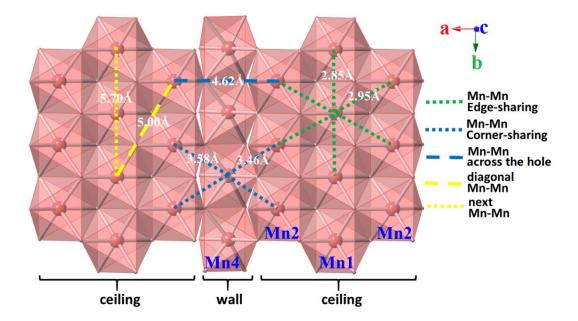


Fig. S6. Interatomic distances calculated for a variety of Mn-Mn pairs and ideal todorokite structure (ICDD #38-0475). The Mn4-Mn4 distance (4.48 Å) which is not reported in the figure is similar to the Mn2-Mn2 distance. Figure modified from Fig. 1(b) in Wu et al. (2019).

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Wu Z., Peacock C.L., Lanson B., Yin H., Zheng L., Chen Z., Tan W., Qiu G., Liu F. and Feng X. (2019) Transformation of Co-containing birnessite to todorokite: Effect of Co on the transformation and implications for Co mobility. *Geochim. Cosmochim. Acta* 246, 21-40.